Combined friction and wear reduction in a reciprocating contact through laser surface texturing

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Abstract

The aim of this study is to gain insights into the interactions between laser-textured surface pockets and friction and wear behaviour of an automotive piston-liner pairing. To do this, a recently developed, reciprocating, test apparatus was used to conduct wear tests under highly loaded conditions. Fused silica specimens with a range of pocket geometries were rubbed against a convex steel pad and the resulting friction and wear data were compared with those from a non-textured specimen. Contact conditions were set to remove the influence of initial surface roughness on texture behaviour. These tests showed that, as the specimen become worn and surface roughness increases, the contact progresses further into the mixed and boundary regime. This leads to a significant improvement in the relative performance of the textured specimens, showing reductions in friction of up to 70\%, compared with the non-texture case. This is consistent with previous results that have shown texture to have the effect of boosting film thickness in the mixed lubrication regime. Surface texture was also shown to reduce the volume of wear, by up to 69\% (corresponding to a change in wear coefficient from $2.67e^{-4}$ to $0.81e^{-4}$[mm$^3$/N·m]). Another, important, finding is that both friction and wear reduce monotonically as the sum of the pocket volumes along the stroke increases. This may aid texture design, since it means that individual pocket width and depth values can largely be ignored, so long as the overall volume is maximised. The only exception to this trend is when the pockets are larger than the contact area. In this case, friction increases due to a collapse of the lubricant film, while wear reduction remains unaffected - a discrepancy which may suggest that pockets reduce wear and friction via separate mechanisms.

1. Introduction

Surface texture has been investigated as a means of controlling friction and wear in sliding contacts for approximately fifty years \cite{1}, since it is relatively simple to implement and may be used in conjunction with other lubrication approaches. Currently, surface texture is receiving increased attention due to the need to improve the energy efficiency of automotive contacts \cite{2}, which, in turn, is driven by stricter CO$_2$ emission limits \cite{3}, and the pressing issues of climate change. It is also the case that methods of producing texture are becoming cheaper and more accurate \cite{4}. Despite all this, there is still a lack of firm evidence regarding the mechanisms by which surface texture affects tribological performance and this is limiting its development and application.
Investigations into how surface texture functions can typically be divided between those that relate to friction and those that relate to wear. The majority of early texture friction experiments were performed under full film hydrodynamic and elastohydrodynamic conditions [5–15]. Under these regimes, no conclusive improvement due to surface texture was demonstrated (with different studies showed the effectiveness of texture as being beneficial [13][Morris], detrimental [14], negligible [15]). Recently, however, more attention has been paid to measuring texture performance under mixed and boundary conditions, where several studies [16–21], have shown how pockets can lead to friction reductions of more than 50%. Furthermore, such conditions of mixed and boundary lubrication have become increasingly prevalent due to the recent trend towards lower viscosity lubricants aimed at reducing hydrodynamic losses [22].

The main explanations of how surface texture affects friction can be summarised as follows. In 1966 Hamilton and Allen [1,23], introduced the concept of “micro-irregularities” and suggested that surface texture can create resistance that acts to prevent fluid from escaping the contact. Following this, Tender proposed that surface patterns generate a “virtual step”, within the contact that resembles the discontinuous change in film thickness associated with a Rayleigh step bearing [24,25]. Later, when studying the behaviour of transverse roughness in EHL lubricated contacts, Morales-Espejel [26] and Greenwood [27] suggested that each asperity peak entering contact zone behaves as a flow exciter that closes and opens the inlet to allow varying amounts of fluid to enter. Years later, Morales-Espejel, Tripp and co-workers [28] observed that, under EHL lubrication conditions, oil is released from the pocket under pressure, as a result of elastic deformation of the material in the contact area, and this could increase load support and hence film thickness. In 2006, the concept of “inlet suction” was identified in Olver and Fowell’s work [29,30], whereby the reduction in pressure afforded by cavitation as lubricant expands into a pocket at the contact inlet leads to an increase in lubricant entrainment and hence film thickness. Very few of these mechanisms have been proved experimentally. In terms of experimental evidence, what is known so far is that pockets boost film thickness in the mixed and boundary regime [31], typically by a few tens of nanometres, which, due to the slope of the Stribeck curve in this lubrication regime, has a significant effect on friction [32]. Furthermore, a single pocket entrainment causes only a temporary increase in film thickness, which must be sustained by the entrainment of subsequent pockets otherwise it will become squeezed from the contact.

Studies into the effects of surface texture on wear have uncovered two main mechanisms, namely those of “lubricant replenishment” and “wear debris trapping”. The first of these relates to starvation – i.e. the reduction in film thickness due to an inadequate volume of lubricant present at the inlet [33], which is a believed to be an issue in automotive piston-liner contacts, caused by the rapid reciprocating motion of the contact and the fact that the oil must lubricate multiple rig contacts [34]. Starvation occurs when lubricant is pushed aside during one pass of the contact and does not flow back in time for the next pass [33]. In such situations, pockets on component surface can act as “micro-reservoirs” and increase the flow of lubricant back to the contact – a process known as ‘lubricant replenishment’ [35]. Several studies have investigated this mechanism. Blatter et al. [36] ran pin-on-disc tests on specimens with varying groove geometries, showing reductions in friction coefficient and wear rate, coupled with the increased wear life that were attributed to the grooves’ ability to store lubricant and replenish the track. In a more recent study, Borghi et al. [37] examined the chemical composition of textured and non-textured surfaces using Energy Dispersive Spectroscopy (EDS). This revealed significant quantities of Mg and Ca inside the dimples, thus demonstrating the retention of lubricant as the means by which the textured surfaces reduced friction and wear. Krupka et al. [38] also concluded that pockets can act as oil reservoirs capable of supplying lubricant into the contact region to overcome starvation. Their tests, carried out under EHL and mixed lubrication conditions, showed micro-dimpled surfaces overcame starvation and increased film thickness, which resulted in a measurable reduction in abrasive
wear. Further important work suggesting that pockets reduce friction by acting as reservoirs for lubricants, and hence aiding a “micro-wedge” effect, has been provided by Rahnejat and co-workers. Finally, the current authors observed that pockets carried lubricant into the cavitated region, and suggested that this could reduce starvation in reciprocating contacts, in which the cavitated region would otherwise be ingested by the inlet following reversal [16].

The second mechanism proposed to explain texture induced wear reduction is that of “debris trapping” whereby pockets remove particles, which would otherwise remain in the contact and accelerate surface damage. One of the first studies into this was carried out by Varenberg et al. using a flat-on-flat sliding contact under dry fretting conditions [39]. They showed wear debris being removed from the contact area and accumulating in the micro-textured pockets. Following this, they showed schematically how the pocket filling takes place: starting at the rim, followed by the centre and finally the base. In addition to this, Pettersson and Jacobson investigated the influence of surface texture on wear under starved and lubricated boundary conditions [18,19] and concluded that recesses can remove debris from the contact zone and hence delay or decelerate wear [19]. They used a reciprocating ball-on-plate tribometer and observed that grooves and squares parallel to the sliding direction reduced wear to negligible levels compared to an untextured surface. Further confirmation of pockets acting as debris traps has been obtained through the use of relatively recent techniques, such as SEM, that allow the morphology and chemistry of textured surfaces to be captured immediately after performing tribological tests. The first of these studies was carried out by Zum Gahr et. al [40] on plain and micro-textured steel/oxide-ceramic and self-mated oxide-ceramics sliding pairs. They noticed that instantaneous spikes in the friction force signal were present after 50 m of sliding for the untextured specimens but did not appear until after 400 m of sliding for the dimpled textured specimens. SEM micrographs of the worn surfaces obtained after the test showed that both fresh and partially compacted debris from the contact area had accumulated in the micro-dimples. Their conclusion was that the friction spikes observed early on in the untextured specimens sliding tests were due particle entrapment, whereas the eventual spikes in the texture surface tests were due to the detachment of layers, or groups, of wear particles which had accumulated in the micro-dimples.

As described above, the majority of investigations into surface texture have, so far, focussed on either friction or wear but not both simultaneously. Since texture is most effective in reducing friction under mixed and boundary conditions, where appreciable surface contact and wear occurs, it seems likely that the effects of pockets on friction and wear are interlinked. The aim of the simple experiments described in this study is to explore the relationship between these three parameters. We hope that this will provide fundamental insights which will inform how deliberately applied surface texture can improve the lubrication of piston/cylinder contacts, since these account for 45% of the frictional energy dissipated in a typical automobile (frictional losses in turn make up 11.5 % of the total fuel energy [2]). Furthermore, for the reasons mentioned above, it is believed that contacts in these components experience starvation and may therefore benefit from texture-enhanced lubricant replenishment. That being said, our findings may also be applied to other vehicle components, such as crankshaft bearings (which consume around 19% of a vehicle’s fuel energy [41]), as well as other conformal contacts in application such as compressors.
2. Experimental procedure

2.1. Test rig description

This study used a custom-built reciprocating test rig capable of measuring friction force under conditions that closely partially replicate those found in an automotive piston liner pairing, as detailed in [16]. The reciprocating rig’s main structural features are presented in detail in Fig. 1. This comprises a linear contact between a stationary convex AISI 52100 steel pad and a fused silica plate that slides along two linear bearings with a controlled amplitude of 28.6 mm and nearly sinusoidal velocity profile by means of a cam mechanism. Dead weights are attached to the silica pad specimen holder in order to generate an operating normal load, while a system composed of a heating circulator and two pumps allows the oil to be supplied directly to either side of the contact at a precisely controlled temperature and flow rate.

A high sensitivity load cell is located in direct contact with the steel pad holder. This holder was designed to allow the self-alignment of the steel pad against the fused silica counter-specimen while having a flexible base to facilitate load cell measurements. The friction force signal from the load cell is digitised and stored using LabView software.

A rotary position encoder fixed to the cam mechanisms’ shaft is used to measure reciprocating velocity and crank angular position. The correspondence between friction force and crank angular position was achieved by a custom-built 9 bit trigger system that synchronised measurement signals. This permits the simultaneous analysis of boundary friction at the reversal point and corresponding wear track position, with a precision of 0.7 degrees of crankshaft revolution.

![Fig. 1 Reciprocating test rig developed in-house](image)

The reliability of test results obtained using this configuration for both textured and non-textured surfaces has been confirmed through extensive repeat tests under different sets of conditions.

2.2. Test Parameters and Specimens

HPFS Standard Grade fused silica rectangular pads with dimensions of 125 × 15 × 5 mm were used as the reciprocating specimens in this study. Textured patterns, consisting in linear grooves orientated normal to the direction of sliding, were applied to these specimens. There are a number of
differences, between the materials used in this study and those found in commercial piston liners, where surfaces are typically cast-iron or aluminium with a typical roughness of 0.2–0.3 μm that results from honing. Furthermore, the use of coating such as DLC are increasing used to coat automotive liners. As discussed in Section 3, this means that the wear mechanism in these experiments differ from those occurring in current piston liner interfaces. However, the model contact between silica and steel was deliberately chosen in order to replicate the exact conditions used in three previous studies [refs]. This enables us to correlate our observations with previously established understanding of how texture induced friction reduction operates (it was important that we tested conditions where laser texture has a proven effect of tribological performance). An additional reason for using silica specimens is their uniform nature and surface finish (12 nm). This has enabled us to measure and characterise the wear behaviour accurately and to remove the influence of initial surface roughness on texture behaviour.

The pockets were produced using Laser Surface Texturing (LST) [4]. Also known as laser honing, this method employs an ultra-fast picosecond laser which emits optical pulses with a duration of 10 ps, frequency of 10 kHz and wavelength of 355 nm, and power of 5 micro joules. Such short pulse durations are vital in order to ablate micron sized portions from a surface before the material has time to undergo significant thermal changes. The resulting geometric uniformity is necessary since we have found that pockets with micron sized ridges, or with walls that are few degrees off perpendicular, result in significant underperformance.

Table 1 Geometry of the selected textured patterns.
The orientation of the pockets was selected based on previous experimental studies [16], which showed that grooves perpendicular to the direction of sliding are most effective in reducing friction under both mixed and boundary lubricated conditions. It is also important that pockets can become entirely trapped within the contact zone, since these perform better than those with pockets that are longer than the contact zone in the direction of travel [15,16,19]. In this study, three different pockets breadths were selected; firstly, two breadths of 40 µm and 80 µm, which are small enough to be completely enclosed within the contact area and secondly, a pocket breadth of 300 µm which is much larger than the elastic contact width. Table 1 summarises the geometries of the selected patterns together with the corresponding images, while Figure 2 shows an example of an interference image of the contact with an 80 µm breadth pocket close to the inlet of the contact (obtained by viewing the contact through transparent silica specimen with an optical microscope).

<table>
<thead>
<tr>
<th>Specimen No.</th>
<th>Breadth [µm]</th>
<th>Depth [µm]</th>
<th>Separation [µm]</th>
<th>Image</th>
<th>Sample name</th>
</tr>
</thead>
<tbody>
<tr>
<td>1</td>
<td>40</td>
<td>7</td>
<td>500</td>
<td>40x7x500</td>
<td></td>
</tr>
<tr>
<td>2</td>
<td>80</td>
<td>3</td>
<td>500</td>
<td>80x3x500</td>
<td></td>
</tr>
<tr>
<td>3</td>
<td>80</td>
<td>3</td>
<td>1000</td>
<td>80x3x1000</td>
<td></td>
</tr>
<tr>
<td>4</td>
<td>300</td>
<td>7</td>
<td>1100</td>
<td>300x7x1100</td>
<td></td>
</tr>
</tbody>
</table>

**Fig. 2** Interference image of the contact showing a pocket close to the inlet
The counterpart specimens – all identical AISI 52100 steel pads replicating the top piston ring –, are 10 mm long and curved in the direction of sliding resulting in a convergent-divergent form. Following a grinding process of the initial fully hardened sample, a minimum stock of 0.15 mm was removed. The 2 mm width side of the specimen was subsequently ground at a 40 mm radius using a dedicated jig, which allows for accurate polishing along the entire length of the pad without modifying the radius. The convex surface was mirror polished according to the roughness presented in Fig. 3.

![Three dimensional surface profile of a mirror-polished steel specimen](image)

In all friction force and wear tests, a normal load of 155 N was applied and a crank angular velocity of 1 Hz was maintained. Such high load, low speed, conditions were chosen in order to minimise lubricant entrainment, increasing surface contact and wear. Wear was also promoted by using an additive-free mineral oil, which was continuously supplied to the contact region while accurately controlling its temperature, and consequently viscosity, throughout the entire testing session. The lubricant properties alongside all other test parameters are listed in Table 2.

**Table 2** Experimental conditions. (note: the quoted lambda value is that for the non-texture specimen, calculated from the mid-stroke film thickness at the start of the test. As the test progress and wear takes place, the lambda value will increase due to roughening of the specimens).

<table>
<thead>
<tr>
<th>Load, W [N]</th>
<th>Crank angular velocity, [Hz]</th>
<th>Oil Temperature, ºC</th>
<th>Oil properties</th>
<th>Time length of the test, t [h]</th>
<th>Initial Lambda Ratio, Λ [-]</th>
</tr>
</thead>
<tbody>
<tr>
<td>155</td>
<td>1</td>
<td>80</td>
<td>Dynamic viscosity, η [mPa·s]</td>
<td>Kinematic viscosity, ν [mm²/s]</td>
<td>Density, ρ [g/cm³]</td>
</tr>
<tr>
<td></td>
<td></td>
<td></td>
<td>18.313</td>
<td>22.947</td>
<td>0.798</td>
</tr>
</tbody>
</table>

2.3. Experimental Protocol
Precise calibrations of the isometric load cell were performed both before and after each test was carried out. To do this, various known weights were attached to the steel pad holder in order to displace it in both tangential directions (tension and compression) and thus define the relationship between load and voltage. During testing, the average of these two was used to convert voltage output into friction force.

The fused silica and the steel specimens were consecutively sonicated in toluene and isopropanol before all traces of solvent were dried out using a heat gun. Wear tests, each with a duration of four hours were carried out on all the textured specimens presented in Table 1 as well as the non-textured reference. During each test, the friction force variation for one complete stroke was recorded after one minute and after three minutes, then every five minutes for the first half hour and every ten minutes for the remaining three and a half hours. At the end of the test, the cleaning procedure previously described was repeated and the topography of wear track around the reversal point was recorded using the Veeco Wyco NT9100 optical profiler. The Veeco’s area of analysis was a rectangle, 2 mm wide by 3.5 mm long. This rectangle was located such that, at reversal, its short axis coincided with that of the elliptical contact. In this way the region of highest wear fell within its borders. In the direction of sliding, the rectangle’s 2 mm width extend outside the wear track, thus encompassing a non-worn region, which could provide a datum from which the wear volume could be calculated.

3. Results and discussion

Many of the studies into the impact of surface texture on wear and frictional response have been conducted on ball-on-plate contact configurations using commercially available pin-on-disc tribometers [18,19,42,43]. These test conditions differ significantly from those found in an actual piston ring-cylinder liner pairing. The test apparatus developed specifically for the present research allows for the friction force variation on every stroke to be plotted as a function of crank angle throughout the wear test. This provides more detailed information than the previously obtained single point measurements. A graphically depicted comparison of the frictional response was thus possible at any time during the test. Figure 4 displays the friction force variation, captured for all four textured specimens together with the smooth reference, plotted as a function of crank angle, throughout the entire wear test.

Over the first few cycles of each test ($t_0$), the smooth pad exhibits the lowest friction force for the majority of the stroke compared with any of the textured specimens. This is explained as follows. At the start of the test, negligible wear has occurred on any of the sliding pairs and the composite roughness was approximately equal to its initial value of 12 nm. For a mid-stroke film thickness of around 38 nm (calculated by applying Dowson and Higginson’s Equation [44] for the non-textured pairing), this roughness gives nominal lambda values in the region of 2.65, showing the bearing contacts are operating in the full film hydrodynamic lubrication regime. This initial predominance of hydrodynamic lubrication is further demonstrated by the flat-bottomed “U” shape of the non-textured friction curve; when considering the sinusoidal nature of the varying entrainment speed, this shape must arise from the contact spending the majority of each reciprocation around the lowest point of the Stribeck Curve. As previously shown in [14,16], it is under these full film lubrication conditions, that surface texture can have the effect of increasing friction, and hence explains why the non-textured specimen performs the best, initially.
After approximately 5 minutes, and as soon as the first scars of wear appear on its surface, the non-textured reference begins to show the highest friction compared to all of the other textured specimens. This is because the appearance of wear signifies the transition from full film to mixed lubrication conditions due to an increase in composite surface roughness that is greater than the lubricant film thickness. It is under this regime that surface texture is most beneficial in terms of reducing friction, as proven by numerous experiments [14,16,17,45]. No further changes in ranking were recorded over the remaining four hours test, with the non-textured specimen continuing to exhibit the highest friction at every time step. Moreover, after approximately three hours of rubbing ($t_0 + 180$ min.), the non-textured bearing reaches the boundary regime, exemplified by a constant friction force along the entire stroke (i.e. the wear has advance to such an extent that the increasing hydrodynamic separation as the bearing accelerates towards mid-stoke is insignificant compared to the roughness).

Further inspection reveals that the most sparsely textured specimen ($80 \times 3 \times 1000$) displays the lowest friction force at the beginning of the testing sequence and the highest friction force by the end of the fourth hour, when compared with the other pocketed specimens. Conversely, the deepest and most dense pattern ($40 \times 7 \times 500$) displays the lowest friction from the first hour of the test onwards. This behaviour suggests that deeper and more densely packed pockets are desired towards the end of the stroke (i.e. in the boundary and mixed lubrication regime), whereas sparser and more shallow pockets are needed towards the middle of the stroke (i.e. in transition between mixed to full film regime), and surface pockets should in fact be avoided when the bearing is operating in full film lubrication regime.
For a better visualisation of the above mentioned trends, the mid-stroke friction force, captured between 60° and 120° crank angle, has been averaged and is plotted as a coefficient of friction against sliding time in Figure 5. Here, significant and constant reductions of up to 70% are evident over the last two hours of tests when comparing the best performing textured pattern (40x7x500) against the non-textured reference.

![Coefficient of friction vs time](image)

**Fig. 5** Variation of coefficient of friction (average values obtained between crank angles of 60° and 120°) with sliding time for all textured and non-textured fused silica specimens

Figure 6 shows the data from Figure 5 replotted in terms of percentage improvement for the four textured bearings compared with the non-textured reference. Here, the relative change in performance between the non-textured and texture specimens can be clearly observed. In addition to this, it can be seen that the pattern consisting of pockets wider than the contact in the direction of travel (300x7x1100) exhibits slightly higher friction than the other texture geometries over the initial stage of the test (*i.e.* during the first hour). This requirement that pocket should be less wide that the contact in order to provide greatest friction reduction is consistent with previous findings for texture specimens. However, it should also be noted that by the end of the test, there are no discernable differences between the friction performances of the 300 µm breadth pockets and the 80 µm breadth ones.
Another key observation from Figure 4 is the presence of significant peaks in the friction traces from certain pocket configurations. Rather than being experimental noise, these peaks correspond to lines of pockets entering the contact. Upon reaching a line of pockets, the contact area reduces considerably (by a factor of 10, which, under mixed lubrication regime has a significant impact on friction force) and, as a consequence, a local spike in friction is observed. As soon as each pocket leaves the contact, the film thickness increases and friction falls below the initial level. This effect is more pronounced for the textured specimens with a more sparse texture pattern, i.e. 300×7×1100 and 80×3×1000. As the frequency with which pockets enter the contact doubles (i.e. with specimens 40×7×500 and 80×3×500), the observation of this local friction effect becomes difficult. This relationship between friction force and the frequency with which textured features pass through a contact is an important topic which will be investigated in future research.
For a better visualisation of this transient effect, the friction force variation of the 80×3×500 textured pad at $t_0 + 1 \text{ min}$ was plotted alongside the non-textured reference as a function of distance travelled along the reciprocating stroke in Fig. 7. It can be observed that, although at the beginning of the stroke both pads exhibit similar friction, the first pocket placed immediately after reversal results in a considerable reduction of frictional response.

The friction force shows a temporary increase as each line of pockets enters the contact, however, as soon as the micro-features leave the contact, the friction force drops significantly compared with the non-textured pad. This behaviour is consistent with a recent study into the transient effects of surface texture [32]. Furthermore, since it is now known that the overall reduction in friction that results from multiple pocket entrainments is due to an increase in film thickness [31], it seems reasonable to assume that the observed transient variation in friction force is due to a corresponding variation in film thickness, caused by the entrainment of pockets. This assumption is supported by the fact that, in Figure 7, the optical profile image shows a region of wear that are localised around the edges of the pockets. The question remains as to exactly what mechanism is responsible for the film thickness increase as each pocket leaves the contact and so far our experiments have not been able to prove or disprove inlet suction, the Trip effect, or some other hydrodynamic mechanism as being responsible.

The counterpart steel specimens were analysed by Veeco Wyko surface profiler and no differences in average mean roughness greater than 5 nm were detected when comparing the five steel specimens. To exemplify this, Fig. 8 illustrates the surface topography of the steel pads before a test (a), after the four hours long wear test against a non-textured fused silica pad (b), and at the end of the test employing the pocketed specimen 40×7×500. Most probably, this low wear behaviour of the steel pads is due to the hardening process that the specimens underwent prior testing. For this reason, this study focuses on the wear observed on the silica liner specimen, and wearing of the steel ring specimen will be the subject of further testing.
Two-dimensional images of the typical wear tracks on the fused silica pads at the reversal point are shown in Figure 9. The 3.5×2 mm images were recorded, for all five tested specimens, at the same point along the stroke. This enabled accurate comparisons of the wear tracks and corresponding wear volumes. The first 0.2 mm slice on the horizontal axis, outside of the contacting region was considered the initial, unworn, level and was set as the datum surface before performing the wear measurements. Similar to IC engines and other applications comprising reciprocating bearings, wear occurred almost exclusively towards the end of the strokes due to a significant reduction in speed and hence lubricant entrainment in these regions. Indeed, optical profiles of the worn surfaces indicate that the greatest wear scar depths on the fused silica specimens occurred in all cases on the first 1.5 mm of the reciprocating stroke, immediately after reversal. Although Figure 9 presents topography for only one end of stroke, a near identical wear pattern was observed at the other end.

Despite similar wear scar dimensions in the direction of sliding (approx. 1.5 mm) for all the specimens, a qualitative and quantitative analysis of the volume of wear removed from the 3.5 mm² area around reversal, as indicated in Fig. 9, was performed. This shows the highest volume of wear for the non-textured specimen, followed by the 80×3×1000 sample, whereas the 300×7×1100 pattern proved most resilient to wear. This ranking of specimen performance leads us to conclude that there is a close connection between the volume of wear removed from the surface and the volume of oil each pattern brings to the contact during one stroke. Figure 10 presents this conclusion graphically: for each specimen, the volume of each pocket multiplied by the number of pockets replenishing the contact with each stroke is shown on the left, while the graph on the right displays the volume of wear removed over the four hour test.
It is immediately apparent from Figure 10 that, as the volume of oil brought into the contact increases from one pattern to the next, the volume of wear observed throughout the test decreases in perfect correlation. The non-textured specimen, with no pockets to replenish the contact area, exhibits the highest wear volume when compared with any of the textured surfaces. The 80×3×1000 pattern on the other hand shows a higher volume of wear removed from the surface when compared with a twice as dense pattern, 80×3×500, which in turn shows lower wear performance when compared with the much deeper 40×7×500 pattern. Finally, the 300×7×1100 textured pattern, which replenishes the contact with a volume of oil at least three times higher than any of the other specimens, exhibited the lowest wear volume throughout the test. It should be noted that this monotonic decrease would not have been observed if wear had been plotted against pocket length, depth or width individually.
To show the link between wear and friction behaviour, Figure 11 plots average friction coefficient over the last 60 minutes of testing alongside final wear volume for different pocket entrainment volumes. Here, a positive correlation can be observed for four out of five specimens – specifically, a lower average friction corresponds to a lower final wear volume. This correlation may at first seem to be due to the fact that pockets cause an increase in film thickness [31], and that this causes a reduction in both friction (as is known to be the case [16,32]), and also wear. However an exception to this trend is the 300×7×1100 pocketed specimen, which showed the lowest wear of all specimens but higher friction than denser pattern samples 40×7×500 and 80×3×500. It is important to note that this exception can be attributed to the 300 µm width of the high friction pockets, which are larger than the 240 µm long contact. Such wide pockets are known to allow oil to flow from the contact and hence cause a collapse of the film [16]. This is an important result and seems to suggest that pockets reduce wear and friction via different mechanisms.
It may be that the additional mechanism with which pockets reduce wear is that of debris trapping. It is also possible that pockets are reducing wear by replenishing lubricant and reducing preventing starvation, however this mechanism would not explain the discrepancy between wear and friction behaviour. Furthermore, it should be noted that the excess oil was supplied to the contact inlet by means of the lubricant pumps, so that starvation is only likely to result from the ingestion of the cavitated region following reversal.

It should be noted that the fused silica pads used in this study will wear differently to metal specimens, however, they were selected so that results presented here can be correlated with film thickness and friction results obtained in previous studies [16,31,32]. Several studies have looked at the evolution of piston/liner wear in using more representative materials [Priest] and the focus of our ongoing work is to extend this understanding to laser textured surfaces by testing rougher, metal-metal contacts using the methods described in this paper.

4. Conclusions

A specially developed test apparatus was employed to assess the effects of laser produced surface texture upon friction and wear behaviour. This involved reciprocating sliding tests on a linear contact for a duration of four hours. Various patterns comprising pockets orientated normal to the direction of sliding were compared against a non-textured reference specimen, and this lead to the following key conclusions:
For unworn contacts with high lambda values, such as those found at the start of testing, the presence of texture causes an increase in friction compared to the non-textured reference case (this agrees with previous research on full film conditions).

As specimens become worn and the contact enters the mixed lubrication regime, surface texture acts to significantly reduce the friction coefficient. Under these conditions, friction reductions of up to 70% were observed when comparing the best performing micro-textured pattern (40×7×500) against the non-textured reference. This suggests that surface texture performance is not only effective on mirror polished laboratory specimens, but is likely to work on actual machine components where it will limit the deleterious effects of wear on friction.

For the rectangular pockets tested, both friction and wear reduce monotonically as the sum of the pocket volumes along the stroke increases. This may aid texture design, since it means that individual pocket width and depth values can largely be ignored, so long as the overall volume is maximised. The only exception to this trend is when the pockets are larger than the contact area. In this case, friction increases due to a collapse of the lubricant film, while wear reduction remains unaffected and may suggest that pockets reduce wear and friction via separate mechanisms. This supports the theory that, in addition to providing a hydrodynamic film thickness boost, pockets act as reservoirs, which reduce starvation by increasing the supply of lubricant to the contact – particularly after reversal when the cavitated outlet becomes the inlet.

However, when textured patterns consist of pocket that are too wide, (compared to the contact width) and too sparsely distributed (for a given sliding speed) they act to locally increase the friction force as they pass through contact, generating local spikes of higher friction. This is due to the significant reduction of the contact area as a line of features enters the contact.

Considering wear reduction when selecting the optimal pattern is thus an important step but must be complemented with thorough analysis of the frequency at which pockets enter the contact, the relative size of the pockets compared to the contact, and the precise pocket location relative to after reversal point.

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