Revisiting Charge Transport in Spiro-OMeTAD using Space-Charge-Limited Current Measurements

Jason A. Röhr^{1,2}, Xingyuan Shi¹, Saif A. Haque², Thomas Kirchartz^{3,4}, Jenny Nelson¹

¹Department of Physics & Centre for Plastic Electronics, Imperial College London, South Kensington Campus, London SW7 2AZ, United Kingdom

²Department of Chemistry & Centre for Plastic Electronics, Imperial College London, South Kensington Campus, London SW7 2AZ, United Kingdom

³IEK5-Photovoltaics, Forschungszentrum Jülich, 52425 Jülich, Germany

⁴Faculty of Engineering and CENIDE, University of Duisburg-Essen, Carl-Benz-Strasse 199, 47057 Duisburg, Germany

Extracting charge-carrier mobilities for organic semiconductors from space-charge-limited conduction measurements is complicated, in practice, by non-ideal factors such as trapping in defects and injection barriers. Here, we show that by allowing the band-like charge-carrier mobility, trap characteristics, injection barrier heights and the shunt resistance to vary in a multiple-trapping drift-diffusion device model, a numerical fit can be obtained to the entire current density-voltage curve from experimental space-charge-limited current measurements on both symmetric and asymmetric Spiro-OMeTAD single-carrier devices. This approach yields a band-like mobility that is more than an order of magnitude higher than the effective mobility obtained using analytical approximations, such as the Mott-Gurney law and the Moving Electrode equation. It is also shown that where these analytical approximations require a temperature-dependent effective mobility to achieve fits, the numerical model can yield a temperature-, electric field- and charge-carrier densityindependent mobility. Finally, we present a novel analytical model describing trap limited current flow through a semiconductor, in a symmetric single-carrier device. We compare the obtained charge-carrier mobility and trap characteristics from this analytical model to the results from the numerical model, showing excellent agreement. This work shows the importance of accounting for traps and injection barriers, explicitly, when analysing current density-voltage curves from space-charge-limited current measurements.

I. INTRODUCTION

In recent years, there has been a great deal of interest in understanding the charge-carrier transport of weakly-doped or undoped semiconducting thin films based on disordered molecular materials, such as π -conjugated small molecules and polymers[1,2,3,4]. Studying space-charge-limited currents (SCLC) in single-carrier devices is an important means to understand electron and hole transport in such semiconductors. Although there have been many earlier reports on the charge-carrier transport of molecular materials using SCLC, the vast majority of these rely on using the Mott-Gurney (MG) law[5] which is not necessarily suitable for the analysis, since this law relies on idealised and trap-free semiconductors in devices with contacts which are not too injection limiting. In contrast, many molecular materials contain charge-carrier traps and single-carrier devices usually have some contact asymmetry due to the relative difficulty of ensuring truly identical and ideal contacts on both sides of the semiconducting thin film whose mobility has to be measured.

Analytical equations have been derived to describe charge-transport when either traps[6,7,8] or a built-in voltage resulting from contact asymmetry is present[9,10]. Mark and Helfrich derived an equation describing SCLC in the drift-dominated voltage regime when the charge transport of the semiconductor was limited by energetic disorder due to the localization of charge carriers in exponential tails in the band gap[8]. Fischer *et al.* have, however, pointed out that this so-called Mark-Helfrich equation is imprecise in estimating the trap characteristics since the equation does

not take diffusion of charge carriers into account[11], which may affect the current density in the Mott-Gurney regime when charge carriers are localised in exponential tails. For that reason, a good analytical model describing space-charge-limited charge transport in disordered semiconductor devices does not currently exist. Moreover, no analytical model exists that describes the common practical situation where both a large built-in voltage and traps are present. As a result, less suitable models, such as the MG law, are often used to analyse the experimental data, despite evidence for both trapping and built-in voltages. Whilst more sophisticated drift-diffusion simulations can be used to analyse the experimental data more accurately, only a small number of publications adopt that approach[3,12,13,14].

It is common within the organic semiconductor community to model the total density of states (DOS) using a Gaussian distribution (alternative approaches using master equations have also been used[15,16]). Hopping charge-transport in such a Gaussian DOS leads to a dependence of the (effective) mobility on temperature, electric field and charge-carrier density that has been parametrized by Pasveer *et al*, and subsequently used in a series of papers describing unipolar transport in organic semiconductors[17,18]. However, it has previously been shown that exponential tails (which is a good approximation of a Gaussian in a limited energy range), or even more complex shapes of the DOS, are sometimes required to describe certain aspects of the physics of organic semiconductor devices[3,12,13,14,19,20]. For instance, molecular dynamics and tight binding simulations on P3HT lead to a DOS that is neither a pure Gaussian nor a pure exponential tail, but rather a combination of both[21]. Thus, there is value in having models that allow us to change the shape of the DOS without starting with the process of parameterizing a mobility as a function of temperature, electric field and charge-carrier density as previously done, but rather account for these effects directly in the drift-diffusion model through the trap characteristics and interface statistics.

Here, we show how using drift-diffusion simulations improves the accuracy and physical interpretation of the determined mobility relative to traditional analytical approaches. We show that the obtained mobility only depends on temperature, electric field and charge-carrier density through the trap and injection characteristics. In addition, we derive a novel simple analytical description of the current density, which can be used to determine trap characteristics of a symmetric single-carrier device with high precision, when the semiconductor has an exponential trap distribution by examination of the low-voltage current regime, rather than from the current in the, intermediate, Mott-Gurney regime. A small molecule system commonly used as a hole selective interlayer for dye-sensitized solar cells and organometallic perovskite cells, Spiro-OMeTAD, is used for the study. Spiro-OMeTAD was chosen based on the expectation that its amorphous microstructure is largely insensitive to the layer thickness allowing for a thickness series to be performed. It is shown that when hole transport in Spiro-OMeTAD is characterised using either the drift-diffusion solver approach (including exponential tails) or by using the new analytical model proposed herein, the band-like hole mobility is determined to be both temperature, field and charge carrier density independent and substantially higher than the effective mobility that has been previously extracted using the more simple Mott-Gurney law[22].

II. BACKGROUND THEORY

First we discuss the concept of single-carrier devices, along with injection barriers and built-in voltages in such devices, and then we discuss some common analytical approximations, that have been used to analyse SCLC data.

A. Single-carrier devices



Figure 1 – Schematic of the energy level diagrams of a) symmetric electron-only device at thermodynamic equilibrium, where $q\phi_{inj}$ and $q\phi_{ext}$ represents the injection and extraction barrier heights respectively, E_c and E_V is the conduction and valence band edge, respectively, E_F is the Fermi level of the semiconductor at thermal equilibrium, and F is the electric field; b) symmetric electron-only device with enough applied voltage to assume drift-dominant transport (Mott-Gurney regime); c) asymmetric electron-only device at equilibrium showing the energy barrier arising from a built-in voltage, $qV_{bi} =$ $q\phi_{ext} - q\phi_{inj}$; and d) is a sketch of the total DOS including exponential tail states (the depth of the tails are given by their respective characteristic energies). Electron transport was shown for simplicity (Hole-transport is completely analogous). In b) and c), forward injection of electrons from the left-hand side is assumed.

When the selected electrodes form contacts both to the conduction band edge, or both to the valence band edge, of an intrinsic (or weakly-doped) semiconductor, the current voltage relation is governed by a single-carrier type, with the charge carrier species determined by the charge selectivity of the contact (an electron-only device is shown in fig. 1a). When a voltage, *V*, is applied across such a single-carrier device, excess charge-carriers, of the same species as the ones present at equilibrium, are injected, and the current is space-charge-limited (fig. 1b). For an electron-only device, if the work functions (WFs) of the contacts are larger than the electron affinity of the semiconductor, injection barriers will arise (shown as $q\phi_{inj}$ and $q\phi_{ext}$ in fig. 1a), and if these values are different (and non-zero), a built-in voltage will be present across the device ($qV_{bi} = q\phi_{ext} - q\phi_{inj}$) which will give rise to large diffusion currents when $V < V_{bi}$ (fig. 1c).

B. The Mott-Gurney law

The most commonly used method for fitting data obtained from single-carrier devices is to use the Mott-Gurney (MG) law. The MG law describes space-charge-limited drift current in an idealised single-carrier device, made from a trap-free and undoped semiconductor, in the intermediate voltage regime, and in the limit of barrier free injection (fig. 1b)[5,6,23]. The MG law is given by,

$$J = \frac{9}{8}\mu\varepsilon_{\rm r}\varepsilon_0\frac{V^2}{L^3} \tag{1}$$

where μ is the charge-carrier mobility for either electrons or holes (μ_n for electrons and μ_h for holes), ε_r is the static relative permittivity, ε_0 is the permittivity of free space, V is the applied voltage and L is the thickness of the semiconductor layer. However, organic semiconducting materials are rarely trap free, rendering the simple MG theory improper for describing charge transport for most realistic cases. It is however still common to assume an effective mobility, μ_{eff} , in the case where the MG law is used even though the semiconductor contains traps. This effective mobility is usually defined as[11,24],

$$\mu_{\rm eff} = \mu_{\rm band} \frac{\langle \rho_{\rm free} \rangle}{\langle \rho_{\rm total} \rangle} \tag{2}$$

where μ_{band} is the band-like mobility of the semiconductor (which will be approximately equal to μ in eq. 1 in an ideal case), $\langle \rho_{\text{free}} \rangle = L^{-1} \int_{0}^{L} \rho_{\text{free}}(x) dx$ is the spatial average of the free charge-carrier density and $\langle \rho_{\text{total}} \rangle = L^{-1} \int_{0}^{L} [\rho_{\text{free}}(x) + \rho_{\text{trapped}}(x)] dx$ is the total charge-carrier density across the thickness of the semiconductor. Since the effective mobility in eq. 2 is defined from the ratio of the free to total charge carrier density, the effective mobility is usually electric field-, and temperature-dependent. However, the band mobility is not inherently dependent on either of these quantities[19]. Note, that this definition of the effective mobility does not account for non-ideal injection.

C. The Mark-Helfrich equation

One of the few analytical equations which describes SCLC charge transport in a semiconductor with traps, assumed to be distributed as exponential tails of states in the DOS, is called the Mark-Helfrich equation[6,8]. For an electron-only device it is given by,

$$J = q^{1-l} \mu_{\rm n} N_{\rm eff} \left\{ \frac{\varepsilon_{\rm r} \varepsilon_0 l}{N_{\rm t}(l+1)} \right\}^l \left\{ \frac{2l+1}{l+1} \right\}^{l+1} \frac{v^{l+1}}{L^{2l+1}}$$
(3)

where *q* is the elementary charge, $l = E_{ch}/k_BT$, with E_{ch} being the characteristic energy of the exponential tail (see fig. 1d and eq. S30 and S31), N_{eff} is the effective density of states, and N_t is the trap density (per unit volume), and k_BT is the thermal energy. Equation 3 predicts that exponential tail states in the band gap give rise to a stronger power-law dependence of voltage on current than expected from the MG law in the intermediate voltage regime. However, it was recently shown that this equation is not accurate since it fails to account for diffusion currents, which can make a significant contribution to the total current, especially when traps are present[11].

D. Built-in voltages

Besides assuming the semiconductor to be trap free, the MG law also assumes that there are negligible energy barriers for both injection $(q\phi_{inj})$ and extraction $(q\phi_{ext})$ from the metal contact into the semiconductor. This criterion can however rarely be met in real experimental cases, and a finite injection barrier can have a significant influence on the probed charge-transport behaviour. With $q\phi_{inj/ext} = 0$ eV being assumed, the MG law therefore also implicitly requires that there exists no built-in voltage, V_{bi} , across the device arising from a difference in the WFs, and hence the injection barrier heights, $V_{bi} = q\phi_{ext} - q\phi_{inj}$ (fig. 1c)[5], a feature which is often difficult

to achieve from a practical device fabrication perspective. The limited choice of contact materials available to match very deep (> ~6 eV below the vacuum level) or very shallow (< ~2 eV below the vacuum level) transport levels often results in contacts with poor injection properties. A common hole-selective material for organic optoelectronic devices is the conductive polymer blend, poly(3,4-ethylenedioxythiophene) polystyrene sulfonate (PEDOT:PSS). Literature values for the WF of this conductive polymer range between 4.8 and 5.2 eV[25,26]. With many semiconducting polymers and small molecules having their highest occupied molecular orbital (HOMO) levels lying much deeper than 5.2 eV below the vacuum level, this interlayer material will not form a perfectly injecting contact with such materials. Another popular hole-selective material is MoO₃, with literature values for the WF ranging between 6 and 6.9 eV[27,28]. Sandwiching a deep HOMO (> 6 eV below the vacuum level) material between these two contact materials, for example, would allow for Ohmic injection from MoO₃ into the HOMO level, whereas the PEDOT:PSS will form an injection-limited contact, while at the same time resulting in the formation of a $V_{\rm bi}$ across the device due to the difference in the contact WFs (fig. 1c). Due to lack of contact materials it is common to measure SCLC on asymmetric single-carrier devices, since such devices are more realistic to fabricate than symmetric single-carrier devices (fig. 1c). However this $V_{\rm bi}$ will greatly affect the current density-voltage (J-V) curves at low and intermediate bias voltages until the internal voltage is overcome[9,10].

In order to correct for the built-in voltage, an effective applied voltage, $V_{\text{eff}} = V - V_{\text{bi}}$, where the V_{bi} is used as a fitting parameter in the MG law (or other analytical equations of variant types, such as the MH equation or the Murgatroyd equation[29]), is commonly used[30,31]. This indirect approach is however rather uncertain, since the band diagram at V = 0 V for a symmetric device and $V = V_{\text{bi}}$ for an asymmetric device are not necessarily similar. At a significant applied bias such that the built-in voltage is overcome, and if, hypothetically, the value of V_{bi} is exactly known, then using this effective voltage approach is a good approximation. The V_{bi} is however rarely known precisely in advance, and is sometimes estimated by shifting the voltage axis to a regime where J varies with V^2 which can eventually lead to misinterpretation of carrier mobility values, especially when traps are present.

Analytical drift-diffusion equations have been derived to account for built-in voltages, directly, in intrinsic single-carrier devices at low voltages directly, such as [9,10],

$$J = \frac{q\mu N_{\rm eff}(\phi_{\rm ext} - \phi_{\rm inj} - b - V) \left\{ \exp\left(\frac{qV}{k_{\rm B}T}\right) - 1 \right\}}{L \exp\left(\frac{q\phi_{\rm inj}}{k_{\rm B}T}\right) \exp\left(\frac{qb}{k_{\rm B}T}\right) \left\{ \exp\left(\frac{q\phi_{\rm ext} - q\phi_{\rm inj} - qb}{k_{\rm B}T}\right) - \exp\left(\frac{qV}{k_{\rm B}T}\right) \right\}}$$
(4)

where *b* is the voltage reduction due to band bending at the injecting interface (the derivation of eq. 4 is shown in the supplementary information[32] and in ref. [10]). Assuming the V_{bi} is large enough so that band bending at the interface is negligible, b = 0, and the injection barrier height at the injection point is zero, $q\phi_{inj} = 0$ eV, eq. 4 is reduced to a simpler form

$$J = \frac{q\mu N_{\rm eff}(V_{\rm bi}-V)\left\{\exp\left(\frac{qV}{k_{\rm B}T}\right)-1\right\}}{L\left\{\exp\left(\frac{qV_{\rm bi}}{k_{\rm B}T}\right)-\exp\left(\frac{qV}{k_{\rm B}T}\right)\right\}}$$
(5)

which is a useful equation for determining the built-in voltage of a single-carrier device[9]. Equation 5 describes the current density of an asymmetric intrinsic semiconductor device for all $V < V_{\rm bi}$. However, it cannot describe the situation where traps are present. Even though analytical models exist to describe both a built-in voltage and traps separately, no analytical model exists to describe the common practical situation featuring both traps and a large built-in voltage.

E. Low voltage regime

The final set of analytical models, presented here, apply to the low voltage regime (from a symmetric single-carrier device) where, in practice, a linear dependence of *J* on *V* is often observed. It has previously been shown that linear currents observed in SCLC *J*-*V* curves are due to one of, or a combination of, the following effects: flow of equilibrium charge-carriers at low voltages in symmetric single-carrier devices (the Moving Electrode equation)[33,34,35], a large increase of the charge-carrier density away from the equilibrium value due to doping[35,36], or bulk saturation of charge carriers inside the device either at high bias or with large injection barriers[37].

In the absence of traps, doping and a built-in voltage, the current density at low voltages is given by the Moving Electrode (ME) equation^[33,34,38,39,40],

$$J = 4\pi^2 \frac{k_B T}{q} \mu \varepsilon_{\rm r} \varepsilon_0 \frac{V}{L^3}.$$
 (6)

Linear currents can however also arise at low voltages in single-carrier devices through shunts due to low resistance pathways through the semiconducting film. This linear current density is given by Ohm's law as

$$J = (R_{\rm P})^{-1} V / L \tag{7}$$

where R_P is the shunt resistivity, or parallel resistivity (in units of Ωcm^2). Shunt currents are especially relevant for asymmetric devices with a large V_{bi} since the current density at low voltages is greatly reduced in these devices (as will be obvious later in this study).

In principle, the total current density of an asymmetric single-carrier device could be calculated by using a combination of eqs. 1 (with the $V - V_{bi}$ correction), 4, and 7, if the semiconductor in the device is intrinsic (a symmetric device could likewise in principle be calculated using a combination of eq. 1 and 6). However, this description of the current density does not account for the presence of traps, which have—on numerous occasions—been shown to be present in organic semiconductors [14,⁴¹].

III. ANALYTICAL RESULTS

Ref. [40] has shown that the spatial distribution of the equilibrium charge-carrier density of an intrinsic and trap-free single-carrier device, n, can be described, to a very good approximation, by,

$$n = \frac{2\pi^2 \varepsilon_{\rm r} \varepsilon_0 k_{\rm B} T}{q^2 L^2} \left[\cos^2 \left\{ \pi \left(\frac{x}{L} - \frac{1}{2} \right) \right\} \right]^{-1} \tag{8}$$

where *x* is the distance from the injecting contact. As we shall show, equation 8 precisely describes the charge-carrier density in the middle of the device, x = L/2, but produces unphysical singularities at the contacts, x = 0 and x = L, as seen in fig. 2a. By taking x = L/2, the equilibrium charge-carrier density, in the middle of the device, is given by,

$$n_0 = \frac{2\pi^2 \varepsilon_{\rm r} \varepsilon_0 k_{\rm B} T}{q^2 L^2}.$$
 (9)

A comparison of eq. 9 with numerically calculated values of the electron density is shown in fig. 2b (solid lines).

The arithmetic mean of eq. 8, $\langle n \rangle = L^{-1} \int_0^L n \, dx$, cannot be calculated, since the integral does not converge. However n^{-1} can be integrated and, thus, the harmonic mean of the charge-carrier density can be written as,

$$\langle n \rangle_{\text{ana.}} = \frac{1}{\frac{1}{L} \int_0^L (n)^{-1} dx}.$$
 (10)

This yields,

$$\langle n \rangle_{\text{ana.}} = \frac{4\pi^2 \varepsilon_{\Gamma} \varepsilon_0 k_{\text{B}} T}{q^2 L^2}$$
 (11)

which is, of course, just $2n_0$. Upon insertion of this charge-carrier density into the drift current equation, $J = q\mu_n \langle n \rangle_{\text{ana.}} \frac{V}{L}$, we obtain the ME equation, eq. 6. The ME equation is a very good approximation of the current density in an intrinsic single-carrier device at low voltages. Equation 9 is a very good approximation of the charge-carrier density in the middle of device at low applied voltages, since the overall shape and magnitude of the charge-carrier density profile does not change significantly between 0 and 0.1 V (see fig 2a).



Figure 2 – a) Electron density as a function of spatial position of a trap free device and a device with localized states in the form of exponential tail trap states in comparison with the eq. 8. b) Average electron densities, $\langle n \rangle$, of the numerical calculation (red dashed line) and the analytical approximation, eq. 11 (green dashed line), and electron density in the middle of the device, n_0 , from the numerical calculation (solid red line) and from eq. 9

(solid green line). c) Schematic of the description of the localised states in the band gap. In this case T = 0 K is assumed for the representation of the occupied density of states. d) Electron density in the middle of the device calculated using eq. 15 (solid lines) and from numerical calculations (dashed lines) when traps are included, for three values of E_{ch} as indicated, 0.04 eV, 0.06 eV and 0.08 eV.

It is important to note that the harmonic mean of the charge-carrier density, $\langle n \rangle_{ana.}$, does not give the same value of the arithmetic mean of the charge-carrier density of a device calculated using a numerical approach, $\langle n \rangle_{num}$, since the device boundaries are not correctly described in the analytical approximation (since the charge-carrier density at the boundary is set by the effective density of states in the numerical calculations and blows up to infinity in eq. 8). For this reason, the ME eq. cannot be directly used to determine the charge-carrier density of the entire device, but only in the middle of the device[³⁵]. Figure 2b shows that $\langle n \rangle_{num} > \langle n \rangle_{ana.}$, in general, for an intrinsic and trap-free device.

We now proceed to derive a formula for the current-voltage response of a single-carrier device in the low voltage regime in the presence of traps. The following derivation follows from the assumptions of trapped and free charge-carrier density statistics, in exponential tails, given by Mark and Helfrich[8]. For simplicity, the derivation is given for the case of electron transport. The derivation in the case of hole transport is completely analogous. The density of localized states in the band gap (electron traps) per unit volume and energy interval, *h*, far away from the device contacts, i.e., in the middle of the device (x = L/2), is described by,

$$h = n_{\rm t} \exp\left(\frac{E}{E_{\rm ch}}\right)$$
 (12)

where $n_t = N_t/E_{ch}$ is the trap density per unit energy right below the conduction band edge, and E is the energy measured below the conduction band edge (note that the conduction band edge is set to zero, $E_c = 0$, and the energy increases upwards in energy, as shown in fig 2c). Given that N_t is much larger than the total amount of charge carriers at absolute zero, n_0 (T = 0 K), it is safe to assume that approximately all of these charge carriers will be trapped and will fill the exponential tail states up to a quasi-Fermi level, $E_{F,t}$. Within this assumption we can write,

$$n_0 = \int_{-\infty}^{E_{\rm F,t}} \frac{N_{\rm t}}{E_{\rm ch}} \exp\left(\frac{E}{E_{\rm ch}}\right) dE = N_{\rm t} \exp\left(\frac{E_{\rm F,t}}{E_{\rm ch}}\right).$$
(13)

As the temperature is increased (T > 0 K), some charge-carriers will escape the traps and will be free to conduct. Given that the Fermi level governing free charge carriers is more than a few k_BT away from the conduction band edge, we can describe the free charge-carrier density using Boltzmann statistics (again implying $E_c = 0$),

$$n_{\rm free} = N_{\rm C} \exp\left(\frac{E_{\rm F}}{k_{\rm B}T}\right)$$
 (14)

where $N_{\rm C}$ is the effective density of electron states at the conduction band. Assuming that $E_{\rm F} = E_{\rm F,t}$ which is a fair assumption given that $E_{\rm ch} > k_{\rm B}T$, we can combine the above two expressions and describe the free charge-carrier density in terms of the total charge-carrier density as,

$$n_{\rm free} = N_{\rm C} N_{\rm t}^{-l} n_0^l \tag{15}$$

where $l = E_{ch}/k_BT$. Figure 2d shows a comparison of eq. 15 with numerical calculations of the charge-carrier density as the characteristic energy is varied from 0.04 eV to 0.08 eV (with a fixed

 $N_{\rm C}$ and $N_{\rm t}$). Since the current is only governed by the free charge-carrier density, we can now describe the drift current as,

$$J = q\mu_{\rm n} n_{\rm free} \frac{V}{L} = q\mu_{\rm n} N_{\rm C} \{ N_{\rm t}^{-l} n_0^l \} \frac{V}{L}.$$
 (16)

Since we are only concerned about the trap density far away from the contacts, we can describe the total charge-carrier density by eq. 9. We then obtain,

$$J = q^{1-2l} \mu_{\rm n} N_{\rm C} \left\{ \frac{4\pi^2 \varepsilon_{\rm r} \varepsilon_0 k_{\rm B} T}{N_{\rm t}} \right\}^l \frac{V}{L^{2l+1}}$$
(17)

which describes the current density at low applied voltages when some charge carriers are trapped in exponential tails. Note that in this case we do not need to introduce the concept of an effective mobility since the reduction of free charge-carriers are included implicitly through $n_{\text{free}} = N_{\text{C}}N_{\text{t}}^{-l}n_{0}^{l}$. μ_{n} is, for that reason, a band-like mobility. Equation 17 reduces to eq. 6 in the trap free limit. The accuracy of estimating trap densities and energies using eq. 17 in the low voltage regime is compared to the accuracy of estimating traps with the MH equation in the intermediate voltage regime is discussed in the supplementary information[32]. It is shown that eq. 17 is more precise in describing charge transport in a semiconductor containing traps (in the investigated parameter space), and can estimate the characteristic energy and trap density more accurately. The reason why eq. 17 works especially well, is due to the fact that the free charge-carrier density is very well described in the middle of the device using the surprisingly simple trap statistics (see figs. 2a and b), and the fact that this middle region of the device dominates the ME current.

IV. EXPERIMENTAL RESULTS

In order to make a comparative evaluation of the different analytical approaches and the numerical drift diffusion simulation, we prepared and measured a set of hole-only devices based on the small molecular semiconductor Spiro-OMeTAD (fig. 3a). We fabricated both symmetric and asymmetric hole-only devices in order to test the theoretical models over a wide voltage range of *J-V* data for the different device types. All devices were fabricated using a similar front contact consisting of a thin layer of PEDOT:PSS (~30 nm) spun cast on top of a predefined tindoped indium oxide (ITO) covered glass slide. The ITO was cleaned by ultrasonication in acetone and isopropanol for 10 minutes, respectively, prior to PEDOT:PSS deposition. Thin films of Spiro-OMeTAD of varying thicknesses were spun cast from chlorobenzene onto the ITO/PEDOT:PSS contacts under atmospheric conditions. For the asymmetric devices, a back contact consisting of a thick layer of aluminium (150 nm) was evaporated under high vacuum (of $\sim 10^{-6}$ mbar). For the symmetric devices, a thin layer of MoO_3 (30 nm) followed by a thick layer of Al (150 nm) was used as a back contact. Two types of steady-state J-V experiments were performed: at room temperature and at varying temperature using a cryostat. The current of the samples was recorded using a Keithley SMU 236 in a nitrogen atmosphere for the room temperature measurements and in a helium atmosphere for the cryostat-based temperature-dependent measurements. The structures of the studied single-carrier devices are depicted in fig, 3b.



Figure 3 – a) Molecular structure of Spiro-OMeTAD, b) symmetric and asymmetric holeonly devices employing MoO_3/Al and Al back contacts respectively (energy levels are taken from literature). The WFs of MoO_3 and PEDOT-PSS will shift to match the HOMO of the organic compound through Fermi-level pinning, resulting in estimated built-in voltages of 0 V and 0.9 V respectively for the two devices. The energy levels are all given relative to the vacuum level.

Based on the values for the contact WFs shown in fig. 3b, one could expect that the $V_{\rm bi}$ of the socalled symmetric Spiro-OMeTAD device is around 1 V, actually rendering the device asymmetric. However, Fermi-level pinning between a metal contact and an organic compound whose HOMO level is shallower than the contact WF, will shift this contact WF to match the HOMO level (upon thermodynamic equilibration), rendering both effective WFs to be equal to the HOMO level and the built-in voltage therefore to be zero. Again, assuming Fermi-level pinning, the built-in voltage would be estimated to be $\approx 0.9V$ for the asymmetric device (using a WF value for PEDOT:PSS of 5.2 eV). However, interfacial states (potentially bearing considerable dipole moments) between the contacts and the organic material, and the highly reactive nature of aluminium under ambient atmosphere, might give rise to deviations from these estimated $V_{\rm bi}$ values.



Figure 4 – a) Forward and reverse bias *J*-*V* curves obtained from SCLC measurements of symmetric (grey circles) and asymmetric Spiro-OMeTAD hole-only devices (orange and red circles) of 200 and 230 nm respectively. Since the forward and reverse bias curves overlap for the symmetric device, only the forward bias curve is shown. Mott-Gurney law forced "fits" and Ohm's law fits are shown as solid and dashed lines respectively. b) plots of $m = (d \log J) / (d \log V)$ of the asymmetric data in a) against voltage. c) *J*-*V* curves of symmetric Spiro-OMeTAD devices at temperatures varying from 200K to 300K in steps of 20 K, showing fitting with the Mott-Gurney law (eq. 1, solid lines) and either the moving electrode equation (eq. 6, dashed lines) or eq. 17. d) *m*-*V* curves of the *J*-*V* curves shown in c). The high-voltage slope values approach a value larger than 2 in both a) and c).

Figure 4a shows the experimental *J-V* curves of a symmetric 200 nm hole-only device and an asymmetric 230 nm device. In contrast to the non-rectifying behaviour observed for the current density between the forward and reverse bias in the symmetric device, a significant rectifying behaviour is present for the asymmetric device. Since a large asymmetry between the forward and reverse bias current is observed, a built-in voltage is present and the current at low voltage must be given by eq. 4 (or eq. 5 given the current is not limited by injection). The local slope of the *J-V* curve on a log-log scale can be given by,

$$m = \frac{d \log J}{d \log V} \tag{18}$$

such that $J \propto V^{m(V)}$. It is seen from the slopes of the *J*-*V* curves on a log-log scale in fig. 4b that the forward bias current for the asymmetric device goes from a linear dependence at low voltages (m = 1), to a large peak at intermediate voltages $(m \cong 30)$, and eventually approaches values $(m \cong 3.7)$ which are larger than what would be expected from trap-free behaviour (m = 2) at high voltages[7]. Also, the reverse bias current is seen to be linear with voltage over the whole regime for the asymmetric device. From the observation of the linear regime in both the forward and reverse bias regime of the symmetric device, it is clear that the low voltage current cannot be explained by eq. 4 alone (fig. S1). However the current can be modelled as a combination of

diffusion currents due to a large V_{bi} and a shunt current (eq. 7). That the current must include a shunt term at low voltages can also be seen from the very sharp onset to the large peak at forward bias (fig. S2) along with the large overlap between the forward and reverse bias current at low voltages. Based on this reasoning, the linear currents at very low voltages can be described, in both forward and reverse bias, using eq. 7 (dashed lines) and a value for the shunt resistivity of $R_P = 18 \text{ M}\Omega \text{cm}^2$ can be extracted. The deviation of the power-law dependence of the *J-V* curves away from m = 2 at high voltages (fig. 4b) gives evidence that the charge transport is governed by traps[6,8], as traps in the form of exponential tail states gives rise to an increase in the slope away from 2 (eq. 3)(fig. S1).

Since it is quite evident from the slope at high voltages that a proper charge-transport analysis cannot be achieved using the simple MG theory, we carry out numerical fitting using a driftdiffusion solver (see supplementary information for details[32]). To reduce the uncertainty in fitting parameters, especially regarding trap states, we analyse *J-V* data as a function of both temperature and semiconductor thickness. Figure 4c shows the series from a 190 nm Spiro-OMeTAD symmetric hole-only device at temperatures varying from 200 K to 300 K along with linear fits with the ME equation or eq. 17 (dashed lines) and fits with the MG law (solid lines). m-*V* curves of the *J*-*V* curves in figure 4c are shown in figure 4d. The linear regime and the apparent Mott-Gurney regime are shown as dashed lines (the linear equations were fitted in the low voltage regime, and the MG law was fitted at the point where m = 2). If a fit with the MG law is performed in the trap-influenced regime, as is sometimes seen in the literature, values for the effective charge-carrier mobilities between $1-3 \times 10^{-4} \text{ cm}^2/\text{Vs}$ for the symmetric devices are obtained (at room temperature). These values are similar to what has been presented in the past using SCLC[22] (a fit to the asymmetric data shown in fig. 4a gave a mobility with a comparable value of 1.1×10^{-4} cm²/Vs). The results of the ME and MG fits, as a function of temperature, are shown in fig. 6a.



Figure 5 – Numerical fits to SCLC data. a) Forward and reverse bias current of an asymmetric device, b) forward bias current of a symmetric device (the forward- and

reverse-bias currents overlap), and c) forward bias current of a symmetric device at varying temperatures. Results of the fits in a) and b) are shown in the graphs. An injection barrier of 0.11 eV and an exponential tail of trap states were required to consistently fit to the temperature series.

Figure 5a shows the fits from the drift-diffusion simulations to both the forward and reverse bias asymmetric device data. Figure 5b shows the fits to the forward bias current of the symmetric device (the forward and reverse bias currents overlapped). Since the shunt resistivity was determined from a fit with Ohm's law (figure 4a), this value could be used as input for the driftdiffusion fit along with the measured device thickness. The field-, temperature- and chargecarrier density independent hole band-like mobility, μ_h , characteristic tail energy, E_{ch} , tail state density, $n_{\rm t}$, and injection and extraction barrier heights, $q\phi_{\rm inj}$ and $q\phi_{\rm ext}$, were allowed to vary during the fitting process for both the symmetric and asymmetric device data. Figure 5c shows the fits using the drift-diffusion model to the temperature series with the inclusion of exponential tails and injection barriers as those obtained in fig. 5b (fitting results along with additional fits performed on a 115 nm and a 290 nm devices are shown in figures S3, S4 and S5. These results are consistent with the results shown in fig. 5. A deviation from the trend is observed for the 115 nm device at low temperature, which is assigned to be due to the experimental conditions). Note that the parameters obtained from this fitting vary slightly with temperature. The relative insensitivity of the parameters to temperature indicate that the underlying transport model is valid. However, a moderately good set of fits to the SCLC *J-V* data can be obtained by making the fit with the constraint that all parameters are temperature independent, as shown in fig. S6b, c and d. Whereas the values for the trap characteristics and barrier heights affect the slope of the J-V curves, since no recombination occurs and the measurement is performed under steady-state conditions, $\mu_{\rm h}$ only affects the magnitude of the current density, meaning that the *m*-V curves can be fitted prior to the *I-V* curves.

V. DISCUSSION

The V_{bi} for the symmetric device (fig. 5b) was found to be zero, as expected, with barrier heights for both injection and extraction of 0.11 eV. The V_{bi} for the asymmetric device (fig. 5a) was determined to be 1.78 V with an injection barrier height of 0.11 eV. The determined V_{bi} for the asymmetric device is therefore 0.88 V higher than the value expected from the nominal energy level offset (fig. 2b). This is likely due to oxidation of the aluminium contact forming a thin aluminium-oxide layer at the contact/semiconductor interface^[42]. An underestimation of the V_{bi} using eq. 5 was observed (fig. S7b and c), since this equation does not account for both traps and injection barrier heights.

Both the symmetric and asymmetric devices showed evidence of shallow exponential tails ($E_{ch} \cong 0.045 \text{ eV}$) obtained from the drift-diffusion simulations, with trap densities, extending from the band edges, of $4.49 \times 10^{19} \text{ cm}^{-3} \text{ eV}^{-1}$ and $7.02 \times 10^{19} \text{ cm}^{-3} \text{ eV}^{-1}$ for the asymmetric and symmetric device, respectively. The MH equation, eq. 3, gives a much higher estimate of E_{ch} ($J \propto V^{l+1}$ with $l = E_{ch}/k_{\rm B}T$)[11] of 0.070 eV if fit to the asymmetric device data shown in fig. 5a. Note that using the MH equation to the temperature dependent data would yield a value for the characteristic energy which would decrease with increased temperature (fig. 4d) since the slope decreases. A temperature dependent characteristic energy was, however, not observed when we analysed the data using the numerical model (figs. S3c, S4c and S5c).

For the devices in figs. 4a, the hole band-like mobility, μ_h , was determined to be 3.22×10^{-3} cm²V⁻¹s⁻¹ and 4.56×10^{-3} cm²V⁻¹s⁻¹, for the asymmetric and symmetric devices respectively, which is orders of magnitude higher than the effective mobility obtained from the MG evaluation (fig.

6a), and an order of magnitude higher than the effective hole mobility reported in the past from SCLC characterisation of Spiro-OMeTAD[22].



Figure 6 - a) Resulting mobility values using various techniques (red for numerical fitting, blue for the Mott-Gurney law fitting, eq. 1, and green for the Moving Electrode equation fitting, eq. 6). The set of various symbols refers to values inferred from devices with different semiconducting layer thicknesses: 290 (squares), 190 (circles) and 115 nm (triangles), respectively. b) Comparison of the band-like mobility with the effective mobility from the MG law modified by the ratio of total to free charge carriers (figure S7a and b), and with the value estimated using eq. 17. c) Comparison of the band-like mobility with the mobility with the MG mobility modified for both traps and injection limitation at the metal/semiconductor interfaces. If the MG mobility was used to fit the entire voltage range for the 190 nm device, an apparent field dependence is observed, as shown in fig. S7d.

Figure 6a compares the hole mobility values obtained from fits with the numerical model to the data (fig. 5c) to mobilities obtained by fitting eqs. 1 or 6 (fig. 4c). The obtained trap characteristics from the numerical fits are shown in the supplementary information (figs. S3, S4 and S5)[32]. Where a temperature-dependent mobility was obtained using analytical fits with eq. 1 and 6, a temperature independent band-like mobility was obtained using numerical fits with the inclusion of traps and the inclusion of injection limitation at the device interfaces. From the numerical model, the obtained charge-carrier mobility at room temperature (300 K) was found to be more than an order of magnitude higher than the effective mobility and more than four orders of magnitude higher at 200 K. While the mobility returned by the numerical model is a band-like mobility, the mobility returned by the analytical equations represents the band-like mobility weighted by the effects of trapping and poor charge injection, i.e., an effective mobility. The temperature dependence of the charge transport then originates from injection limitation along with de-trapping being a thermally activated process that becomes more difficult at lower temperatures, and not from a temperature dependent band-like mobility.

Figure 6b compares the temperature-independent band-like mobility with the mobility obtained from eq. 17 and the effective mobility extracted from fitting with the MG law (modified by the ratio of the arithmetic means of the total to free charge carriers obtained from the driftdiffusion simulations, $\mu_{MG} \cdot \langle n_{total} \rangle / \langle n_{free} \rangle$). Note that the MG law modification was only possible by simulating the densities of free and trapped holes in the device. Thus, this modification would not be available if a purely analytical approach was used. The ratio, $\langle n_{total} \rangle / \langle n_{free} \rangle$, was evaluated at the same applied voltage as the Mott-Gurney law (the voltage at which m = 2) (see fig. S8). Good agreement is found between the mobility obtained from eq. 17 and the mobility obtained from the numerical fits when the average value for the characteristic energy and trap density from the simulations, averaged over all thicknesses and temperatures, was used in eq. 17 ($n_t =$ $1.31 \cdot 10^{20}$ cm⁻³eV⁻¹ and $E_{ch} = 0.044$ eV, respectively). Similar to fitting with the MH equation, in order to use eq. 17 for determining the mobility and trap characteristics, without any prior knowledge of either the mobility or trap characteristics, a thickness and temperature series should be analysed to reach a convergence of the parameters as shown in fig. 6b. However, the much improved accuracy of determining the band-like mobility, E_{ch} and n_t using eq. 17 compared to the MH equation is shown in the SI (fig. S9). Contrary to the good agreement between eq. 17 and the numerical calculations, the overall magnitude of the effective mobility from the modified MG law is seen to only approach the band-like mobility at 300 K.

Figure 6c compares the band-like mobility with the effective mobility now corrected for both trapping and a term for injection limitation, $\mu_{MG} \cdot \langle n_{total} \rangle / \langle n_{free} \rangle \cdot \exp(q\phi_{inj}/k_BT)$. Interestingly, the temperature independence of this effective charge-carrier mobility is now comparable to that of the band-like mobility, however with the overall magnitude being an order of magnitude too high.

From the above discussion it is evident that a proper (non *ad hoc*) inclusion of traps is important to correctly extract trap characteristics, such as the application of eq. 17 or the numerical approach. Furthermore, proper inclusion of traps and injection limitation is crucial for fitting to the entire voltage range, i.e., for the correct determination of the band-like charge-carrier mobility using a numerical drift-diffusion simulator, since the probed current-voltage behaviour is affected by these effects at higher voltages, simultaneously, in a non-trivial manner.

VI. CONCLUSIONS

By allowing the charge-carrier mobility, trap characteristics, injection barrier heights and the shunt resistivity to vary, we have shown that a numerical fit can be obtained to the entire J-Vcurve measured from both symmetric and asymmetric single-carrier devices made from an organic semiconductor (Spiro-OMeTAD). The obtained charge-carrier mobilities and trap densities for both the symmetric and the asymmetric devices agree within a factor of less than two, across several device thicknesses and over a large range of temperatures. Moreover, the hole band-like mobility obtained from numerical fitting was more than an order of magnitude higher $(4.56 \times 10^{-3} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1})$ than the effective hole mobility determined using the Mott-Gurney law $(2.30 \times 10^{-4} \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1})$ at room temperature (300 K) and more than four orders of magnitude at 200 K. We further show that, whilst simple analytical equations required a temperaturedependent mobility to achieve fits to the experimental data, the use of either an analytical model that accounts for traps – such as the novel analytical model presented in this paper – or a numerical model, give rise to a temperature-independent band-like mobility whilst simultaneously yielding information about trap characteristics and injection statistics. Our analysis and results highlight the importance of: Either showing, and using a numerical model to fit to, the entire *J*-*V* curve while accounting for traps, the built-in voltage and injection limitation;

or to use a more sophisticated analytical model which can correctly account for traps, such as the model presented herein.

VII. ACKNOWLEDGEMENTS

JAR, SAH and JN would like to thank the Engineering and Physical Sciences Research Council (EPSRC grant nos. EP/K030671/1 and EP/K010298/1) and the Doctoral Training Centre for Plastic Electronics (EP/G037515) for funding. XS and JN would like to thank the Engineering and Physical Sciences Research Council (via grant nos. EP/M025020/1 and EP/K029843/1). TK acknowledges funding via an Imperial College Junior Research Fellowship.

VIII. REFERENCES

[1] H. Sirringhaus, P.J. Brown, R.H. Friend, M.M. Nielsen, K. Bechgaard, and A.J.H. Spiering, <u>Two-dimensional charge transport in self-organized, high-mobility</u> <u>conjugated polymers</u>, Nature 401, 685 (1999).

[2] H. Bässler and A. Köhler, <u>Charge transport in organic semiconductors</u>, Top. Curr. Chem. 312, 1 (2012).

[3] J. Dacuña, W. Xie, and A. Salleo, <u>Estimation of the spatial distribution of traps</u> <u>using space-charge-limited current measurements in an organic single crystal</u>, Phys. Rev. B 86, 115202 (2012).

[4] R. Noriega, J. Rivnay, K. Vandewal, F.P. V Koch, N. Stingelin, P. Smith, M.F. Toney, and A. Salleo, <u>A general relationship between disorder, aggregation and charge transport in conjugated polymers</u>, Nat. Mater. 12, 1038 (2013).

[5] N.F. Mott and R.W. Gurney, Electronic Processes in Ionic Crystals (Oxford University Press, 1940).

[6] A. Rose, <u>Space-charge-limited currents in solids</u>, Phys. Rev. 97, 1538 (1955).

[7] M.A. Lampert, <u>Simplified theory of space-charge-limited currents in an</u> insulator with traps, Phys. Rev. 103, 1648 (1956).

[8] P. Mark and W. Helfrich, <u>Space-charge-limited currents in organic crystals</u>, J. Appl. Phys. 33, 205 (1962).

[9] P.H. Nguyen, S. Scheinert, S. Berleb, W. Brutting, and G. Paasch, <u>The influence</u> of deep traps on transient current–voltage characteristics of organic lightemitting diodes, Org. Electron. 2, 105 (2001).

[10] P. De Bruyn, A.H.P. Van Rest, G.A.H. Wetzelaer, D.M. De Leeuw, and P.W.M. Blom, <u>Diffusion-Limited Current in Organic Metal-Insulator-Metal Diodes</u>, Phys. Rev. Lett. 111, 1 (2013).

[11] J. Fischer, W. Tress, H. Kleemann, J. Widmer, K. Leo, and M. Riede,

Exploiting diffusion currents at Ohmic contacts for trap characterization in organic semiconductors, Org. Electron. 15, 2428 (2014).

[12] H.T. Nicolai, M.M. Mandoc, and P.W.M. Blom, Electron traps in

semiconducting polymers: Exponential versus Gaussian trap distribution, Phys. Rev. B 83, 195204 (2011).

[13] J. Dacuña and A. Salleo, <u>Modeling space-charge-limited currents in organic</u> <u>semiconductors: Extracting trap density and mobility</u>, Phys. Rev. B 84, 195209 (2011). [14] H.T. Nicolai, M. Kuik, G. a H. Wetzelaer, B. de Boer, C. Campbell, C. Risko, J.L. Brédas, and P.W.M. Blom, <u>Unification of trap-limited electron transport in</u> <u>semiconducting polymers</u>, Nat. Mater. 11, 882 (2012).

[15] F. Liu, A. Massé, P. Friederich, F. Symalla, R. Nitsche, W. Wenzel, and P.A. Bobbert, <u>*Ab initio* modeling of steady-state and time-dependent charge</u> <u>transport in hole-only α-NPD devices</u>, Appl. Phys. Lett. 109, 243301(1) (2016).
[16] F. Liu, H. Van Eersel, B. Xu, J.G.E. Wilbers, M.P. De Jong, W.G. Van Der Wiel,

P.A. Bobbert, and R. Coehoorn, <u>Effect of Coulomb correlation on charge</u> <u>transport in disordered organic semiconductors</u>, Phys. Rev. B 96, 205203(1) (2017).

[17] W.F. Pasveer, J. Cottaar, C. Tanase, R. Coehoorn, P.A. Bobbert, P.W.M. Blom, M. De Leeuw, and M.A.J. Michels, <u>Unified description of charge-carrier mobilities</u> <u>in disordered semiconducting polymers</u>, Phys. Rev. Lett. 94, 206601(4) (2005).

[18] J.C. Blakesley, H.S. Clubb, and N.C. Greenham, <u>Temperature-dependent</u> <u>electron and hole transport in disordered semiconducting polymers: Analysis of</u> <u>energetic disorder</u>, Phys. Rev. B 81, 45210 (2010).

[19] J.M. Montero and J. Bisquert, <u>Trap origin of field-dependent mobility of the</u>

carrier transport in organic layers, Solid. State. Electron. 55, 1 (2011). [20] R.C.I. MacKenzie, T. Kirchartz, G.F.A. Dibb, and J. Nelson, <u>Modeling</u> <u>nongeminate recombination in P3HT: PCBM solar cells</u>, J. Phys. Chem. C 115, 9806 (2011).

[21] J.M. Frost, J. Kirkpatrick, J. Nelson, T. Kirchartz, and J. Nelson, <u>Parameter</u> <u>free calculation of the subgap density of states in poly (3-hexylthiophene)</u>, Faraday Discuss. 174, 1 (2014).

[22] D. Poplavskyy and J. Nelson, <u>Nondispersive hole transport in amorphous</u> <u>films of methoxy-spirofluorene-arylamine organic compound</u>, J. Appl. Phys. 93, 341 (2003).

[23] R.W. Smith and A. Rose, <u>Space-charge-limited currents in single crystals of cadmium sulfide</u>, Phys. Rev. 97, 1531 (1955).

[24] T. Kirchartz, <u>Influence of diffusion on space-charge-limited current</u> <u>measurements in organic semiconductors</u>, Beilstein J. Nanotechnol. 4, 180 (2013).

[25] A.M. Nardes, M. Kemerink, M.M. De Kok, E. Vinken, K. Maturova, and R.A.J. Janssen, Org. Electron. 9, 727 (2008).

[26] Y. Kim, A. Ballantyne, J. Nelson, and D.D.C. Bradley, <u>Conductivity, work</u> <u>function, and environmental stability of PEDOT: PSS thin films treated with</u> <u>sorbitol</u>, Org. Electron. 10, 205 (2009).

[27] M. Kröger, S. Hamwi, J.M. Philips, T. Riedl, and W. Kowalsky, <u>Role of the</u> <u>deep-lying electronic states of in the enhancement of hole-injection in organic</u> <u>thin films</u>, Appl. Phys. Lett. 95, 123301 (2009).

[28] M.T. Greiner, L. Chai, M.G. Helander, W. Tang, and Z. Lu, <u>Transition metal</u> oxide work functions: the influence of cation oxidation state and oxygen vacancies, Adv. Funct. Mater. 22, 4557 (2012).

[29] P.N. Murgatroyd, <u>Theory of space-charge-limited current enhanced by</u> <u>Frenkel effect</u>, J. Phys. D Appl. Phys. 3, 151 (1970). [30] J.C. Blakesley, F. a. Castro, W. Kylberg, G.F. a Dibb, C. Arantes, R. Valaski, M. Cremona, J.S. Kim, and J.S. Kim, <u>Towards reliable charge-mobility benchmark</u> <u>measurements for organic semiconductors</u>, Org. Electron. 15, 1263 (2014).
[31] Z. He, C. Zhong, X. Huang, W.-Y. Wong, H. Wu, L. Chen, S. Su, and Y. Cao, <u>Simultaneous enhancement of open-circuit voltage, short-circuit current</u>

<u>density, and fill factor in polymer solar cells</u>, Adv. Mater. 23, 4636 (2011). [32] supplementary information

[33] A.A. Grinberg and S. Luryi, <u>Space-charge-limited current and capacitance in</u> <u>double-junction diodes</u>, J. Appl. Phys. 61, 1181 (1987).

[34] G. a. H. Wetzelaer and P.W.M. Blom, <u>Ohmic current in organic metal-insulator-metal diodes revisited</u>, Phys. Rev. B 89, 241201 (2014).

[35] J.A. Röhr, T. Kirchartz, and J. Nelson, <u>On the correct interpretation of the</u> <u>low voltage regime in intrinsic single-carrier devices</u>, J. Phys. Condens. Matter 29, 205901(9) (2017).

[36] P.E. Schmidt and H.E. Henisch, <u>Drift-diffusion theory of symmetrical</u> <u>double-junction diodes</u>, Solid-State-Electronics 25, 1129 (1982).

[37] J.G. Simmons, <u>Richardson-Schottky effect in solids</u>, Phys. Rev. Lett. 15, 967 (1965).

[38] R. de Levie, N.G. Seidah, and H. Moreia, <u>Transport of ions of one kind</u> through thin membranes, J. Membr. Biol. 10, 171 (1972).

[39] J.G. Simmons, <u>Theory of metallic contacts on high resistivity solids—I.</u> <u>Shallow traps</u>, J. Phys. Chem. Solids 32, 1987 (1971).

[40] S.L.M. van Mensfoort and R. Coehoorn, <u>Effect of Gaussian disorder on the</u> voltage dependence of the current density in sandwich-type devices based on <u>organic semiconductors</u>, Phys. Rev. B 78, 085207(16) (2008).

[41] T. Kirchartz, B.E. Pieters, J. Kirkpatrick, U. Rau, and J. Nelson,

Recombination via tail states in polythiophene: fullerene solar cells, Phys. Rev. B 83, 115209 (2011).

[42] R. Michel, J. Gastaldi, C. Allasia, and C. Jourdan, <u>Initial interaction of oxygen</u> <u>with aluminium single crystal faces: a LEED, AES and work function study</u>, Surf. Sci. 95, 309 (1980).