# **Breakdown of phonon band theory in MgO**

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We present a series of detailed images of the distribution of kinetic energy among frequencies and wave vectors in the bulk of an MgO crystal as it is heated slowly until it melts. These spectra, which are Fourier transforms of mass-weighted velocity-velocity correlation functions calculated from accurate molecular dynamics (MD) simulations, provide a valuable perspective on the growth of thermal disorder in ionic crystals. We use them to explain why the most striking and rapidly progressing departures from a band structure occur among longitudinal optical (LO) modes, which would be the least active modes at low temperature (*T* ) if phonons did not interact. The degradation of the LO band begins, at low *T* , as an anomalously large broadening of modes near the center of the Brillouin zone (BZ), which gradually spreads towards the BZ boundary. The LO band all but vanishes before the crystal melts, and transverse optical (TO) modes' spectral peaks become so broad that the TO branches no longer appear band-like. Acoustic bands remain relatively well defined until melting of the crystal manifests in the spectra as their sudden disappearance. We argue that, even at high  $T$ , the long wavelength acoustic (LWA) phonons of an ionic crystal can remain partially immune to disorder generated by its LO phonons; whereas, even at low *T*, its LO phonons can be strongly affected by LWA phonons. This is because LO displacements average out in much less than the period of an LWA phonon; whereas during each period of an LO phonon, an LWA phonon appears as a quasistatic perturbation of the crystal, which warps the LO mode's intrinsic electric field. LO phonons are highly sensitive to acoustic warping of their intrinsic fields because their frequencies depend strongly on them: They cause the large frequency difference between LO and TO bands known as *LO-TO splitting*. We calculate vibrational spectra from MD trajectories using a method that we show to be classically exact and therefore applicable, with equal validity, to any solid or liquid in any thermal or nonthermal state. By demonstrating its power and generality, we show that it has become possible to go far beyond the reach of perturbation theories and mean-field theories in the study of vibrations in materials.

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# **I. INTRODUCTION**

Band theories are theories of elementary excitations in crystals that are derived under the simplifying assumption that the elementary excitations are approximately independent of one another [\[1–4\]](#page-21-0). For example, the band theory of electrons assumes that excited electronic states of a crystal are composed of weakly interacting *quasiparticles* (QPs) [\[5\]](#page-21-0), which are collective excitations of the electrons in which only one of the electrons plays a prominent role. A QP can roughly be described as a single electron whose electric field is screened, and whose energy and inertia are changed, by its many weak interactions with other electrons. When interactions between QPs are strong, they are not mutually independent and each one is not really a QP, but part of a more complex collective excitation in which multiple electrons play prominent roles [\[5\]](#page-21-0). The assumptions underpinning band theory are not valid for such excitations, and when they cause band theory to break

down as an approximation, the electrons are said to be *strongly correlated*.

Similarly, the band theory of phonons assumes that the vibrational energy of a crystal can be approximated well by a sum of energies of individual *phonons*, or *phonon quasiparticles*, which are lattice waves with well defined frequencies and wave vectors. Phonon band theory breaks down when strong interactions between phonons with very different frequencies and/or wave vectors strongly correlate their motions, resulting in motion that cannot be characterized by a single wave vector and a single frequency, or even by a narrow range of wave vectors and a narrow range of frequencies.

Although strong electronic correlation has been the subject of intense study for decades, there have been few fundamental direct studies of strong phononic correlation [\[6–10\]](#page-21-0). One reason for this is the so-called *terahertz gap*, which is the region of the electromagnetic spectrum between about 100 GHz and ∼10 THz where detectors, and intense continuous wave or pulsed sources, are either unavailable or not widely available [\[11,12\]](#page-21-0). Another reason is that strong phononic correlation is difficult to observe experimentally because measured spectra tend to have low resolutions, making it difficult to detect when, or to what degree, the band theory of weakly interacting phonons has broken down.

In this work, we use atomistic simulations to study how the vibrational spectrum of MgO changes as its temperature  $(T)$ 

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<span id="page-1-0"></span>

FIG. 1. Distribution of kinetic energy of MgO in reciprocal spacetime at a selection of temperatures (*T* ) up to and above the melting temperature *Tm*. The values of *T* are approximate; precise values are provided in Sec. [IV B.](#page-9-0) The blue and white background image is data from molecular dynamics simulations using an *ab initio*-parameterized force field  $U_{FF}$ ; at each *T*, the kinetic energy density in reciprocal spacetime  $(\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega))$  was calculated and normalized to one by dividing it by its integral over the first Brillouin zone. This normalization allowed the same map between colors and energy density values to be used at each *T* . The image is pixelated, with the width and height of each pixel equal to the resolutions imposed on our simulations by our finite simulation cell size and simulation times, respectively. We also project the full spectrum onto the normal mode eigenvectors calculated at zero temperature; and we plot the locations of the *quasiparticle* peaks as orange symbols in the lower right panel. The red and green symbols on each plot are the phonon frequencies calculated with  $U_{\text{FF}}$  at zero temperature and using the quasiharmonic approximation, respectively. The frequencies measured by Sangster experimentally [\[13\]](#page-21-0) are plotted with blue squares and the black lines were calculated by Karki *et al.* using density functional perturbation theory (DFPT) and the quasiharmonic approximation [\[14\]](#page-21-0). The side plots at each temperature are plots of  $\sum_{\mathbf{k}} \hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega)$  vs  $\omega$ , with  $\omega$  on the vertical axis.

increases from  $T = 100$  K, where band theory is accurate, to  $T = 3800$  K, where the crystal has melted and the spectrum has lost every semblance of a band structure. At each *T* , we calculate the distribution,  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$ , of kinetic energy in  $(k, \omega)$  *space*, or *reciprocal spacetime*, which are the terms we use to refer to the space of all wave vectors and frequencies. We present these spectra in Fig. 1, and analyze them and discuss them in detail in Sec. [V.](#page-10-0)

Just as perturbation theories and mean-field approximations (e.g., Hartree-Fock) cannot accurately describe strongly correlated electrons, mean-field approximations for phonons (so-called *self-consistent phonon approximations* [\[1,2,15,16\]](#page-21-0)) cannot accurately describe strongly correlated phonons. Therefore mean-field based methods of calculating vibrational spectra [\[17–](#page-21-0)[21\]](#page-22-0) fail when phononic correlation is sufficiently strong, such as at high temperatures (*T* ), or in a liquid. However, the method we use to calculate spectra from our MD simulations is classically exact, which means that its accuracy is unaffected by the strength of phononic correlation.

Each distribution  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  is the Fourier transform (FT), with respect to both space and time, of a mass-weighted velocity-velocity correlation function (VVCF) calculated from a molecular dynamics (MD) simulation at a different *T* . The MD simulations were performed with a polarizable-ion force field whose parameters have been fit closely to the density functional theory (DFT) potential energy surface [\[22,23\]](#page-22-0).

It is common to Fourier transform velocity *autocorrelation* functions (VACFs) with respect to time to produce frequencyresolved spectra, such as those along the right-hand vertical edge of each panel in Fig. 1. However, wave-vector-resolved spectra are relatively rare: They were calculated for a one dimensional material in Ref. [\[6\]](#page-21-0), and more recently they have been calculated for three dimensional crystals using *ab initio* MD [\[7,9](#page-21-0)[,24\]](#page-22-0). However, the computational expense of *ab initio* MD severely restricts the number of frequencies and wave vectors at which  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  can be calculated. Therefore, so far, the resolutions of the spectra calculated by *ab initio* MD have been too low to see bands.

Force fields whose mathematical forms can mimic the electronic response to nuclear motion, and whose parameters are fit to enormous datasets calculated *ab initio*, provide a very useful balance of speed and accuracy [\[22,25–32\]](#page-22-0). As Fig. [1](#page-1-0) illustrates, they allow accurate spectra to be calculated with resolutions that are high enough to see a crystal's band structure. For example, Lahnsteiner and Bokdam [\[33\]](#page-22-0) recently used them to calculate detailed spectra at two temperatures in order to extract phonon QP frequencies for use within band theory. Our purpose is very different.

The primary purpose of this work is to study strong phononic correlation. To this end, we have undertaken a systematic study of the strengthening of phononic correlation, and the consequent breakdown of band theory, as a crystal is heated.

We chose MgO for our study because it is a material whose vibrational properties are important in many contexts, from studies of seismic waves traveling through the Earth's lower mantle, to its use as a thermal or electrical insulator, as a substrate for growing superconducting or ferroelectric perovskites, and in countless other important applications [\[8,14](#page-21-0)[,23,34–36\]](#page-22-0). As well as being a technologically important material, MgO is one of the simplest oxides: It is a strongly ionic insulator with the same cubic crystal structure as rocksalt. For these reasons, it plays a similar representative role for oxides to that played by silicon for semiconductors: It is often the simplest setting in which properties or phenomena that are common to many oxides can be investigated. Several experimental and computational studies of phonon-phonon interactions in MgO and similar materials have recently been published [\[8,10](#page-21-0)[,37–39\]](#page-22-0); therefore it is a natural and obvious starting point for investigating strong phononic correlation in oxides.

Calculations of detailed accurate spectra like those presented in Fig. [1](#page-1-0) and by Lahnsteiner and Bokdam have been possible for a decade or more. One reason for their rarity may be that it is not commonly known that, within classical physics, the FT of the VVCF,  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega)$ , is *exactly* the distribution of kinetic energy in reciprocal spacetime: To our knowledge, all existing derivations and discussions of the theory rely on two simplifying assumptions: They assume that the crystal is at thermal equilibrium and that  $T$  is low enough for the vibrational spectrum to be a band structure [\[40,41\]](#page-22-0). After making these assumptions, the *equipartition theorem* is usually invoked to relate  $\tilde{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  to the vibrational density of states (VDOS) [\[40,41\]](#page-22-0).

Lahnsteiner and Bokdam are among those who state that  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  (our notation) is the wave-vector-resolved VDOS, and they support this statement by citing Refs. [\[7](#page-21-0)[,24,42,43\]](#page-22-0). This interpretation of  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$ , which can also be found in many other works [\[40,41,44–47\]](#page-22-0), is correct in the limit  $T \rightarrow 0$  and under the simplifying assumptions that lead to

band theory. It is also the appropriate interpretation of existing theory, because derivations such as those of Dove [\[40\]](#page-22-0) and Lee *et al.* [\[41\]](#page-22-0) only provide a clear physical interpretation of  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  at thermal equilibrium in the  $T\to 0$ limit.

However, it is shown in [A](#page-20-0)ppendix  $\overline{A}$  and Ref. [\[48\]](#page-22-0) that Fourier transforming the VVCF is a much more powerful and general method than it is currently believed to be: It is shown that  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  is exactly the distribution of kinetic energy in reciprocal spacetime. Appendix [A](#page-20-0) contains an illustrative proof of this result, for the case of a vibrating string. Therefore each of the spectra presented in Fig. [1](#page-1-0) is *exactly* the distribution, among points  $(k, \omega)$  in reciprocal spacetime, of the classical kinetic energy of the MD simulation from which it was calculated.

Furthermore, proving that  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  is the kinetic energy distribution does not require any assumptions to be made about the statistical state or the structure of the material. It is a result that applies, with equal theoretical validity, to any crystalline or amorphous material, in any stationary or nonstationary state, regardless of the strength of the correlation between different vibrations and waves. This means that its range of possible applications is vast.

For example, it could be used to calculate the spectra of solids or liquids while they are being resonantly excited by THz radiation; or to study how spectra change during order-disorder phase transitions; or to investigate the relationships between structure, energetics, and diffusion in liquids. Using it may deepen our understanding of strongly correlated electronic systems, which are much harder to study computationally, by providing insight into aspects of strong correlation that are common to both phononic and electronic systems. It can also be used to assess the accuracies of approximate methods, such those based on perturbation theory, self-consistent mean-field methods, and methods that assume thermal equilibrium. As it is the only method that provides the exact classical spectrum, it is the method against which the accuracies of all other methods should be judged. A secondary purpose of this work is to demonstrate the power of this theoretical result and computational method.

By applying it to MgO we uncover a simple, strong, and general nonresonant mechanism by which the highest frequency optical modes of an ionic crystal can be disrupted by the lowest frequency acoustic modes, leading to the disintegration, or *melting*, of the optical bands. We call this mechanism *acoustic warping of optical phonon fields*. We also explain why, despite this mechanism causing longitudinal optical (LO) bands to melt, the acoustic bands responsible for their melting remain intact: It is because acoustic bands are *adiabatically decoupled* from LO phonons in the same way that heavy nuclei are adiabatically decoupled from electrons, despite pushing the electron density around as they move. Among many other analyses of the results presented in Fig. [1,](#page-1-0) we compare them to a second set of spectra (Fig. [3\)](#page-14-0) which were calculated from MD simulations performed at the  $T \rightarrow 0$  density. This allows us to discover which *T*-induced changes to the spectrum can be explained by thermal expansion and which cannot. It also provides insight into the strengths and limitations of the *quasiharmonic approximation* [\[14](#page-21-0)[,40,49\]](#page-22-0).

<span id="page-3-0"></span>In the next section, we discuss the effects of  $T$  on phonon band structures in general terms. In Sec. [III,](#page-5-0) we explain our notation and some aspects of phonon theory that can be different when phonons are strongly correlated than they are at low *T* where perturbation theories are applicable. For example, at high *T* each mode is not Lorentzian, in general, and so the Lorentzian width is not a good measure of the degree to which *T* has broadened it. In Sec. [IV,](#page-8-0) we explain how we performed our simulations.

We begin Sec. [V](#page-10-0) by discussing the most important limitations of our simulations. Then we begin discussing and numerically analysing the spectra presented in Figs. [1](#page-1-0) and [3.](#page-14-0) We focus our discussions and analyses on the most important and striking features of these spectra, and we include an explanation of the acoustic warping mechanism. We summarize our conclusions in Sec. [VI.](#page-19-0)

# **II. QUALITATIVE EFFECTS OF TEMPERATURE ON BAND STRUCTURES**

In this section, we discuss the qualitative effects of temperature on vibrational spectra. We begin with an illustration of a pertinent mathematical point.

Figure  $2(a)$  is a plot of two curves, each of which is the sum of ten randomly positioned Gaussians. The only difference between the two curves is that the width of the Gaussians contributing to the red curve is a factor of ten larger than those contributing to the blue curve. The convolution of the red curve with itself and the convolution of the blue curve with itself are plotted in Fig.  $2(b)$ , and the Fourier transforms of these convolutions are plotted in Fig. 2(c).

It is well known that the more localized a smooth function is, the more delocalized its Fourier transform is [\[50\]](#page-22-0). These plots illustrate that the Fourier transform of the convolution of a smooth function with itself is more delocalized when the function is localized and vice-versa.

 $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  is the FT of the VVCF and the VVCF is the correlation of the spacetime distribution of  $\sqrt{\text{mass}}$ -weighted atomic velocities with itself. The spacetime distribution of the atoms' kinetic energy is more localized/delocalized when the spacetime distribution of their velocities are more localized/delocalized. Therefore Fig. 2 illustrates the fact that as kinetic energy becomes more localized in real spacetime it becomes more delocalized in reciprocal spacetime and vice versa. With this in mind, we now discuss the qualitative effects of temperature on vibrations in crystals.

## **A. Normal mode vibrations, phonon quasiparticles, and strong phononic correlation**

Classically, and in the  $T \rightarrow 0$  limit, the term *phonon* refers to an oscillation of one of a crystal's normal modes of vibration, which are standing waves of the lattice that all of the crystal's atoms participate in. Each normal mode is characterized by a single frequency  $(\omega)$  and a single wave vector  $(\mathbf{k})$ [\[1,3](#page-21-0)[,48\]](#page-22-0). Therefore the energy of each normal mode vibration (NMV) can be regarded as localized at a point  $(\mathbf{k}, \omega)$  in reciprocal spacetime.



FIG. 2. (a) Each of the plotted curves is the sum of an array of Gaussians at the same randomly chosen positions. The only difference between the curves is that the widths of the Gaussians contributing to red curve are ten times larger than those contributing to the blue curve. (b) The convolutions of each curve plotted in (a) with itself. (c) The Fourier transforms of the convolutions plotted in (b).

The amplitudes of NMVs vanish in the  $T \to 0$  limit, making them perfectly harmonic and mutually noninteracting. Therefore the energy of each one is delocalized in spacetime. It is delocalized in space because every atom in the crystal participates in it and shares its energy. It is "delocalized" in time in the sense that it is not transient: NMVs last forever in the  $T \to 0$  limit because, without interactions, they cannot dissipate their energies; and they cannot disperse because each one is characterized by a single frequency and a single wave vector.

As well as being the limit in which kinetic energy is delocalized in spacetime and localized at a point in reciprocal spacetime, the  $T \rightarrow 0$  limit is the limit in which atoms are strongly correlated and the limit in which phonons are uncorrelated. Each contribution to the atoms' kinetic energy is a collective motion of all atoms, whereas each phonon possesses its own kinetic energy, which is independent of the energies of other phonons.

At finite *T*, phonons' amplitudes are finite, and they interact to some degree. When they interact, their motions become correlated and each one is no longer characterized by a single frequency and wave vector: it has become a superposition of NMVs with different frequencies and wave vectors.

This interaction-induced mixing of frequencies and wave vectors is analogous to what happens when a pair of harmonic oscillators with frequencies  $\omega_1$  and  $\omega_2$  are coupled. The motion of each one becomes characterized by both frequencies; or equivalently, each one becomes an oscillation at frequency  $\frac{1}{2}(\omega_1 + \omega_2)$  whose amplitude is modulated by an oscillation at frequency  $\frac{1}{2}|\omega_1 - \omega_2|$ . When many oscillators are coupled strongly, the motion of each one becomes a superposition of oscillations at many different frequencies. Similarly, interactions cause each phonon's energy to be distributed among the frequencies and wave vectors of all lattice waves contributing to its motion.

## *1. Phonon quasiparticles*

At low finite *T* , each phonon is still mostly composed of a single NMV, but it contains small contributions to its motion from the other NMVs with which it interacts [\[1\]](#page-21-0). As *T* increases and interactions strengthen the fraction of its kinetic energy contributed by the original NMV decreases and the fractions contributed by others increase. Therefore, as *T* increases from the  $T \rightarrow 0$  limit, each NMV's kinetic energy spreads out from the point  $(k, \omega)$  at which it was localized and becomes distributed among the frequencies and wave vectors of all of the NMVs with which it interacts.

At low *T* the vibrational spectrum remains sharply peaked near the frequencies and wave vectors of the normal modes, but the peaks have finite widths. This means that each phonon contributing to each peak is a QP, i.e., a superposition of NMVs with very similar wave vectors and frequencies. Phonon QPs can dissipate their energies and disperse: their average lifetime is inversely proportional to the width of their spectral peak  $[1,51]$  $[1,51]$ . Therefore, at low *T*, they are still reasonably long-lived excitations with reasonably well-defined frequencies and wave vectors, and one can think of each one as an NMV that is dressed by its interactions with other NMVs.

Interactions make the average frequency and wave vector of the QPs *T* -dependent, and cause their spatial coherence (size) and temporal coherence (lifetime) to reduce as *T* increases. The reduction in their sizes means that they are no longer collective motions of all atoms in the crystal. Therefore they are no longer standing waves, but traveling wave packets.

The kinetic energy spectrum,  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega)$ , is called a *band structure* at low *T* because the set of points at which there are spectral peaks forms a set of three dimensional surfaces, or *bands*, in four-dimensional reciprocal spacetime. As each phonon becomes a mix of NMVs, these spectral peaks broaden and move, which causes the bands to blur, lose definition, and shift in frequency.

In quantum mechanics, the energies of NMVs and QPs are quantized and it is the quanta that are known as phonons. However, because most of our simulations and analyses are classical, we use the term phonon to refer to NMVs, in the  $T \rightarrow 0$  limit, and to phonon QPs, at finite *T*. A classical phonon is simply a vibration of the crystal whose wave vector and frequency are reasonably well defined, meaning that the distribution of its energy in reciprocal spacetime is sharply peaked, and most of its energy is localized in a small neighborhood of its peak.

#### *2. Strong phononic correlation*

As *T* increases further, sooner or later the *T* -induced changes to the spectrum become more complex than a gradual shifting and symmetric broadening of QP peaks. Peaks may broaden so much that they vanish; one peak may become two peaks at very different frequencies and wave vectors; or the spectrum may deviate from a superposition of well-defined QP peaks in other ways, depending on the natures and strengths of the interactions. When the spectrum can no longer be approximated by a sum of reasonably localized QP peaks, the quasiparticle approximation has broken down and the system can be regarded as strongly correlated.

When interactions between phonons are strong enough that the QP approximation has broken down, band theory also breaks down, because kinetic energy is so delocalized in reciprocal spacetime that it no longer forms bands.

For example, the energy of each oscillation in a liquid is localized in spacetime and delocalized in reciprocal spacetime. Oscillations are localized in space because spatial correlations are very short-ranged, which implies that only a small cluster of atoms participates in each one. They are localized in time because their strong coupling to other oscillatory and translational motions makes them transient: they have short lifetimes because they quickly disperse and/or dissipate their energies. Their short lifetimes can also be viewed as them morphing into other forms of motion, namely, diffusive motion or oscillations of different frequencies. For example, we can view the short lifetime of an oscillation of an atom as a consequence of the potential well in which it oscillates changing shape as neighboring atoms move.

The eventual failure of band theory as *T* increases is inevitable for every material because temperature generates disorder and because each point on a band represents a phonon with a different characteristic pattern of atomic displacements, or *eigenvector*, and with periodicities  $\lambda = 2\pi/|\mathbf{k}|$ and  $\tau = 2\pi/\omega$  in space and time, respectively. When such a vibration exists, it causes correlation between the velocity of an atom at time *t* and the velocity of another atom, which is displaced from it by  $\lambda$  in the direction of **k**, at time  $t + \tau$ . Therefore, when there is a *T* -induced reduction of the correlation length to less than  $\lambda$ , or of the correlation time to less than  $\tau$ , the spectral intensity at point  $(k, \omega)$  almost vanishes. When a particular mode or band loses all or most of its intensity in this way, we say that it has melted.

Band melting occurs gradually at most temperatures, and, as Fig. [1](#page-1-0) illustrates, it occurs at different rates for different bands, because correlation lengths and times are different for motions along different eigenvectors, in general. The rate at which each band melts is determined by the natures and strengths of the interactions between the band's phonons and other phonons. However, as Figs.  $1(g)$  and  $1(h)$  illustrate, when an acoustic band melts suddenly and completely, it means that the crystal has become structurally unstable and has undergone a phase transition. In some crystals, such as MgO, this does not occur until it becomes a liquid at the melting temperature of the crystal,  $T_m$ . In others, such as ferro-electric BaTiO<sub>3</sub> [\[52–55\]](#page-22-0), there are also one or more transitions between crystalline phases at *T* 's lower than *Tm*.

# **III. ELEMENTS OF THE THEORY OF INTERACTING PHONONS**

<span id="page-5-0"></span>In this section, we present some elements of phonon theory that we will use in this work. A more complete account of this theory can be found in Ref. [\[48\]](#page-22-0) and elsewhere in the literature  $[1-3]$ .

## **A. Notation and definitions of key quantities**

# *1. Structure of the crystal*

We use *N* to denote the number of atoms in each primitive cell, which is two in this case, and we denote the number of primitive cells in the crystal's bulk by  $N_c$ . We denote the primitive lattice vectors by  $\{a_1, a_2, a_3\}$  and we identify primitive cells by their positions, **R**, relative to an origin in the bulk of the crystal. These positions are lattice vectors, i.e.,  $\mathbf{R} = \sum_{\alpha=1}^{3} R^{\alpha} \mathbf{a}_{\alpha}$ , where  $R^1, R^2, R^3 \in \mathbb{Z}$ .

We use the compound index  $\mathbf{R}$ *j* to identify the *j*th atom in primitive cell **R** and we use  $\mathbf{R}$  *j* $\alpha$  to identify its  $\alpha$ th lattice coordinate. We denote its displacement, at time *t*, from its  $T \rightarrow 0$  equilibrium position as  $\mathbf{u}^{\mathbf{R}}(t) \equiv \sum_{\alpha} u^{\mathbf{R}}j\alpha(t) \mathbf{a}_{\alpha}$ . However, it is not very convenient to use *N* vectors  $\mathbf{u}^{\mathbf{R}}$   $\in \mathbb{R}^3$  to specify the internal structure of each primitive cell. Instead, we specify it with a single vector  $|\psi_{\mathbf{R}}\rangle = \sum_{j\alpha} \psi^{\mathbf{R}j\alpha}|j\alpha\rangle \in \mathbb{R}^{3N}$ , where  $\psi^{Rj\alpha} \equiv \sqrt{m_j}u^{Rj\alpha}$ ,  $m_j$  is the mass of the *j*th atom, and the set  $\{|j\alpha\rangle : \alpha \in \{1, 2, 3\}, j \in \{1, ..., N\}, \langle j\alpha | k\beta \rangle = \delta_{ik}\delta_{\alpha\beta}\}\$ is a complete orthonormal basis of  $\mathbb{R}^{3N}$ .

For simplicity, and despite its  $\sqrt{mass}$  weighting, we will often refer to the vector  $|\psi_{\bf R}(t)\rangle$  and its time derivative  $|\psi_{\bf R}(t)\rangle$  as the *displacement* and *velocity*, respectively, of primitive cell **R**. The kinetic energy of cell **R** is  $\frac{1}{2}\langle \dot{\psi}_{\mathbf{R}}|\dot{\psi}_{\mathbf{R}}\rangle = \frac{1}{2}\sum_j m_j |\dot{\mathbf{u}}^{\mathbf{R}}^j|^2.$ 

## *2. Correlation functions and their Fourier transforms*

The mass-weighted velocity-velocity correlation function (VVCF) is

$$
C(\mathbf{R},t) \equiv \langle \langle \dot{\psi}_{\mathbf{R}_0}(t_0) | \dot{\psi}_{\mathbf{R}_0+\mathbf{R}}(t_0+t) \rangle \rangle_{\mathbf{R}_0,t_0},\tag{1}
$$

where the average is performed over bulk cells  $\mathbf{R}_0$  and over times  $t_0$ . It is shown in Ref.  $[48]$  that the average kinetic energy per bulk unit cell is given by

$$
\frac{\langle K \rangle}{N_c} \equiv \sum_{\mathbf{k}} \sum_{\omega > 0} \hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega), \tag{2}
$$

where K is the total kinetic energy of all bulk cells;  $\langle K \rangle$  is its time average; and  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  is the discrete Fourier transform of  $C(\mathbf{R}, t)$  with respect to both **R** and  $t \, \hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega)$  is the distribution, in reciprocal spacetime, of the time-average of the crystal's kinetic energy per bulk primitive cell. As discussed in Sec. [I,](#page-0-0) Eq. (2) is an exact expression, which is proved for the case of a vibrating string in Appendix [A,](#page-20-0) and for a crystal in Ref. [\[48\]](#page-22-0). Theoretically, it is no less valid to apply it to a nonequilibrium liquid than it is to apply it to a low temperature crystal.

The first sum in Eq. (2) is over the set  $\hat{\Omega}$  of all wave vectors **k** within the first Brillouin zone,  $\Omega$ , that are compatible with the boundary conditions of the crystal. Within our treatment of the theory, wave vectors that differ by a reciprocal lattice vector, **G**, are regarded as equivalent to the same wave vector in the first Brillouin zone,  $\Omega$ . If we wanted to distinguish between them, the right-hand side of Eq. (2) would have the form  $\sum_{\mathbf{k}} \sum_{\omega > 0} (\sum_{\mathbf{G}} F(\mathbf{k} + \mathbf{G}, \omega))$ , where *F* is a function whose domain is the set of all wave vectors. However, we are choosing to define  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega) \equiv \sum_{\mathbf{G}} F(\mathbf{k} + \mathbf{G}, \omega)$  and to restrict our attention to the finite set of wave vectors  $\hat{\Omega}$ . This means that we do not explicitly distinguish between so-called *Umklapp* phonon interactions and normal interactions. This distinction is commonly made when treating interactions as scattering events, but in this work we emphasize the wave natures of phonons. Both in our classical simulations and, when *T* is high enough, in a real crystal, phonons exchange energy with one another quasicontinuously.

The second sum in Eq.  $(2)$  is over all possible frequencies. This set is countable because if the total time for which a crystal is observed or simulated is  $\mathcal{T}$ , complete oscillations whose periods are longer than  $T$  are not observed or simulated. Therefore oscillations with frequencies smaller than  $2\pi/\mathcal{T}$  cannot contribute to  $C(\mathbf{R}, t)$ . Furthermore, any two frequencies that do contribute to *C*(**R**,*t*) are only *observably* different if they differ by more than  $2\pi/\mathcal{T}$  [\[48\]](#page-22-0). Therefore  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  is only defined for  $\omega = 2\pi m/\mathcal{T}$ , where *m* is a nonnegative integer.

In a crystal, the  $T \to 0$  limit of  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega)$  is a band structure. Therefore, when studying the breakdown of band theory, it is useful to decompose it into contributions from motions along each of the crystal's normal mode eigenvectors. In Ref.  $[48]$  and many textbooks  $[1-3]$ , it is shown that, deep within the bulk of a large crystal, each normal mode is associated with a particular wave vector,  $\mathbf{k} \in \hat{\Omega}$ , and can be labeled by  $\mathbf{k}\mu$ , where  $\mu \in \{1, \dots, 3N\}$  is the *band index*. At each wave vector **k** the dynamical matrix has 3*N* real eigenvectors,  $|\varepsilon_{\mathbf{k}\mu}\rangle \in \mathbb{R}^{3N}$ , where  $\mu \in \{1, ..., 3N\}$  is a *band index*. We choose them to have unit norms and refer to them as the *cell eigenvectors*. At each **k**, the set of cell eigenvectors,  $\{\ket{\varepsilon_{\mathbf{k}\mu}}\}_{\mu=1}^{3N}$ , is orthonormal  $(\langle \varepsilon_{\mathbf{k}\mu} | \varepsilon_{\mathbf{k}\nu} \rangle = \delta_{\mu\nu})$  and is a complete basis of  $\mathbb{R}^{3N}$ . The cell eigenvectors are  $\sqrt{\text{mass}}$ -weighted *polarization vectors* [\[2,3\]](#page-21-0), but polarization vectors are not mutually orthogonal when there is more than one atomic species.

For any wave vector **k**, we can express the identity in  $\mathbb{R}^{3N}$ as  $\sum_{\mu=1}^{3N} |\varepsilon_{\mathbf{k}\mu}\rangle\langle\varepsilon_{\mathbf{k}\mu}|$  and insert it into Eq. (1), to define the set of *mode-projected correlation functions* at wave vector **k**,

$$
C_{\mathbf{k}\mu}(\mathbf{R},t) \equiv \langle \langle \dot{\psi}_{\mathbf{R}_0}(t_0) | \varepsilon_{\mathbf{k}\mu} \rangle \langle \varepsilon_{\mathbf{k}\mu} | \dot{\psi}_{\mathbf{R}_0+\mathbf{R}}(t_0+t) \rangle \rangle_{\mathbf{R}_0,t_0}.
$$

From this definition, and Eqs. (1) and (2), it follows that  $C(\mathbf{R}, t) = \sum_{\mathbf{k}\mu} C_{\mathbf{k}\mu}(\mathbf{R}, t)$  and  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega) = \sum_{\mu} \hat{\mathcal{E}}^{\mathcal{K}}_{\mathbf{k}\mu}(\mathbf{k}, \omega),$ where  $\hat{\mathcal{E}}_{\mathbf{k}\mu}^{K}(\mathbf{k},\omega)$  denotes the discrete Fourier transform of  $C_{\mathbf{k}\mu}(\mathbf{R}, t)$  with respect to **R** and *t*. In Sec. [V,](#page-10-0) we will pay particular attention to the LO mode and we will denote its cell eigenvector and mode-projected correlation function at **k** by  $\ket{\varepsilon_{\mathbf{k}}^{\mathrm{LO}}}$  and  $C_{\mathbf{k}}^{\mathrm{LO}}(\mathbf{R}, t)$ , respectively.

When studying individual modes, we will make use of the *normalized mode spectrum* or *mode distribution* of mode **k**μ, defined as

$$
f_{\mathbf{k}\mu}(\omega) \equiv \frac{\hat{\mathcal{E}}_{\mathbf{k}\mu}^{\mathcal{K}}(\mathbf{k}, \omega)}{\sum_{\omega} \hat{\mathcal{E}}_{\mathbf{k}\mu}^{\mathcal{K}}(\mathbf{k}, \omega)}.
$$
 (3)

<span id="page-6-0"></span>This function is the distribution among frequencies  $\omega$  of the energy of oscillatory motion in the bulk of the crystal along the eigenvector of normal mode  $\mathbf{k}\mu$ .

When studying the spectrum as a whole, calculating  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega)$  from  $C(\mathbf{R}, t)$  is equivalent to calculating  $\hat{\mathcal{E}}^{\mathcal{K}}_{\mathbf{k}\mu}(\mathbf{k}, \omega)$ from  $C_{\mathbf{k}\mu}(\mathbf{R},t)$  for each mode  $\mathbf{k}\mu$  and summing over all modes. Therefore the spectra plotted in Fig. [1](#page-1-0) are plots of

$$
f(\mathbf{k}, \omega) \equiv \frac{\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega)}{\sum_{\mathbf{k}\omega} \hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega)} = \frac{\sum_{\mu} \hat{\mathcal{E}}^{\mathcal{K}}_{\mathbf{k}\mu}(\mathbf{k}, \omega)}{\sum_{\mathbf{k}\omega} \sum_{\mu} \hat{\mathcal{E}}^{\mathcal{K}}_{\mathbf{k}\mu}(\mathbf{k}, \omega)}.
$$
 (4)

At thermal equilibrium, the equipartition theorem states that the expectation value of the kinetic energy of each mode **k** $\mu$  is  $\frac{1}{2}k_BT$  $\frac{1}{2}k_BT$  $\frac{1}{2}k_BT$ . Therefore, if the spectra plotted in Fig. 1 were converged fully, the denominators of both expressions in Eq. (4) would be  $\frac{3}{2}Nk_BT$ . However, regardless of whether or not  $f(\mathbf{k}, \omega)$  is converged with respect to simulation time or supercell size, it is the exact distribution, among wave vectors and frequencies, of the average kinetic energy per primitive cell of the trajectory from which the VVCF was calculated.

### **B. Energy expanded in mode coordinates**

The displacement of the crystal from equilibrium can be expressed as a sum of displacements along its 3*NNc* normal mode eigenvectors. We denote the frequency of mode  $\mathbf{k}\mu$  by  $\omega_{\mathbf{k}\mu}$ , and the projection of the crystal's displacement from equilibrium at time *t* onto its normalized eigenvector by  $Q_{\mathbf{k}\mu}(t)$ , which we refer to as its *mode coordinate*. The total energy of the crystal, as a function of the mode coordinates and their time derivatives, can be expressed as

$$
\mathcal{H} = \mathcal{H}_0 + \mathcal{K} + \mathcal{H}_2 + \mathcal{H}_3 + \mathcal{H}_4 + \mathcal{H}_5 + \cdots, \qquad (5)
$$

where  $\mathcal{H}_0$  is the potential energy of the static lattice in the  $T \rightarrow 0$  limit (excluding zero point energy);  $\mathcal{H}_1$ vanishes because the first partial derivatives of the potential energy vanish at equilibrium; the kinetic energy is  $\mathcal{K} = \frac{1}{2} \sum_{\mathbf{k}\mu} |\dot{Q}_{\mathbf{k}\mu}|^2$ ; the harmonic term of the potential energy is  $\mathcal{H}_2 = \frac{1}{2} \sum_{\mathbf{k}\mu} \omega_{\mathbf{k}\mu}^2 |Q_{\mathbf{k}\mu}|^2$ , and  $\mathcal{H}_p$ , where  $p > 2$ , denotes an anharmonic term of order *Q<sup>p</sup>*.

## **C. Difference between the kinetic energy distribution and the VDOS**

As discussed in Sec. [I,](#page-0-0)  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k},\omega)$  is usually interpreted as being proportional to the VDOS at low *T* [\[40–47\]](#page-22-0). However, this interpretation is only valid under the simplifying assumptions that there are as many vibrational states as there are degrees of freedom and that the crystal is at thermal equilibrium. When these assumptions are not valid, the kinetic energy distribution is a very different quantity to the VDOS because there is kinetic energy at  $(k, \omega)$  if one or more waves exist with wave vector **k** and frequency  $\omega$ ; and because the most general and rigorous treatments of phonon theory do not place any restrictions on which waves may exist at finite *T* .

Therefore let us assume that one of the defining characteristics of each "state" that contributes to the VDOS is that it is independent, or approximately independent in the temperature range of interest, of the amount of kinetic energy that "occupies" it. Then, the only viable definition of the set of all states is this one: at every point  $(k, \omega)$ , there are exactly as many occupiable vibrational states as there are degrees of freedom in a primitive unit cell.

For example, in our MD simulations, there could be kinetic energy at any point on a four-dimensional lattice in reciprocal spacetime whose lattice spacing along the frequency axis is inversely proportional to the total simulation time and whose lattice spacings along the three wave vector axes are inversely proportional to the linear dimensions of the simulation supercell. Therefore the VDOS of the simulated system is a uniform distribution with exactly  $3N = 6$  states per point  $(k, \omega)$  of the lattice that our simulation samples.

To illustrate that this is the case, note that each spectrum in Fig. [1](#page-1-0) is pixelated. The 2d lattice consisting of points at the centers of the pixels is a 2d representation of the 4d VDOS. Therefore there are six states at the center of every pixel, including at the centers of pixels that are white. Pixels are white when none of the states at their centers are 'occupied' by kinetic energy. However, the fact that they are not occupied does not mean that they are not occupiable, and some of them can be seen to turn blue at high *T* (e.g., when the crystal melts), indicating that they have become 'occupied' by some of the kinetic energy.

#### **D. Independent-phonon approximations**

In the bottom right panel of Fig. [1,](#page-1-0) we compare the  $T = 300 \text{ K}$  band structures measured experimentally [\[13\]](#page-21-0) with those that we have calculated using three different (quasi)independent-phonon approximations; namely, the harmonic approximation (HA), the quasiharmonic approximation (QHA), and the quasiparticle approximation (QPA).

In the HA, all cubic and higher-order anharmonic terms in the potential energy are discarded, which is tantamount to assuming that there is no interaction between different modes, and the Helmholtz free energy,  $\mathcal{F}(V, T)$ , can be approximated as [\[1\]](#page-21-0)

$$
\mathcal{F}_H(V,T) = \mathcal{H}_0(V) + \sum_{\mathbf{k}\mu} \left[ \frac{1}{2} \hbar \omega_{\mathbf{k}\mu}(V) + k_B T \ln(1 - e^{-\hbar \omega_{\mathbf{k}\mu}(V)/k_B T}) \right], \quad (6)
$$

where  $k_B$  is the Boltzmann constant, and the '*H*' subscript on  $\mathcal{F}_H$  indicates that it is the free energy within the *harmonic* approximation,  $\mathcal{H} \approx \mathcal{H}_0 + \mathcal{K} + \mathcal{H}_2$ . Notice that  $\mathcal{F}_H$  only depends on *T* via the second term in the summation over modes, but that it depends on *V* via the cohesive energy,  $\mathcal{H}_0(V)$ , and the  $T \to 0$  mode frequencies,  $\omega_{\mathbf{k}\mu}(V)$ .

When the third term vanishes in the  $T \rightarrow 0$  limit, it becomes  $\lim_{T\to 0} \mathcal{F}_H(V, T) = \mathcal{H}_0(V) + \sum_{\mathbf{k}\mu} \frac{1}{2} \hbar \omega_{\mathbf{k}\mu}$ . However, if phonons are treated classically, as in our MD simulations, the zero-point energy term does not exist and we simply have  $\lim_{T\to 0} \mathcal{F}_H(V,T) = \mathcal{H}_0(V)$ . Therefore the  $T \to 0$  limits of the mode frequencies calculated within the QHA and the QPA are different. They are the harmonic frequencies calculated at the volumes that minimize  $\mathcal{H}_0(V) + \sum_{\mathbf{k}\mu} \frac{1}{2} \hbar \omega_{\mathbf{k}\mu}(V)$  and  $\mathcal{H}_0(V)$ , respectively.

#### *1. Quasiharmonic approximation (QHA)*

<span id="page-7-0"></span>One of the most important ways in which the vibrational spectra of materials change with *T* is via thermal expansion: Lengthened bonds tend to be weakened bonds, and weakened bonds vibrate with lower frequencies. Therefore thermal expansion tends to lower phonon frequencies, on average. Within perturbation theory, this effect is often modelled using the QHA, which entails calculating the normal mode frequencies at a range of volumes and using them in Eq.  $(6)$ to calculate  $\mathcal{F}_H(V, T)$ . The volume  $V_{\text{min}}(T)$  that minimizes  $\mathcal{F}_H(V, T)$  is then found and treated as the equilibrium volume at temperature *T* , and the normal mode frequencies at this volume are used to calculate thermodynamic properties from  $\mathcal{F}_H(V_{\text{min}}, T)$ . In the most commonly used form of the QHA  $[10,14]$ , which is the form that we use, the dependence of  $V_{\text{min}}$ on *T* is ignored when taking derivatives of  $\mathcal{F}_H(V_{\text{min}}, T)$  with respect to *T* .

The QHA approximates the shift of phonon frequencies by thermal expansion, which is often the largest effect of anharmonicity on vibrational spectra at low *T* . However it does not explicitly describe any phonon-phonon interactions and therefore cannot describe other important effects of anharmonicity, such as the broadening of peaks in the mode spectra,  $f_{\mathbf{k}\mu}$ , as *T* increases.

#### *2. Quasiparticle approximation (QPA)*

At very high *T*, such as in the liquid, each mode spectrum  $f_{\mathbf{k}\mu}$  is not strongly peaked at any particular frequency. However, in the  $T \to 0$  limit, it becomes the delta function  $f_{\mathbf{k}\mu}(\omega) = \delta(\omega - \omega_{\mathbf{k}\mu})$  and at small finite values of *T* it is a sharply peaked distribution of finite width.

We denote the peak position of mode distribution  $f_{k\mu}$  at temperature *T* and volume *V* by

$$
\bar{\omega}_{\mathbf{k}\mu}(V,T) \equiv \omega_{\mathbf{k}\mu}(V) + \Delta \omega_{\mathbf{k}\mu}(T),\tag{7}
$$

where  $\omega_{\mathbf{k}\mu}(V)$  is the harmonic frequency in the  $T \to 0$  limit at volume *V* and  $\Delta \omega_{\mathbf{k}\mu}(T)$  is a finite-*T* correction, which does not depend on *V* to leading order in anharmonicity [\[1\]](#page-21-0). At constant pressure,  $V$  is determined by  $T$ ; therefore we can write  $\bar{\omega}_{\mathbf{k}\mu}(V,T) = \bar{\omega}_{\mathbf{k}\mu}(V(T),T) = \bar{\omega}_{\mathbf{k}\mu}(T)$ . The shifted frequencies,  $\bar{\omega}_{k\mu}(T)$ , are the QP frequencies. They can be used to calculate thermodynamic properties in a manner that is similar, but not identical for all properties [\[1\]](#page-21-0), to how harmonic phonon frequencies are used.

### **E. Band broadening**

As discussed,  $f_{\mathbf{k}\mu}$  is a delta function in the  $T \rightarrow 0$  limit and, as *T* increases from this limit, its first effects on  $f_{\mathbf{k}\mu}(\omega)$ are to broaden it and to shift it in frequency.

If phonons were independent entities that are created and annihilated in sudden random occurrences, which might be described as *collisions* or *scattering events*, then, at thermal equilibrium, the average rate at which phonons of each mode  $k\mu$  were created would be equal to the rate at which they were annihilated; and annihilation events would be Poisson distributed. It follows that phonon lifetimes would be exponentially distributed and that each mode correlation function

 $C_{k\mu}(\mathbf{R},t)$  would decay exponentially as a function of time [\[56\]](#page-23-0).

In the  $T \to 0$  limit,  $C_{k\mu}(\mathbf{R}, t)$  is sinusoidal in both **R** and *t*. Therefore, when annihilation events are Poisson distributed at finite  $T$ , it becomes an exponentially decaying sinusoid,  $C_{\mathbf{k}\mu}(\mathbf{R},t) \sim e^{-\gamma \mathbf{k}\mu t} \cos(\omega_{\mathbf{k}\mu}t)$ , and its Fourier transform,  $\hat{\mathcal{E}}_{\mathbf{k}\mu}(\mathbf{k},\omega)$ , is Lorentzian. Therefore  $f_{\mathbf{k}\mu}(\omega)$  is also Lorentzian, i.e.,

$$
f_{\mathbf{k}\mu}(\omega) \approx \frac{A_{\mathbf{k}\mu}}{(\omega - \bar{\omega}_{\mathbf{k}\mu})^2 + \gamma_{\mathbf{k}\mu}^2},\tag{8}
$$

where  $A_{\mathbf{k}\mu}$  is a constant,  $\bar{\omega}_{\mathbf{k}\mu}$  is the QP frequency,  $2\gamma_{\mathbf{k}\mu}$  is the full width at half maximum (FWHM) of the Lorentzian and  $\gamma_{k\mu}$  is the rate of exponential decay of the energy of a phonon QP of mode  $\mathbf{k}\mu$ . Therefore, when  $f_{\mathbf{k}\mu}$  can be fit closely by a Lorentzian,  $\gamma_{\mathbf{k}\mu}$  quantifies the degree to which it has been broadened by temperature.

On the other hand, when phonons are treated as lattice waves that continuously exchange energy, each *T* -broadened spectrum,  $f_{\mathbf{k}\mu}(\omega)$ , is only Lorentzian at very low *T* when it is a very narrow peak. At higher *T* , its shape is determined by the relative strengths of its couplings to other modes.

At any finite T, if  $\Delta \omega > 0$  is sufficiently small we can interpret  $f_{\mathbf{k}u}(\omega)\Delta\omega$  as the probability that the frequency of the oscillation along the eigenvector of mode  $\mathbf{k}\mu$  at a randomly chosen time is between  $\omega - \frac{1}{2}\Delta\omega$  and  $\omega + \frac{1}{2}\Delta\omega$ . We could also interpret it as the probability that the duration of the complete oscillation that begins at a randomly chosen time is between  $2\pi/(\omega + \frac{1}{2}\Delta\omega)$  and  $2\pi/(\omega - \frac{1}{2}\Delta\omega)$ . Therefore we can use Shannon's theorem to quantify the effects of *T* on mode  $\mathbf{k}\mu$  by quantifying the degree of uncertainty in its frequency. As Shannon demonstrated in the context of signal processing [\[57\]](#page-23-0), the correct way to quantify this uncertainty is by the *mode entropy*,

$$
S_{\mathbf{k}\mu} \equiv -\sum_{\omega} f_{\mathbf{k}\mu}(\omega) \ln f_{\mathbf{k}\mu}(\omega).
$$
 (9)

For a given variance of  $f_{\mathbf{k}\mu}$ , the shape that maximizes  $S_{\mathbf{k}\mu}$  is a Gaussian.

Although using the mode entropy,  $S_{\mathbf{k}\mu}$ , to quantify the degree of broadening is both more general, and better justified theoretically, than fitting to a Lorentzian, the latter is more common in the experimental literature. Therefore we calculated both  $S_{\mathbf{k}\mu}$  and  $\gamma_{\mathbf{k}\mu}$ .

#### **F. Anharmonicity**

Phonon-phonon interactions can be considered explicitly in perturbation theory by including cubic and higher-order terms in the truncated Taylor expansion of the potential energy. However, at any chosen order, phonon perturbation theory fails when *T* is high enough. Furthermore, although in the  $T \to 0$  limit each individual term of order  $Q^m$  contributing to  $\mathcal{H}_m$  in Eq. [\(5\)](#page-6-0) is larger than each individual term of a higher order  $Q^p$  contributing to  $\mathcal{H}_p$ , there are more nonvanishing  $Q^p$  terms than  $Q^m$  terms, in general. For example, as Ashcroft and Mermin [\[3\]](#page-21-0) point out, the number of quartic terms that do not vanish by symmetry can be much larger than the number of nonvanishing cubic terms. As a result, even at low  $T$ , the magnitude of  $H_4$  can be comparable to, <span id="page-8-0"></span>or greater than, the magnitude of  $H_3$ . They also point out that a crystal would not be stable if its Hamiltonian was  $\mathcal{H} \equiv \mathcal{H}_0 + \mathcal{K} + \mathcal{H}_2 + \mathcal{H}_3$ ;  $\mathcal{H}_4$  must be added for stability.

The fact that  $|\mathcal{H}_m|$  is not necessarily greater than  $|\mathcal{H}_p|$  when  $p > m$  makes truncation of Eq. [\(5\)](#page-6-0), which is a necessary step in any application of perturbation theory to real materials at finite *T* , formally unjustified. The method based on correlation functions that we use is free of such complications. Therefore it is a useful tool for checking when/whether perturbation theories are applicable and for assessing their accuracies.

A related limitation of perturbation theory is that it is often necessary to calculate and analyze so many phonon interaction terms that the underlying physics can become lost in the clutter. For example, the anomalous broadening of the longitudinal optical (LO) mode near the BZ center evident in Fig. [1](#page-1-0) has been studied by perturbation theory and explained, in part, as a consequence of the band structure [\[10,](#page-21-0)[38\]](#page-22-0): the number of nonvanishing contributions to  $\mathcal{H}_3$  that involve the LO mode near  $\Gamma$  is very high because of the locations in  $(\mathbf{k}, \omega)$ space of the other modes.

This is an important observation, which is likely to be part of a complete explanation, but on its own it is not a complete and satisfactory explanation. Explaining one feature of a spectrum as being a consequence of its other features is not satisfactory because the spectrum as a whole is not selfdetermined: it is determined by interactions between waves passing through a lattice of ions. A more complete explanation of a spectral feature would explain it in terms of the motions of waves and ions.

It is also important to note that, when treating phonon  $\mathbf{k}\mu$ at finite *T* as a QP whose dynamics are damped with a decay constant  $\gamma_{k\mu}$ , the contribution of modes  $\mathbf{q}v_1$  and  $\mathbf{k} + \mathbf{\bar{q}}v_2$  to the value of  $\gamma_{k\mu}$  could be negligible despite the magnitude of the term in  $\mathcal{H}_3$  proportional to  $Q_{\mathbf{k}\mu}Q_{\mathbf{q}\nu_1}Q_{\bar{\mathbf{k}}+\bar{\mathbf{q}}\nu_2}$  being very large. It is not enough that modes  $\mathbf{q}v_1$  and  $\mathbf{k} + \mathbf{\bar{q}}v_2$  exchange a lot of energy with mode **k**μ: they must do so *irreversibly*. This means that if they absorb some of mode  $\mathbf{k}\mu$ 's energy, they must dissipate it, by coupling to more modes, before it has time to return to mode  $\mathbf{k}\mu$ . For example, when two harmonic oscillators are coupled, their combined energy oscillates back and forth between them. There are times when one of them has all of the energy and times when the other has it all. Therefore, in a thermal population of phonons, it is not necessarily possible to calculate decay constants accurately as sums of few-phonon contributions.

Reference [\[6\]](#page-21-0) provides an illustration of this point. It was found that when two coaxial nanotubes are in relative sliding motion, phonons are resonantly excited at *every* sliding velocity. However, it is only at a small number of velocities that these resonances manifest as a strong friction force that slow the sliding motion down. The energy exchange between the mechanical motion of the tubes and most of the resonantly excited phonons is equitable. High friction only occurs at velocities for which the resonantly excited phonons dissipate the energy they absorb from the sliding motion more quickly than they return it to that motion.

*Irreversible* dissipation of energy always requires the participation of very large numbers of modes. Therefore because, in practice, perturbation theories are limited to considering few-phonon processes, they have a fundamental limitation that the MD-based method used here and in Ref. [\[6\]](#page-21-0) does not share.

The purpose of this section is not to denigrate phonon perturbation theories. It is to point out that both perturbation theories and the correlation function approach to calculating spectra are limited, but in different ways; therefore they complement one another. We have used harmonic phonon theory extensively to analyze our spectra, and it is likely that we would have learned much more with the help of anharmonic terms.

In Sec. [VB3,](#page-12-0) we provide a simple explanation of the anomalous LO broadening, in terms of ions and waves, and without referring to any other feature specific to the vibrational spectrum of MgO and the other similar materials in which anomalous LO broadening has also been observed [\[58–67\]](#page-23-0).

We suggest that the *acoustic warping* mechanism we propose in Sec.  $VB3$  modulates the frequencies of LO modes, and therefore the expectation values of their occupation numbers. At thermal equilibrium, a mode whose frequency is fixed exchanges energy with other modes equitably: its *net* rate of energy exchange with other modes vanishes. However, because the LO mode's frequency  $\omega^{\text{LO}}$  is changing, the expectation value,  $[\exp(\hbar \omega^{LO}(t)/k_B T) - 1]^{-1}$ , of its occupation number is changing, and this might make its net rate of energy exchange with other modes finite. This net exchange of energy may happen quickly (e.g., for the reasons explained by Giura *et al.* [\[38\]](#page-22-0)) relative to the period of the acoustic mode modulating  $\omega^{LO}$ , and so it may contribute significantly to the degradation of lower-frequency bands.

## **IV. SIMULATION DETAILS**

### **A. Atomistic force field**

Atomistic molecular dynamics (MD) and lattice dynamics (LD) simulations were performed using a polarizable-ion potential of the form described in Refs. [\[25,28,29\]](#page-22-0), and used to study the pressure dependence of the melting temperature of MgO in Ref. [\[23\]](#page-22-0).

Our method of force field construction [\[22\]](#page-22-0) is a form of supervized machine learning [\[32,](#page-22-0)[68\]](#page-23-0), albeit one that predates the widespread adoption of the term *machine-learning* in this context. However, the interactions described by most machine learning force fields are *near-sighted* [\[68\]](#page-23-0), whereas a realistic description of long-ranged Coulomb interactions is essential when studying zone center LO phonons. Therefore we did not impose any accuracy-lowering near-sightedness constraint on the mathematical form of our potential.

Our force field's parameters were fit to an effectively infinite dataset of forces, energy differences and stress tensors from density functional theory (DFT) calculations using the PBEsol functional [\[69\]](#page-23-0). As discussed in Refs. [\[22](#page-22-0)[,70,71\]](#page-23-0), describing the polarizability of oxide anions, either implicitly or explicitly, is necessary for an atomistic model of an oxide to accurately describe the long-range fields that are intrinsic to LO phonons. However, cations' electrons tend to be much more tightly bound and we did not find a significant improvement in the fit to DFT data when Mg cations were polarizable, so we assigned a polarizability of zero to them.

<span id="page-9-0"></span>TABLE I. Comparison between the average lattice constants in our constant pressure MD simulations,  $a_{MD}$ , and the *quasiharmonic lattice* constants,  $a_{QHA}$ , at the same temperatures. The numbers in brackets are the percentage changes in volume with respect to the  $T \rightarrow 0$  volume.

0K	100K	300 K	500 K	1000 K	2000 K	3000 K	3500 K
					$a_{\text{QHA}}/\text{\AA}$ 4.2294 (0.0) 4.2296 (0.02) 4.2350 (0.40) 4.2442 (1.05) 4.2727 (3.10) 4.3558 (9.24) a <sub>MD</sub> /Å 4.2119 (0.0) 4.2202 (0.59) 4.2305 (1.33) 4.2410 (2.09) 4.2670 (3.98) 4.3296 (8.62) 4.4120 (14.9) 4.4640 (19.1)		

The mathematical form of the potential and the values of the parameters used are quoted in Appendix [B.](#page-21-0) In brief, it is the sum of a pairwise interaction, comprising a Morse potential and a 1/*r* Coulomb interaction, and the Coulomb interactions between dipoles induced on oxygen anions and the charges and induced dipoles of other ions. The dipole moment of the *i*th oxygen anion is expressed as  $\mathbf{p}_i = \mathbf{p}_i^{\text{SR}} + \mathbf{p}_i^{\text{LR}}$ , where  $\mathbf{p}_i^{SR}$  is a short-range (SR) contribution caused by asymmetry of the space in which its electron cloud is confined by its six cation neighbors, and  $\mathbf{p}_i^{\text{LR}}$  is the dipole moment induced by the local electric field  $(E_i)$  from the charges and dipole moments of all other ions. At each step of the MD simulation, we first use the method of Wilson and Madden to calculate  $\mathbf{p}_i^{\text{SR}}$  [\[72\]](#page-23-0) as a function of the distances of ion *i* to neighboring ions. Then we iterate the coupled equations (one for each anion)  $\mathbf{p}_i = \mathbf{p}_i^{SR} + \alpha \mathbf{E}_i[\{\mathbf{p}_j\}_{j \neq i}]$  to self-consistency in the set of all dipole moments  $\{p_i\}$ , where oxygen's polarizability,  $\alpha$ , is among the parameters fit to DFT data. Finally, we calculate the Coulomb energy of interaction between each dipole moment and the charges and dipole moments of all other ions.

#### **B. Molecular dynamics**

We simulate under periodic boundary conditions and our supercell is a  $10 \times 10 \times 10$  repetition of the two-atom rhombohedral primitive unit cell. We use velocity rescaling, followed by ≈10 ps of equilibration in the *NVE* ensemble, to prepare for production runs at each temperature, *T* . Our production runs of  $T \approx 100$  ps are also performed in the *NVE* ensemble and the reported values of *T* are calculated from the average kinetic energy of the production run. We use a time step of 0.725 fs and sample positions and velocities every ten steps for later analysis. We chose to perform our MD simulations at constant volume, but with  $P \approx 0$ , instead of at a fixed average pressure and a variable volume. Although the magnitudes of the fluctuations of each primitive cell's volume would be reduced, to some degree, by simulating with the supercell volume fixed, this was deemed preferable to polluting our spectra with unphysical artifacts of an MD barostat.

We performed one set of simulations with the volume  $V(T)$ at each *T* chosen such that the average pressure (*P*) in the crystal at that *T* is close to zero. The averages of  $(P, T)$  in these simulations were (100 K, 0.01 GPa), (301 K, −0.01 GPa), (502 K, −0.02 GPa), (1004 K, 0.16 GPa), (1995 K, 0.10 GPa), (2998 K, −0.06 GPa), (3521 K, 0.16 GPa), and (3813 K, 6.7 GPa). The pressure is large in the highest temperature simulation because the crystal has melted and we did not repeat the simulation with the volume adjusted for the liquid phase. At all lower temperatures, the simulation cell is crystalline and the maximum estimated percentage error in the simulated volume is  $|\Delta V/V| = \Delta P/B \approx 0.1\%$ , where  $B \approx 160$  GPa is its bulk modulus under ambient conditions [\[14](#page-21-0)[,73\]](#page-23-0).

Another set of MD simulations at approximately the same values of *T* was performed with the volume fixed at its value in the low temperature limit, i.e., at the value obtained by minimizing the enthalpy with respect to atomic positions and the lattice parameter. The averages of  $(P, T)$  in this set of simulations were (100 K, 1.1 GPa), (301 K, 2.4 GPa), (506 K, 3.8 GPa), (1008 K, 6.9 GPa), (1982 K, 12.8 GPa), (2986 K, 18.6 GPa), and (3458 K, 21.3 GPa).

### *1. Correlation functions and their Fourier transforms*

We used *fast Fourier transforms* (FFTs) to compute vibrational spectra from correlation functions. We calculated spatial correlations up to the maximum distance possible with our supercell, which is  $L = 5a$ , where *a* is the primitive lattice parameter. We used the entire production run trajectory of length  $T$  to calculate temporal correlations. The resolutions,  $2\pi/L$  and  $1/\mathcal{T}$ , with which spectra can be calculated as functions of wave vector  $(k)$  and frequency  $(f)$ , respectively, are determined by the sizes of the domains in space  $(L)$  and time  $(T)$ , respectively, on which the correlation functions are calculated. With our supercell we are able to sample five commensurate *k* points between  $\Gamma$  ([0,0,0]) and *X* ([0.5,0,0.5]), where **k** is a *commensurate k* point if waves with wave vector **k** respect the periodic boundary conditions, i.e., if  $\mathbf{k} = m_1 \mathbf{k}_1 + m_2 \mathbf{k}_2 + m_3 \mathbf{k}_3$ , where each  $m_i$  is an integer,  $\mathbf{k}_i$ is parallel to the *i*th supercell lattice vector, and if the length of that lattice vector is an integer multiple of  $\lambda = 2\pi/|\mathbf{k}_i|$ . The frequency resolution of our raw spectra is 0.01 THz, but we smooth them along the frequency axis by convolving them with Gaussians of standard deviation 0.05 THz, for the spectra at  $T \lesssim 1000$  K, and 0.1 THz for the spectra at  $T \gtrsim 2000$  K.

## **C. Lattice dynamics**

To calculate the phonon band structure within the harmonic approximation, we find the equilibrium  $(T \to 0)$  structure of the supercell and then calculate the dynamical matrix from finite differences of forces after displacing atoms slightly from equilibrium. To avoid artefacts of interpolation, we only plot the frequencies at commensurate *k* points, but we increase their density in reciprocal space by calculating phonon spectra on  $M \times M \times M$  supercells for  $M \in \{6, 7, 8, 9, 10, 11\}.$ 

As discussed in Sec.  $IIID1$ , the QHA simply amounts to calculating the phonon band structure at the equilibrium volume for the given temperature,  $T'$ , rather than at the equilibrium volume of the  $T \rightarrow 0$  limit. The equilibrium volume at temperature  $T'$  is the volume that minimizes the free energy; but we cannot calculate the free energy easily, precisely, or accurately due to important contributions to it from improbable microstates. Improbable microstates are unlikely to

<span id="page-10-0"></span>be sampled in an MD simulation and if, by chance, they did occur, they would be oversampled. However, we can calculate the free energy under the simplistic assumptions that phonons do not interact with one another and that the vibrational energy is distributed among the phonon bands according to the Bose-Einstein distribution. Therefore, when we apply the QHA we assume that band theory is a good approximation and we calculate the phonon bands as a function of volume. We use these frequencies in Eq.  $(6)$ , which is exact in the  $T \rightarrow 0$  limit where band theory is exact, to find the volume that minimizes  $F(V, T)|_{T=T'}$ .

We calculated  $F(V, T)|_{T=T'}$  for a range of volumes corresponding to a set of lattice parameters with uniform spacing 0.02 Å, and we interpolated between these values with cubic splines. To calculate the phonon frequencies at each volume we use a  $10\times10\times10$  supercell to calculate the dynamical matrix at each **k** in the  $10 \times 10 \times 10$  commensurate set from finite differences of forces. When calculating  $F(V, T)|_{T=T'}$  from Eq. [\(6\)](#page-6-0), the sum over **k** is a sum over the set of commensurate **k** points.

## **V. RESULTS**

We presented the central result of this work in Fig. [1,](#page-1-0) which contains eight kinetic energy spectra that show how thermal disorder degrades the optical bands of MgO. Before analyzing and discussing these spectra, we discuss some limitations of our calculations that should be borne in mind while interpreting them.

### **A. Sources of inaccuracy and imprecision**

The first limitation is our use of approximate interatomic forces. The accuracy with which our force field calculates phonon frequencies at 300 K is evident in the bottom-right panel of Fig. [1,](#page-1-0) where we compare with measured phonon frequencies and with those calculated *ab initio* with DFPT. The underestimation of LO frequencies at small wave vectors is common for force fields of this kind [\[22,29,](#page-22-0)[74\]](#page-23-0), and has been discussed in Ref. [\[29\]](#page-22-0). It is likely to be caused by the induced dipoles of the force field *over*screening the long wavelength electric fields that LO phonons create, and which increase the frequencies of LO phonons relative to transverse optical (TO) phonons.

The accuracy with which our force field describes thermal expansion is evident in Table [I.](#page-9-0) Its accuracy for other properties and at higher *T* is more difficult to assess because we have neither more accurate calculations nor experimental measurements to compare with. However, force fields of the same mathematical form, and parameterized in the same way, were used in Refs. [\[22,23\]](#page-22-0) and shown to predict the pressure dependences of the crystal's volume (*V* ) and melting temperature  $(T_m)$  accurately, and to produce pair correlation functions for molten MgO in perfect agreement with those produced by *ab initio* MD simulations.

To parametrize the force field used in this work we fit to DFT data calculated with the PBEsol functional [\[69\]](#page-23-0), which tends to be more accurate than the local density approximation used in Refs. [\[22,23\]](#page-22-0). Furthermore, we fit the parameters to DFT calculations of *crystalline* MgO, whereas those used in

Refs. [\[22,23\]](#page-22-0) were required to describe MgO in both molten and crystalline forms, which meant compromising on the accuracy with which each phase was described.

Based on the tests performed here and in those previous works we suggest that, at worst, our force field should be regarded as describing an MgO-like material. We further suggest that some of the gross features of the *T* dependence of the spectra calculated with it, including those that we have selected for discussion below, are caused by simple physical mechanisms that would occur in other ionic crystals and therefore might manifest in their vibrational spectra.

A second limitation of our calculations is that our MD simulations are classical simulations. Therefore their accuracy at low *T* is questionable. However, when *T* is comparable to, or greater than, the Debye temperature ( $T_D \sim 1000$  K), which is the range of most interest for studying the breakdown of band theory, this approximation should not affect our results significantly. Furthermore, we will show in Sec.  $VC$  that the heat capacities calculated from the quasiparticle energies extracted from MD simulations are in very good agreement with measurements at all values of *T* between 300 and 1800 K.

We regard our use of a supercell of finite size in our MD simulations as by far the most important source of inaccuracy and imprecision in the kinetic energy spectra presented. It means that the only lattice waves present our MD simulations were those whose wave vectors are commensurate with the supercell. Our use of a  $10 \times 10 \times 10$  supercell means that the longest finite wavelength among the phonons in our simulation was only ten times the length of a primitive lattice vector. To make clear that we only calculate spectra at a finite number of points in  $(k, \omega)$  space, we have pixelated the spectra, with one pixel centered at each commensurate **k** point. However, the absence of any vibrations at incommensurate  $(k, \omega)$  points would change how energy is distributed among the commensurate set of points. For example, in a real crystal, there would be far more channels (modes) through which energy and momenta could be exchanged.

#### **B. Deviations of finite-***T* **spectra from band structures**

Let us now begin discussing Fig. [1,](#page-1-0) which compares the full  $(\mathbf{k}, \omega)$ -resolved kinetic energy spectra,  $\hat{\mathcal{E}}^{\mathcal{K}}(\mathbf{k}, \omega)$ , from our MD simulations at different values of *T* , after each one has been normalized so that it integrates to one.

As mentioned in Sec. III  $A_2$ , the equipartition theorem implies that if the spectra were converged fully with respect to simulation time, this normalization would be equivalent to dividing each one by the same constant and by *T* . Regions of high and low energy densities are colored dark blue and white, respectively, with the same color scale used at each *T* . For comparison, the  $T \rightarrow 0$  band structure and the finite- $T$ QHA band structures are plotted over the full spectrum with red and green triangles, respectively.

The spectra show a progressive transition between two limits of *T* : Phonon bands are well defined at low *T* , whereas at very high *T* they are much less well defined or not defined. At 300 K the phonon dispersions do not differ substantially from the  $T \rightarrow 0$  bands calculated within the HA: all of the vibrational energy is localized near the normal mode points,  $(k, \omega_{k\mu})$ , and the widths of the peaks at these points are <span id="page-11-0"></span>small. At 3800 K, on the other hand, the phonon bands have vanished and the spectrum is much more uniform and has little observable structure. This is because the crystal has melted and because spatial and temporal correlations are very short in a liquid. Therefore the phonon assumption of atoms moving collectively as waves has broken down completely.

The spectra at other *T* s show various stages of the progression between the low-*T* limit, in which atoms move as lattice waves with well defined frequencies and wave vectors, and the liquid, in which each atom moves independently of all other atoms, except those closest to it.

Note that MgO melts at  $T_m \lesssim 3100 \,\mathrm{K}$  [\[23,36\]](#page-22-0), but it is well known that, by imposing a degree of long-range order, the periodic boundary conditions used in MD simulations can prevent melting until  $T$  is significantly larger than  $T_m$ [\[75\]](#page-23-0). Therefore the  $T = 3500$  K spectrum in Fig. [1](#page-1-0) should be regarded as the spectrum of a superheated MgO crystal.

## *1. Selected features of the kinetic energy spectra*

There is a lot of complexity in the *T* dependence of the full spectrum (Fig. [1\)](#page-1-0) and a much more extensive and detailed study would be required to explain it all. Therefore we provide the data used to produce Fig. 1 in Ref. [\[76\]](#page-23-0) so that others may analyze it further. We focus our attention on two important gross features of the spectrum's *T* dependence, and on one striking specific feature.

The first gross feature is that all phonons shift to lower frequencies as *T* increases. Most of this softening can be attributed to the weakening of bonds by thermal expansion. We will discuss this in more detail in Sec. VC.

The second gross feature is that optical bands, and particularly LO bands, lose definition much more rapidly with increasing *T* than acoustic bands; and zone center modes  $(\mathbf{k} \to 0, \lambda \to \infty)$  lose definition much more rapidly than zone boundary modes. Much of the intensity seen at high frequencies and small wave vectors at low *T* gradually moves towards lower frequencies and larger wave vectors as *T* increases. This is easiest to see along the wave vector paths  $\Gamma \rightarrow X$  and  $\Gamma \rightarrow L$ , which are straight line segments connecting the zone center  $(\Gamma)$  to the high symmetry points *X* and *L* on the zone boundary. We will denote the wave vectors at *X* and *L* by  $\mathbf{k}_X$ and **k***L*, respectively.

When analyzing spectra, we will ignore the point  $\Gamma$  itself, which does not represent the limit  $\mathbf{k} \to 0$ , but the point  $\mathbf{k} = 0$ . The intensity at  $\Gamma$  in Fig. [1](#page-1-0) is the kinetic energy of rigid relative motion of the Mg and O sublattices in our simulations, which does not interest us. However, note the disappearance, at high *T* , of most of the intensity at optical frequencies and at wave vectors of  $\frac{1}{5}$ **k**<sub>*X*</sub>,  $\frac{2}{5}$ **k**<sub>*X*</sub>,  $\frac{1}{5}$ **k**<sub>*L*</sub>, and  $\frac{2}{5}$ **k**<sub>*L*</sub>, which are the smallest finite wave vectors along these paths that our simulation supercell can accommodate. The redistribution of intensity away from high frequencies at these wave vectors means that, at very high *T* , little of the crystal's kinetic energy exists as *coherent* optical waves whose wavelengths are greater than about five or ten lattice spacings. The TO branches retain significant amounts of energy at these wavelengths, but the LO bands have almost vanished. As discussed in Sec.  $II$ , much of this is a manifestation of thermal disorder reducing the correlation lengths and times of collective motions along

the cell eigenvectors of the affected modes, causing them to become less wavelike and more localized in spacetime.

While this is happening to optical bands at small wave vectors, the acoustic branches of the spectrum remain relatively robust. Melting manifests in the spectra as the sudden total loss of definition and integrity of the acoustic bands between  $T = 3500$  and 3800 K.

The broadening of bands does not *necessarily* mean that the waves contributing to the band are less coherent. A band of finite width can broaden further without the phonons' coherence lengths and lifetimes reducing further if the broadening is caused by a very slow modulation of the properties of the underlying lattice. As we discuss further below, the frequency and/or wave vector of an optical mode could be modulated by an acoustic phonon whose period and wavelength are much larger than the optical phonons' coherence length and lifetime, respectively.

The specific striking feature of the spectra that we have chosen to discuss is that the loss of definition of the optical modes appears to begin with an anomalously large broadening of the LO modes nearest to  $\Gamma$ , and to spread out from these  $(k, \omega)$  points as *T* increases. The broadening of these modes is even visible at 100 K, which is the lowest *T* at which we calculated the full spectrum, and it occurs for the LO phonons at all wave vectors  $\bf{k}$  near  $\Gamma$ , regardless of their directions.

We are not the first to observe the anomalous LO linewidth at low  $T$  in MgO  $[10,38]$  $[10,38]$ ; and similar features have been observed experimentally in similar materials since the 1960s, and more recently in calculated spectra [\[58–67\]](#page-23-0). However, we have not found other studies that show how it evolves as the crystal is heated to a very high  $T$ , or show that it marks the beginning of the gradual melting of the entire LO band. When the set of spectra in Fig. [1](#page-1-0) are examined collectively, the melting of the LO band may be the most noticeable feature of the spectrum's *T* dependence between 100 K and 3500 K.

Because the LO band melts gradually and systematically from its apparent origin, at low  $T$ , as a localized anomaly near the BZ center, and because similar anomalies have been observed in other materials, we believe that its primary cause is a simple and general physical mechanism, which we refer to as *acoustic warping of optical phonon fields*. The effects of this mechanism on vibrational spectra would be particular pronounced in crystals with large LO-TO splittings, such as strongly ionic materials with the rocksalt crystal structure. We explain it in the sections that follow.

We begin by suggesting an explanation for why strong coupling between low frequency acoustic phonons and optical phonons would melt optical bands much faster than acoustic bands. Then we explain the acoustic warping mechanism, why it would cause LO bands to melt, and why this melting would begin near the BZ center before gradually progressing outwards, until the only LO modes that remain are those at the BZ boundary.

# *2. Separation of optical and acoustic timescales in the long wavelength limit*

The acoustic branches can be thought of as a skeleton on which the optical branches "hang," because *spatially coher-* <span id="page-12-0"></span>*ent* countermotion of cations and anions about a reference structure would not happen unless the reference structure was itself spatially coherent. For example, consider the projection,  $\langle \varepsilon_{\mathbf{k}}^{\text{LO}} | \psi_{\mathbf{R}} \rangle$  of the displacement of cell **R** onto the cell eigenvector of the LO mode at **k**. When regarded as a function of **R**, this projection is very unlikely to have order on length scale  $\lambda$  if the crystal does not have order on length scale  $\lambda$ . Crystalline order is gradually lost as *T* increases, and this manifests as a gradual reduction of the correlation lengths and times of this projection, i.e., as an increase in the rate of decay of  $|C_k^{\text{LO}}(\mathbf{R}, t)|$  as a function of **R** at fixed *t* and as a function of *t* at fixed finite **R**.

The energy cost of acoustic distortions protects long-range order in the *time average* of the crystal's structure. However, if |**k**| is small, the crystal having long-range order when averaged over several periods of a long-wavelength acoustic (LWA) mode is not sufficient to protect long-range order in  $\langle \varepsilon_{\mathbf{k}}^{\text{LO}} | \psi_{\mathbf{R}} \rangle$ . It is insufficient because the periods (frequencies) of zone center LO modes are much shorter (higher) than those of LWA modes. Therefore long-range order of  $\langle \varepsilon_k^{\text{LO}} | \psi_{\text{R}} \rangle$  as a function of **R** relies, not on the degree to which the *time average* of LWA mode displacements preserve crystalline order (i.e., vanish), but on the degree to which they preserve it on timescales much shorter than LWA mode periods. Long-range order of the optical branches requires the *amplitudes* of LWA modes to be small, not their time-averaged displacements.

The converse is not true and, in either the limit of large LO frequency  $\omega_k^{\text{LO}}$  or the limit of small LWA frequency  $\omega_k^{\text{LWA}}$ , the coupling between LWA and LO modes is almost unidirectional: LO vibrations are highly sensitive to LWA mode displacements, whereas LWA vibrations are much less sensitive to LO mode displacements. The reason for the high sensitivity of LO phonons to LWA phonons will be discussed in Sec. VB3.

LWA phonons are relatively insensitive to LO phonons because an LO mode's period,  $2\pi/\omega_k^{\text{LO}}$ , is so short that LO displacements average to zero in much less than the period,  $2\pi/\omega_{\mathbf{k}}^{\text{LWA}}$ , of an LWA phonon. Therefore the forces they exert on an LWA mode cancel one another before the LWA mode has had time to respond to them. On the length and time scales relevant to the lowest-frequency LWA modes of a macroscopic crystal, the crystal is a continuum and the LWA modes do not see optical mode disorder directly, but experience its effects indirectly through the *T* dependence of the crystal's elastic constants.

This partial decoupling of a slow-moving degree of freedom from a much faster one is known as *adiabatic decoupling* [\[77–81\]](#page-23-0). It is exploited by the Born-Oppenheimer approximation: An excellent approximation to the force exerted on a heavy nucleus by electrons is calculable from the electron density, which can loosely be thought of as the time average of the electrons' positions.

Adiabatic decoupling does not mean that the LWA modes do not exchange energy with the optical modes. It means that their energy exchange is so rapid that they barely notice. The net energy exchange in a time  $\tau$  can be very large if  $\tau < 2\pi/\omega_{k_{\rm min}}^{\rm LO}$  but is likely to be negligible if  $2\pi/\omega_k^{LO} \ll \tau \ll 2\pi/\omega_k^{LWA}$ , because its average over one complete LO period is small and its average over many complete periods is even smaller. Optical mode disorder changes

so quickly that every complete LWA oscillation occurs in the presence of an almost equivalent background of optical displacements, whereas LWA disorder changes so slowly that every complete LO oscillation occurs in the presence of a unique and inequivalent background of LWA displacements.

Therefore it is approximately true that LWA phonons only experience the many rapidly changing LO displacements as a slight *dressing*, which changes their frequencies very little. Nevertheless, and at the same time, if there exists an effective LWA-LO coupling mechanism, the frequencies and coherence lengths and times of LO phonons are strongly influenced by whatever LWA displacements exist during their lifetimes.

Of course, the adiabatic decoupling picture, in which optical modes see 'frozen' acoustic modes and acoustic modes do not see optical modes because the time averages of their displacements vanish, is very much an idealized limiting case. Adiabatic decoupling of acoustic modes from optical modes becomes a perfect decoupling in the  $\mathbf{k} \to 0$  limit, but is likely to be far from perfect at the smallest finite wave vectors present in our simulations. Clearly this picture would not apply to the transverse acoustic (TA) and TO modes in the regions of the BZ where they occupy the same small frequency window between about 10 THz and 15 THz. However, it does seem to explain why the lowest-frequency acoustic modes are broadened less than the highest-frequency optical modes.

As *T* increases from the  $T \to 0$  limit, it is the LWA modes that are first to become active, because they are lowest in energy. Therefore there may be a common explanation for why the LO band is the first to melt, or partially melt, for why zone center modes melt before zone boundary modes, and for why acoustic bands are degraded less at the highest temperatures than optical bands: Acoustic modes are degraded less because of the partial immunity to optical disorder that adiabatic decoupling affords them. LO modes are not protected by adiabatic decoupling and, as we now explain, there is a very simple and effective mechanism by which they can be disrupted, and their frequencies changed, by acoustic disorder; or even by a single quasistatic acoustic perturbation of the crystal.

#### *3. Acoustic warping of the LO electric field*

By modulating the relative displacements of cations and anions, an LO wave of wave vector **k** creates regions of excess charge at its nodes, from which emanates an electric field,  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$ , of the same wavelength,  $\lambda_{\mathbf{k}}^{\text{LO}} \equiv 2\pi/|\mathbf{k}|$ . This field, which we referred to as the LO mode's *intrinsic field* above, opposes the LO wave's motion, thereby increasing its frequency,  $\omega_k^{\text{LO}}$ [\[3](#page-21-0)[,82\]](#page-23-0).

One way to understand the origin of  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  is to consider the case in which  $\lambda_k^{\text{LO}}$  is orders of magnitude larger than a primitive lattice spacing; and to imagine partitioning the crystal into primitive unit cells of dipole moment **d**, whose volume we will treat as infinitesimal.

If the LO mode at wave vector **k** becomes active, it modulates **d** along an axis parallel to  $\mathbf{k} \equiv \mathbf{k}/|\mathbf{k}|$  by modulating the displacements of Mg cations from the O anions with which they share a primitive cell. On length

scale  $\lambda_k^{\text{LO}} \gg |\mathbf{a}_1|, |\mathbf{a}_2|, |\mathbf{a}_3|$ , we can define a local spatial average of the cell dipole moment per unit volume,  $P \equiv \langle d \rangle / |a_1 \cdot (a_2 \times a_3)|$ , and treat it as a continuous function of position. The dependence of **d** on the choice of primitive cell makes **P** ill-defined [\[83\]](#page-23-0); however its derivatives with respect to space and time are the same for every choice. Therefore the density  $\rho_b \equiv -\nabla \cdot \mathbf{P}$  of *excess charge* or *bound ("b") charge* is independent of the choice of primitive cell and is a well-defined physical quantity.

If only the LO mode at **k** is active,  $\rho_b$  varies in direction **k** with wavelength  $\lambda_k^{\text{LO}}$ . Its magnitude is largest at the wave's *nodes*, which is where the Mg-O displacement varies most from cell to adjacent cell along **k**ˆ. When multiple LO modes are active,  $\rho_b$  has a contribution from each one and we will denote the contribution from the one with wave vector **k** by *ρ*LO *<sup>b</sup>*,**<sup>k</sup>**.

 $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  is the field emanating from  $\rho_{b,\mathbf{k}}^{\text{LO}}$ . If it was absent or negligible ( $\mathbf{E}_{\mathbf{k}}^{\text{LO}} \approx 0$ ) the LO and TO modes would have the same frequency in the long wavelength limit by symmetry [\[82\]](#page-23-0). Therefore near  $\Gamma$ ,  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  increases the value of  $\omega_{\mathbf{k}}^{\text{LO}}/2\pi$  by almost 10 THz. It follows that any weakening or strengthening of  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  could change  $\omega_{\mathbf{k}}^{\text{LO}}$  quite dramatically: a 10% reduction in  $|\mathbf{E}_{\mathbf{k}}^{\text{LO}}|$  near  $\Gamma$  would reduce  $\omega_{\mathbf{k}}^{\text{LO}}/2\pi$  by 10% of the difference between the frequencies of the LO and TO modes, which is ∼ 1 THz.

A quasistatic acoustic perturbation of the crystal changes  $\omega_{\mathbf{k}}^{\text{LO}}$  by weakening or strengthening  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$ ; and *different* quasistatic acoustic perturbations would result in different frequencies. Therefore the distribution of the LO mode's energy among frequencies, which is localized at a single frequency in the  $T \to 0$  limit, should broaden significantly as *T* activates the acoustic modes.

There are some obvious mechanisms by which TA and longitudinal acoustic (LA) distortions would change **E**LO **<sup>k</sup>** . The first is simply disorder: Acoustic perturbations whose wavelengths differed from  $\lambda_k^{\text{LO}}$  would break the periodicity of  $\mathbf{E}_k^{\text{LO}}$ <br>by breaking the periodicity of  $\rho_{b,k}^{\text{LO}}$ . This would reduce the LO wave's coherence and, in many or most cases, reduce  $\omega_k^{LO}$  by weakening **E**LO **<sup>k</sup>** .

A second mechanism is that a TA or LA perturbation with wave vector **k** (or  $m\mathbf{k}$ , where  $m \in \mathbb{Z}$ ), would change  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$ by perturbing the charge reservoirs centered at the antinodes of  $\rho_{b,k}^{\text{LO}}$ . A TA perturbation would displace the positive and negative reservoirs relative to one another along an axis perpendicular to  $\hat{\mathbf{k}}$ , whereas an LA perturbation would expand or compress them.

In planes perpendicular to **k**, the part of the crystal that a phonon with wave vector **k** perturbs is finite in size at any finite *T*. If the LO wave's lateral extent was smaller than  $\lambda_k^{\text{LO}}$ , a transverse relative displacement of  $\rho_{b,k}^{\text{LO}}$ 's oppositely charged antinodes would change the direction of  $\mathbf{E}_{k}^{LO}$  locally. This would reduce the magnitude of its component along  $\hat{\mathbf{k}}$ , thereby reducing  $\omega_k^{\text{LO}}$ .

An LA perturbation, on the other hand, would modulate the magnitude of  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  along  $\hat{\mathbf{k}}$ . If, at a given moment, the positive antinodes of  $\rho_{b,k}^{\text{LO}}$  were compressed (expanded) by an LA wave of wave vector **k**, its negative antinodes would be expanded (compressed) at that moment. However the phase velocities of LO and acoustic waves are different, in general, which means that the antinodes of an acoustic wave would be moving relative to those of  $\rho_{b,k}^{\text{LO}}$ , making it a time-dependent perturbation of the LO wave.

A third mechanism is that a quasistatic acoustic perturbation of the crystal would create regions in which the crystal is compressed and regions in which it is expanded. Compressing an ionic crystal increases its Madelung energy and the magnitudes of local electric fields. In MgO, this results in the electronic band gap widening under pressure [\[84\]](#page-23-0). It increases the electrostatic potential at oxygen sites, thereby lowering the energy of the highest-energy valence electrons. It has also been shown that  $\omega_k^{\text{LO}}$  increases under pressure [\[34\]](#page-22-0), which may be a result of all fields increasing in magnitude when interionic distances are shortened. Therefore an acoustic wave with a wavelength much larger than  $\lambda_k^{\text{LO}}$  might increase  $\omega_k^{\text{LO}}$ in compressed regions of the crystal and reduce it in expanded regions.

The acoustic-LO coupling mechanisms discussed above suggest that, as soon as acoustic modes become active, we should see a range of LO frequencies corresponding to LO vibrations occurring in the presence of different backgrounds of acoustic deformations of the crystal. Let us now examine the spectra presented in Fig. [1](#page-1-0) for consistency with this prediction. We will focus on the lowest-*T* spectra and on what happens near  $\Gamma$ , which is the region of the BZ occupied by the first modes to become thermally active as the crystal is heated from the  $T \to 0$  limit. To see clearly how the degradation of the LO band depends on wave vector magnitude, let us focus again on the paths  $\Gamma \to X$  and  $\Gamma \to L$  at the left hand side and at the right-hand side, respectively, of each spectrum in Fig. [1.](#page-1-0)

As *T* increases, the earliest deviations from a perfect band structure occur close to  $\Gamma$  along both paths (and also along other paths, which we do not discuss). For example, at 300 K and 500 K the LO modes at  $\frac{1}{5}$ **k**<sub>*X*</sub> and  $\frac{1}{5}$ **k**<sub>*L*</sub> have broadened noticeably in frequency. Along  $\Gamma \rightarrow X$  the LO broadening is accompanied by a noticeable broadening of the LA modes, whereas along  $\Gamma \rightarrow L$  it is accompanied by a noticeable broadening of the TA modes. In the latter case the broadening of the TA mode is smaller at  $\mathbf{k} = \frac{1}{5} \mathbf{k}_L$  than it is further away from  $\Gamma$ , despite the largest broadening of the LO mode being at  $\mathbf{k} = \frac{1}{5} \mathbf{k}_L$ . However,  $\frac{1}{5} \mathbf{k}_L$  is the wave vector at which the difference between LO and TA frequencies is greatest, which means that it is the wave vector at which the adiabatic decoupling of acoustic modes from optical modes discussed in Sec. [VB2](#page-11-0) would be most effective. Therefore the low-*T* broadening of the LO modes near  $\Gamma$ , its progression to larger wave vectors as *T* increases, and the broadening of acoustic modes that accompanies it, all appear to be consistent with acoustic modes changing  $\omega_k^{\text{LO}}$  and making LO waves less coherent by warping **E**LO **<sup>k</sup>** .

The main point of this section  $(VB3)$  is that there exists a mechanism for strong coupling between the acoustic and LO modes of an ionic crystal, which would degrade its LO bands but degrade its acoustic bands to a much lesser degree. We have provided simple physical reasons why we *should* see anomalous broadening of the LO mode in our spectra and, as we analyze the spectra of individual modes in more detail in Sec. [V D,](#page-15-0) we will provide further indirect evidence that the mechanism we proposed is responsible for the observed

<span id="page-14-0"></span>

FIG. 3. Distribution of kinetic energy in reciprocal spacetime when thermal expansion is suppressed by performing simulations at the  $T \rightarrow 0$  volume. The normalization of the data and the color scale of each plot are the same as those used for each plot in Fig. [1,](#page-1-0) As in Fig. 1, the red and green triangles are phonon frequencies at zero temperature and at 300 K within the QHA, respectively. The temperatures are approximate.

broadening. We end this section by summarizing our physical reasoning.

It is well known  $[1,3,82]$  $[1,3,82]$  that the frequency of a long wavelength LO phonon is increased by its intrinsic electric field,  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$ . It is also known that, as *T* increases from the  $T \rightarrow 0$  limit, the earliest contributions to thermal disorder come from the lowest energy modes, which are LWA modes. It is obvious that, by breaking the long-range order of primitive cell displacements along LO cell eigenvectors, acoustic perturbations of the crystal would warp  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$ . It follows that acoustic phonons would change the frequencies of LO waves and, by disordering the lattice, make them less coherent.

As *T* increases, the first manifestation of this effect in the spectra would be the LO band near the BZ center beginning to broaden and melt. As *T* continues to increase, causing significant activity among higher-energy and larger-wave vector acoustic modes, we should expect to see the melting of the LO band spread from BZ's center towards its boundary. All of this is consistent with what is observed in Fig. [1.](#page-1-0)

# **C. Effects of thermal expansion**

To disentangle the effects of thermal expansion from other mechanisms by which *T* changes the spectra, let us turn our attention to Fig. 3, which presents the spectra calculated from MD simulations in which the volume of the crystal at all values of *T* was constrained at the value  $V(0) \equiv \lim_{T \to 0} V(T)$ . We will refer to the MD simulations whose spectra are pre-sented in Figs. [1](#page-1-0) and 3 as the  $V(T)$  and  $V(0)$  simulations, respectively.

The comparison between Figs. [1](#page-1-0) and 3 demonstrates that the overall softening of most modes that occurs when the crystal is free to thermally expand does not occur when it is prevented from expanding. This, and the comparison with the QHA results below, confirm that thermal expansion bears most of the responsibilty for this softening.

The other features of the spectra that we have chosen to discuss, namely, that melting of the band structure begins with the LO mode near  $\Gamma$ , before spreading to the BZ boundary and to other modes, are present in both Figs. [1](#page-1-0) and 3. Therefore we can rule thermal expansion out as their cause. However it is worth noting that, at very high *T* , the TO branches look much more like bands in the  $V(0)$  spectra than they do in the  $V(T)$  spectra. This may be because thermal expansion causes the TO and TA branches of the spectrum to overlap more in frequency, which would strengthen TO-TA interactions.

The QHA provides a reasonably accurate approximation to the phonon band structure up to at least 500 K, albeit one that is not noticeably closer to the MD band structure than the HA. However, its overestimation of the softening of TO frequencies is quite dramatic at  $T = 1000$  K and 2000 K. Referring to Table [I,](#page-9-0) we see that the QHA lattice parameter is larger than the MD lattice parameter at each *T* . However, these differences are much too small to explain the differences in frequency between the QHA bands and the *V* (*T* ) MD bands. When the *T* -dependent frequency of each mode is plotted as a function of the *T* -dependent volume for both sets of calculations, the QHA curves lie below the *V* (*T* ) MD curves and the gap between them widens as *T* increases. Thermal expansion must explain the softening of the QHA bands with increasing

<span id="page-15-0"></span>

FIG. 4. Isobaric heat capacity versus temperature. Symbols denote the measured values of Isaak *et al.* [\[85\]](#page-23-0), the orange line is the result of the DFPT QHA calculations by Karki *et al.* [\[14\]](#page-21-0) and the blue line is the result of our calculations within the QPA.

*T* , because the difference in *V* is the only difference between calculations performed at different *T* . However the degree to which the QHA oversoftens bands cannot be explained fully by inaccuracies of QHA thermal expansivities. We will return to this issue in Sec. V D.

Karki *et al.* [\[14\]](#page-21-0) found that QHA calculations of several thermodynamic properties started to deviate significantly from experimental measurements at around 1000 K, with both thermal expansivity and the isobaric heat capacity  $C^{(P)}$  being overestimated at high *T* and the degrees to which they were overestimated increasing with *T*. Therefore, to investigate whether or not this is a failure of the quasi-independent phonon assumption we have calculated the heat capacity within the QPA [\[1\]](#page-21-0). For each mode  $\mathbf{k}\mu$  we fitted the mode spectrum,  $f_{\mathbf{k}\mu}(\omega)$ , with a Lorentzian [Eq. [\(8\)](#page-7-0)], and treated the peak frequencies of the fitted Lorentzians as the QP frequencies.

The comparison between our QPA calculation, the QHA calculations of Karki *et al.*, and the measurements of Isaak, Anderson, and Goto are presented in Fig. 4.

The QPA heat capacities are in excellent agreement with experiment at all values of *T* for which experimental data is available. Therefore the inaccuracy of the QHA calculation at high *T* is unlikely to be a result of treating phonons as quasi-independent excitations. It must be caused by the QHA failing to account for the *self-energies* of these phonons. It is their self-energies that turn them from NMVs into phonon QPs [\[1,5\]](#page-21-0).

#### **D. Thermal broadening of bands**

In Sec. V<sub>B</sub>, we suggested some mechanisms by which LO bands can be broadened and shifted in frequency by acoustic modes, without acoustic bands degrading to the same degree. In Sec.  $VC$ , we showed that thermal expansion causes most bands to shift to lower frequencies as *T* increases, but is not the cause of the anomalous broadening of the LO band. In this section we take a closer look at thermal broadening of bands.

Figure 5 presents the results of fitting Lorentzian functions [Eq.  $(8)$ ] to individual mode spectra [Eq.  $(3)$ ]. The upper panels are plots of the *T* -induced frequency shifts [Eq. [\(7\)](#page-7-0)],



FIG. 5. Results of fitting Lorentzian functions to optical mode spectra at, from left to right, the wave vectors  $\frac{1}{5}$ **k**<sub>*X*</sub>, **k**<sub>*X*</sub>,  $\frac{1}{5}$ **k**<sub>*L*</sub>, and **k**<sub>*L*</sub>, where **k**<sub>*X*</sub>  $\equiv \frac{1}{2}(\mathbf{b}^1 + \mathbf{b}^3)$ , **k**<sub>*L*</sub>  $\equiv \frac{1}{2}(\mathbf{b}^1 + \mathbf{b}^2 + \mathbf{b}^3)$ , and { $\mathbf{b}^1$ ,  $\mathbf{b}^2$ ,  $\mathbf{b}^3$ } is the set of primitive reciprocal lattice vectors. The upper panels plot the difference between the fitted Lorentzian's peak frequency and the  $T \rightarrow 0$  phonon frequency vs *T*. The lower panels plot the fitted Lorentzians' full widths at half maximum vs *T* and the left-most plot includes a comparison to the values measured by Calandrini *et al.* at a much smaller wave vector  $[10]$ . The insets specify the meanings of the symbols and their colors in all eight plots.

 $\Delta \omega_{\mathbf{k}\mu}(T) \equiv \bar{\omega}_{\mathbf{k}\mu}(T) - \bar{\omega}_{\mathbf{k}\mu}(0)$ , of the three optical modes at each of the four wave vectors  $\frac{1}{5}$ **k**<sub>*X*</sub>, **k**<sub>*X*</sub>,  $\frac{1}{5}$ **k**<sub>*L*</sub>, and **k**<sub>*L*</sub>, where  $\bar{\omega}_{k\mu}(0)$  is the frequency of mode  $k\mu$  at zero pressure in the  $T \to 0$  limit. In our MD simulations,  $\bar{\omega}_{\mathbf{k}\mu}(0)$  is the frequency of normal mode  $\mathbf{k}\mu$  at the *V* that minimizes the potential energy, whereas in the QHA it is its frequency at the *V* that minimizes the sum of the potential energy and the zero-point energy of the phonon modes. Results from our  $V(T)$  and *V* (0) MD simulations are plotted in blue and red, respectively, and our QHA results are plotted in green. In each case the frequency shifts of TO and LO modes are plotted as circles and filled squares, respectively. We use the same symbols and colors to plot the Lorentzian linewidths  $\gamma_{k\mu}(T)$  in the lower panels, but instead of comparing to QHA linewidths, which we did not calculate, we compare to infrared reflectivity data of Calandrini *et al.* [\[10\]](#page-21-0). We find good agreement, despite comparing calculations at  $\mathbf{k} = \frac{1}{5} \mathbf{k}_X$  to measurements at much smaller wave vectors.

Overall, it is less easy to see firm trends in the data pre-sented in Fig. 5 than it is to see them in Fig. [6,](#page-16-0) which uses the same symbols and colors to plot the shifts in the mean,  $\Delta \omega_{\mathbf{k}\mu}(T)$ , and the entropies,  $S_{\mathbf{k}\mu}(T)$ , of the mode distributions,  $f_{\mathbf{k}\mu}(\omega)$ . Therefore we include Fig. 5 for two reasons: to compare our MD linewidths with measured linewidths, and to illustrate why, for the purpose of investigating the effects of increasing *T* , we have chosen to *not* assume that mode spectra are approximately Lorentzian. The entropy of a Lorentzian distribution is  $\sim \ln \gamma_{k\mu}$ , so Fig. 5(b) might be clearer if its vertical axis had a logarithmic scale, but the entropy is a more general measure of uncertainty than either the width of the best-fit Lorentzian or its logarithm, and using it does not entail making any physical assumptions.

<span id="page-16-0"></span>

FIG. 6. (Top) Difference between the mean of each optical mode's spectrum and the mode's  $T \to 0$  frequency as a function of *T* in our *V* (0) MD simulations, our *V* (*T* ) MD simulations, and our QHA calculations. (Bottom) Entropies of the optical mode spectra vs *T* . The insets specify the meanings of the symbols and their colors in all eight plots.

The QHA's overestimation of the rates of decrease of optical mode frequencies with respect to *T* is very clear in Fig. 6. Figure 7 demonstrates that the frequency shifts from our  $V(T)$  simulations are not well approximated by adding the shift caused by increasing  $T$  at a fixed  $V$  of  $V(0)$  to the shift caused by changing *V* at a fixed *T* of zero. The former is positive, which is consistent with the fact that phonon frequencies increase under pressure [\[14](#page-21-0)[,34\]](#page-22-0): as quantified in Sec. [IV B,](#page-9-0) when *V* is fixed, *P* increases as *T* increases. Increasing *P* makes it more difficult for individual primitive cells of the crystal to strain in order to accommodate optical mode displacements. This elastic response would reduce the



FIG. 7. LO and TO frequency shifts as a function of *T* at all finite wave vectors in our simulations between  $\Gamma$  and *X*. The insets specify the meanings of the symbols and their colors, which are the same as in Fig. 6 apart from the inclusion of the sums of the  $V(0)$  MD shifts and the QHA shifts. These are plotted in orange to demonstrate that they differ from the  $V(T)$  MD shifts.



FIG. 8. (Top) Difference between the mean of each acoustic mode's spectrum and the mode's  $T \to 0$  frequency as a function of *T* . (Bottom) Entropies of the acoustic mode spectra vs *T* . The insets specify the meanings of the symbols and their colors in all eight plots.

energy cost of the optical displacements; therefore optical frequencies are increased when this response mechanism is suppressed. The orange symbols in Fig. 7 would coincide with their blue counterparts if changes to *T* and *V* contributed frequency shifts whose origins were approximately independent. The fact that they do not suggests that *T* has other important effects. This highlights, again, the importance of interactions between phonons and the fact that each mode spectrum is not the spectrum of bare NMVs, but of QPs.

Figure 8 shows that TA modes are softened much less by *T* than optical modes, especially at small wave vectors. At  $\mathbf{k} = \frac{1}{5}\mathbf{k}_X$  the QHA shifts are negligible and the *V*(*T*) and *V*(0) shifts are almost equal. At **k** =  $\frac{1}{5}$ **k**<sub>*L*</sub>, by contrast, the  $V(0)$  shifts are negligible and the  $V(T)$  and QHA shifts are almost equal. At all wave vectors except  $\mathbf{k} = \frac{1}{5} \mathbf{k}_L$ , the QHA overestimates the softening of the LA mode at 1000 K and 2000 K. There is undoubtedly much to learn from a more detailed analysis of the acoustic modes than we have performed. The nonmonotonicity of TA modes at low *T* is particularly intriguing.

Notice that, at  $\mathbf{k}_X$  and  $\mathbf{k}_L$ , where acoustic frequencies are as large as optical frequencies, the entropies of acoustic and optical modes have similar magnitudes. However, near  $\Gamma$  the entropies of acoustic modes, and particularly the TA modes, are much smaller than those of optical modes. Near  $\Gamma$  is where acoustic frequencies are small enough that adiabatic decoupling is likely to provide acoustic modes with the most protection from disorder generated by the optical modes.

Also notice that, at all values of  $T$ , the LA mode's entropy is significantly greater than the TA mode's entropy at  $\mathbf{k} = \frac{1}{5}\mathbf{k}_X$ ; whereas at  $\mathbf{k} = \frac{1}{5}\mathbf{k}_L$  it is the TA mode whose entropy is larger at low *T*; and at  $T \ge 1000$  K the LA and TA modes have similar entropies. Entropy quantifies uncertainty, which increases with a mode distribution's width. Therefore Fig. 8 quantifies our observation in Sec. [VB3](#page-12-0) that, at low *T* , there is broadening of the LA band along  $\Gamma \rightarrow X$  and of the TA bands along  $\Gamma \rightarrow L$ .

<span id="page-17-0"></span>

FIG. 9. (Bottom) Degree to which each mode is an optical mode at  $T = 0$  as a function of **k** along wave vector paths  $\Gamma \rightarrow X$  (left) and  $\Gamma \rightarrow L$  (right). (Top) Characteristics of the normalized mode spectrum  $f_{\mathbf{k}\mu}(\omega)$  [Eq. [\(3\)](#page-5-0)] as a function of wave vector **k** at  $T = 500$  K. The four plotted characteristics of  $f_{\mathbf{k}\mu}$  are, from top to second-frombottom, the mean  $\bar{\omega}_{\mathbf{k}\mu}/2\pi$ , the entropy  $S_{\mathbf{k}\mu}$  [Eq. [\(9\)](#page-7-0)], the standard deviation  $\sigma$ , and the *skewness*, where  $\sigma^2$  and the skewness are, respectively, the second  $(m = 2)$  and third  $(m = 3)$  central moments,  $(2\pi)^{-m} \sum_{\omega} (\omega - \bar{\omega}_{\mathbf{k}\mu})^m f_{\mathbf{k}\mu}(\omega).$ 

Figure 9 provides more insight into how the mode spectra change between the center and the boundary of the BZ. It consists of five vertically stacked pairs of plots, with the horizontal axes of each pair's left-hand and right-hand plots specifying one of the five finite wave vectors in our simulations along  $\Gamma \to X$  and  $\Gamma \to L$ , respectively. The data plotted in the lowermost pair of plots applies to the  $T \rightarrow 0$  limit, because it was calculated from normal mode eigenvectors. We will discuss it below. The four uppermost pairs are plots of characteristics of the mode spectra from  $V(T)$  MD simulations at  $T = 500$  K. The uppermost plots are simply plots of the mean frequencies; i.e., the bands. The next panels down are plots of the modes' entropies,  $S_{\mathbf{k}\mu}$  [Eq. [\(9\)](#page-7-0)]. The third panels down are plots of the standard deviations (σ) of the mode spectra, which we include only to allow σ to be compared to *S***<sup>k</sup>**<sup>μ</sup> as a measure of thermal disorder. The plots second from the bottom show that most of the mode spectra are skewed towards higher frequencies, which would be consistent with them leaving behind a highfrequency tail when thermal expansion shifts their means downwards.

We will not speculate further about why most mode spectra have positive skewnesses. However in Sec.  $VB3$  we suggested a mechanism that might contribute negatively to the skewness of only one of the six modes. An acoustic perturbation that warps  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  is likely to reduce its magnitude, thereby lowering  $\omega_k^{\text{LO}}$ . Therefore a thermal distribution of acoustic perturbations might contribute low-frequency tails

to LO mode spectra. This might explain why the LO mode has a skewness that is smaller in magnitude than other modes when it is positive, why it is the only mode whose skewness is negative at some wave vectors, and why the skewness changes from negative to positive between the BZ center and its boundary.

Now let us examine the plots of entropy in Fig. 9. Notice that, starting at  $\mathbf{k} = \frac{1}{5}\mathbf{k}_X$  or  $\mathbf{k} = \frac{1}{5}\mathbf{k}_L$ , the acoustic bands' entropies increase, initially, as |**k**| increases. This is to be expected because each mode's entropy vanishes in the  $T \rightarrow 0$  limit and it is only the interactions between modes that makes it finite. Acoustic modes' frequencies move closer to those of optical modes as their wave vectors move away from the BZ center. Therefore we should expect their interactions with optical modes to strengthen as |**k**| increases, and this would increase their entropies.

If a mode was damped by many weak interactions, its entropy would increase gradually and monotonically as either the number of interactions or their strengths increased gradually. However, sudden reductions of entropy occur to the LA mode and the TA modes along  $\Gamma \rightarrow X$ ; and there is a less dramatic reduction of the TA modes' entropies along  $\Gamma \rightarrow L$ . After increasing monotonically between  $\mathbf{k} = \frac{1}{5} \mathbf{k}_X$ and  $\mathbf{k} = \frac{4}{5}\mathbf{k}_X$ , the entropies of the LA band and the higherentropy TA mode drop abruptly at  $\mathbf{k} = \mathbf{k}_X$ . In the right-hand plot the entropies of both TA bands suddenly begin to decrease at  $\mathbf{k} = \frac{4}{5} \mathbf{k}_L$ .

These kinks in the entropy as a function of **k** are interesting because they suggest the existence of strong interactions between modes, which contribute sizeable fractions of their total entropies. If a mode's entropy only had contributions from weak interactions, a sudden decrease would require many interactions to weaken or vanish simultaneously (at the same wave vector). This is improbable. Each kink is much more likely to be explained by the disappearance or weakening of few very strong interactions, than by many weak ones. Therefore a kink in a band's entropy is likely to mean that, on the higher-entropy side of the kink, the mode is involved in at least one particularly strong interaction.

The fact that the entropy of the LA mode at  $\mathbf{k} = \frac{1}{5} \mathbf{k}_X$  is so high adds to the suspicion that it is caused by coupling to the LO mode, because  $\omega_k^{\text{LA}}$  is so low that only TA modes have similar frequencies. However, the TA modes' entropies are smaller, suggesting that they do not interact with other modes as strongly as the LA mode does. Furthermore, if the high entropy of the LA band was caused by almost-resonant coupling between LA and TA modes, it should decrease with increasing  $|\mathbf{k}|$ , because  $\omega_{\mathbf{k}}^{\text{LA}}$  and  $\omega_{\mathbf{k}}^{\text{TA}}$  rapidly diverge. For a similar reason, the fact that the TO modes' entropies remain relatively constant as  $\omega_k^{\text{LA}}$  approaches  $\omega_k^{\text{TO}}$  makes it unlikely that the LA entropy has a large contribution from LA-TO coupling. The LA-TO interaction should strengthen as their frequencies get closer, which would increase the TO entropy.

However, if the LA entropy was high due to LA-LO coupling, we would not necessarily see the LO mode entropy increase as  $\omega_k^{\text{LA}}$  increased. If the high LO entropy was caused by the LA mode warping  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$ , the effects of this disruption on the LO band would not necessarily increase as  $\omega_{\mathbf{k}}^{\text{LA}}$  drew

closer to  $\omega_k^{\text{LO}}$ , because the strength of this coupling mechanism does not rely on the coupled modes having similar frequencies. Furthermore, the degree to which  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  shifts  $\omega_{\mathbf{k}}^{\text{LO}}$ upwards decreases with increasing wave vector. Therefore, if the high LO entropy was caused by warping of  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$ , it should decrease with increasing |**k**|, as it can be seen to do in Fig. [9.](#page-17-0) This diminishing importance of  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  as the BZ boundary is approached would also contribute negatively to the LA band's entropy, but the increase of the LA band's frequency means that its interactions with most optical modes strengthen with increasing wave vector as it gets closer to resonance with them. This positive contribution to its entropy might more than offset the entropy reduction caused by its interaction with the LO mode weakening. A further clue that the high entropy of the LA mode is caused by its interaction with the LO mode via warping of  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  is that  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  vanishes at *X*, which is where the LA entropy suddenly drops. The value it drops to is similar to that of the blue TA mode, which increases monotonically between  $\Gamma$  and *X*.

An obvious question to address is why the entropy of one of the TA bands increases more than the other, before also dropping suddenly at  $\mathbf{k} = \mathbf{k}_X$ . The entropy of this band increases much more rapidly than the entropy of the other TA band, despite their mean frequencies remaining equal. However, it is important to note that the eigenvectors of two degenerate modes can be rotated in the two-dimensional space that they span, without changing their frequencies. For this reason, the two TA bands do not have independent identities and, for some rotations of TA eigenvectors within the subspace that they span, the entropies of the bands would be equal. Therefore we should not refer to entropies of the two TA bands individually; we should say that the combined entropy of the TA modes increases with increasing wave vector along  $\Gamma \rightarrow X$ , before suddenly decreasing at  $\mathbf{k} = \mathbf{k}_x$ . This means that the results plotted in the left-hand panels of Fig. [9](#page-17-0) are consistent with acoustic warping of  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$ , which mostly involves the LA band near  $\Gamma$ , but with the TA modes becoming more involved as the BZ boundary is approached.

The sudden decrease of the TA entropy at  $\mathbf{k} = \mathbf{k}_X$  is difficult to explain if its increase between  $\mathbf{k} = \frac{1}{5} \mathbf{k}_X$  and  $\mathbf{k} = \frac{4}{5} \mathbf{k}_X$ is caused by coupling to the TO modes. The data plotted in the bottom left panel of Fig. [9](#page-17-0) also makes TO-TA coupling less likely to be responsible for the TA entropy being nonmonotonic. This is a plot of the degree to which each mode is an optical mode. An optical mode modulates the cation-anion displacement, whereas an acoustic mode modulates their center of mass. Therefore, instead of expressing eigenvectors in terms of displacements of Mg and O ions, we can express them in terms of Mg-O relative displacements and center of mass displacements. The former vanishes for an acoustic mode, the latter vanishes for an optical mode. Therefore we use 100 times the magnitude of latter divided by the magnitude of their sum to quantify the degree to which each mode is an optical mode. What we find is that, at  $\mathbf{k} = \frac{4}{5} \mathbf{k}_X$ , the LA and LO modes become significantly optical and acoustic, respectively. In other words, they mix strongly to become acoustic/optical hybrids.

At  $\mathbf{k} = \mathbf{k}_X$ , the polarization vectors of the LA and LO modes are identical, by symmetry. Mathematically, the differ-

ence in their frequencies can be viewed as a consequence of their normalized eigenvectors being different scalar multiples of the same polarization vector. The LO and LA eigenvectors are the polarization vector multiplied by the square roots of the O and Mg masses, respectively.

It is well known that when two modes couple strongly enough, they tend to hybridize: each one's eigenvector becomes a mix of both eigenvectors, such that the modes described by the hybrid eigenvectors interact much more weakly. This is what causes avoided crossings in band structures. The LO-LA hybridization does not tell us anything directly about TO-TA coupling, but the bottom left plot of Fig. [9](#page-17-0) shows that the TO and TA modes do not hybridize noticeably. At  $\mathbf{k} = \frac{3}{5}\mathbf{k}_X$  each TA (TO) mode is only 1.4% optical (acoustic), and this increases to only 3.2% at *X*. Therefore the sudden decrease of the TA modes' entropy at  $\mathbf{k} = \mathbf{k}_X$  is unlikely to be a consequence of hybridization and we must look elsewhere for an explanation of it. The involvement of the TA modes in the acoustic-LO coupling is an obvious alternative explanation.

Now let us turn our attention to the path  $\Gamma \rightarrow L$ . The bottom-right plot of Fig. [9](#page-17-0) shows that the TO and TA modes begin to hybridize at  $\mathbf{k} = \frac{4}{5} \mathbf{k}_L$ , and that all modes become hybrids of optical and acoustic modes at  $\mathbf{k} = \mathbf{k}_L$ , with each mode's optical/acoustic ratio being approximately 60/40 or 40/60. Therefore the nonmonotonic behavior of the TA entropy might be explained by its sudden decoupling from the TO mode when they transform into TO/TA hybrids. This would make sense because they are both transverse modes, which become very close in frequency near  $\mathbf{k} = \mathbf{k}_L$ . Therefore it seems plausible to interpret the kink in the entropy as a signature of strong TO-TA coupling at  $\mathbf{k} < \frac{4}{5} \mathbf{k}_L$ , which would explain why the entropy of the TA mode is greater than that of the LA mode near  $\Gamma$ .

There are also some strong hints that acoustic modes are disrupting  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  along  $\Gamma \to L$ . Firstly, the LO mode has the highest entropy, despite being the least active mode at low *T* . Secondly, the LO mode has a negative skewness until  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  vanishes at  $\mathbf{k} = \mathbf{k}_L$ , which is consistent with the acoustic warping of **E**LO **<sup>k</sup>** giving it a low-frequency tail. Thirdly, the variation of the skewness with **k** appears to mirror the variation of the mean frequency: As one increases the other decreases. This is consistent with the negative LO skewness being caused by this mechanism because it is the increasing strength of  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$  that causes  $\omega_{\mathbf{k}}^{\text{LO}}$  to increase between *L* and  $\Gamma$ . As the field's contribution to  $\omega_k^{LO}$  increases in magnitude,  $\omega_{\mathbf{k}}^{\text{LO}}$  becomes more vulnerable to being lowered by acoustic perturbations.

All of the acoustic-LO coupling might be LA-LO coupling, but it is also possible that the TA mode is involved. However, near the BZ center, strong TA-LO coupling via TA disruption of  $\mathbf{E}_{k}^{\text{LO}}$  requires the lateral extent of the LO phonon to be not much greater than its wavelength. This is because, if we approximately describe  $\rho_{b,k}^{\text{LO}}$  as a stack of parallel *uniformly*charged infinite planes, a transverse displacement of each plane would not alter the constant field emanating from it. At finite *T* the region perturbed by an LO phonon in a plane perpendicular to **k** has a finite area. Therefore transverse displacements would change  $E_k^{\text{LO}}$  near the boundary of this region and, if a large enough fraction of the atoms participating

<span id="page-19-0"></span>in the LO motion were affected by this change, a TA phonon could couple significantly to it by this mechanism.

When the LO wavelength is small, on the other hand, the crystal planes displaced laterally by a TA phonon cannot be treated as uniformly charged. Their atomistic structures have a significant influence on the value of  $\mathbf{E}_{\mathbf{k}}^{\text{LO}}$ . Therefore, near the BZ boundary, there could be strong TA-LO coupling for LO phonons whose lateral extents are much larger than their wavelengths.

## **VI. CONCLUSIONS**

In this work, we have studied strong phononic correlation in crystalline MgO with a method of calculating vibrational spectra that does not suffer from the limitations of perturbation theories or mean-field theories, and which can produce accurate wave vector-resolved spectra with enough detail to observe band structures. These calculations, and recent similar calculations by Lahnsteiner and Bokhdam [\[33\]](#page-22-0), have been made possible by force fields whose parameters are fit to vast quantities of data calculated *ab initio* [\[22,26,27,32,](#page-22-0)[86\]](#page-23-0), and whose low computational expense, relative to *ab initio* MD, makes calculations of *high resolution* spectra possible.

[A](#page-20-0)s shown in Appendix A and Ref.  $[48]$ , the method we have used to calculate spectra from our MD simulations is much more powerful and generally applicable than existing derivations of it suggest [\[7](#page-21-0)[,24,40–47\]](#page-22-0). It is not a new method, having been used to calculate frequency- and wave vector-resolved spectra at least as far back as 2006 [\[6\]](#page-21-0), and to calculate spectra as functions of frequency only for many years prior to that. However existing derivations of it do not justify using it to calculate spectra at very high  $T$ , when phononic correlation is strong, as we have done. Therefore it has usually been interpreted as a method of calculating the VDOS at thermal equilibrium when phonons do not interact (i.e., in the  $T \rightarrow 0$  limit) or when their interactions are sufficiently simple that each phonon is an exponentially decaying harmonic wave, in which case its spectral peak is Lorentzian [\[24,33,43,](#page-22-0)[87\]](#page-23-0).

However, we have shown that it is an exact method, whose applicability is not restricted to crystals, to low  $T$ , or even to states of thermal equilibrium. This means that it is a powerful tool for studying vibrations in materials, which complements and is complemented by, perturbation theories. It can also be used to assess the accuracies of perturbative methods, meanfield methods, and other methods that rely on simplifying assumptions, such as the assumption of a state of thermal equilbrium [\[20\]](#page-22-0). For these reasons, it seems likely to play an increasingly important role as it becomes possible to calculate forces accurately and efficiently for more materials.

For example, it should help with the development of materials that are either transparent or opaque to THz radiation in a particular frequency window. In other words, it should help to guide us in the rational design of materials that behave as THz bandpass filters for use as active or passive components of THz devices.

Another application is in the study of order-disorder transitions between crystalline phases, which occur when a single phonon band melts but the other bands do not, causing displacements and velocities along the eigenvectors of the melted band's modes to become disordered. This critical weakening of correlations causes both the time average and the spatial average of the displacements to vanish. As a result, when the crystal is observed at either a low spatial resolution or a low temporal resolution, its symmetry appears to have increased. The ability to see how the vibrational spectrum changes as *T* crosses the transition temperature may lead to step changes in our understandings of these transitions, or even to a step change in our understanding of this class of transitions.

A third application is in the study of materials at very high *T*, such as thermal barrier coatings, where there are high densities of defects, and therefore high densities of vibrations that are localized in spacetime and delocalized in reciprocal spacetime. A fourth application, which is related to the third, is in the identification or design of materials with reduced or enhanced thermal conductivities. For example, the thermal conductivities of materials with low-frequency localized *rattler* modes [\[88\]](#page-23-0) are low because the energy of these modes, being localized in spacetime, is delocalized in  $(\mathbf{k}, \omega)$  space. Therefore they overlap with acoustic modes in reciprocal spacetime, which allows the acoustic modes to couple with them, and with one another through them, to disperse their energy.

However the generality of this method means that one of the most important applications of it may be to study general properties of strong phononic correlation. The simplest and most obvious first step is to study the strengthening of phononic correlation in a simple prototype crystal as it is heated. This has been the purpose of this work.

We have found that, despite the apparent banality of crystalline MgO at thermal equilibrium, there is so much to be learned from the spectra presented herein that we have been forced to focus most of our analyses of them on a few of their gross features. We provide our raw  $V(T)$  spectra, in the form of tabulations of  $\hat{\mathcal{E}}^{\hat{\mathcal{K}}}(\mathbf{k},\omega)$  at each value of T, as supplementary material in case others wish to analyse it further or differently [\[76\]](#page-23-0).

By calculating spectra at both constant volume and constant pressure, we have been able to show conclusively that some features of the spectrum's *T* dependence, such as the tendency of bands to soften as *T* increases, are consequences of thermal expansion; and that others, such as the rapid degradation of the LO band, are not. We have shed substantial light on the strengths and limitations of the quasiharmonic approximation and we have shown that some of its weaknesses are rectified by the quasiparticle approximation.

Most importantly, we have identified, and discussed in some detail, two physical considerations that are likely to have observable consequences for many or most crystals. The first is that LO phonons are highly sensitive to the amplitudes of acoustic phonons because a large fraction (∼  $\frac{1}{2}$  for MgO) of  $\omega_k^{LO}$  is contributed by the LO mode's intrinsic field, which acoustic phonons warp. This simple mechanism must occur, to some degree, in every partially ionic crystal, because every LO phonon has an intrinsic field that opposes its motion and increases its frequency. The second consideration is that, despite this strong acoustic-LO coupling, long wavelength acoustic phonons are adiabatically decoupled from LO modes: they strongly modulate the energies and frequencies

<span id="page-20-0"></span>of LO modes, while maintaining approximately constant energies and frequencies themselves.

The feature of the spectrum's *T* dependence that led us to these ideas and explanations is very similar to features that have been observed experimentally in the spectra of other ionic crystals since the 1960s [\[58–67\]](#page-23-0). It has also been studied in MgO and similar materials using perturbation theory [\[10](#page-21-0)[,38,39\]](#page-22-0), but it has lacked a simple and intuitive physical explanation until now.

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# **APPENDIX A: DISTRIBUTION OF THE KINETIC ENERGY OF A VIBRATING STRING IN RECIPROCAL SPACETIME**

Consider a taut string, of uniform mass per unit length ρ, whose ends are fixed at *x* = −*L*/2 and *L*/2, and which lies along the *x* axis when it is at equilibrium. Its displacement from equilibrium at spacetime coordinates  $(x, t)$  is denoted by  $u(x, t)$ , and its kinetic energy per unit length is denoted by  $u(x, t)$ , and its kinetic energy per unit length is  $\kappa(x, t) \equiv \frac{1}{2}v(x, t)^2$ , where  $v \equiv \sqrt{\rho dx} \dot{u}$  and  $dx$  is an infinitesimal length.

Let us assume that the string only moves during the time interval  $(-T/2, T/2)$ , or that it is only observed for times  $t \in (-T/2, T/2)$ . In the latter case, we assume that, at each  $x, v(x, t)$  has been tapered smoothly, but arbitrarily rapidly, to zero at times  $-\mathcal{T}/2$  and  $\mathcal{T}/2$ . Similarly,  $v(x, t)$  is both smooth and vanishes for  $x \notin (-L/2, L/2)$ . Therefore, for all of our purposes within this Appendix,  $\int_{-\tau/2}^{\tau/2} dt$  is equivalent to  $\int_{\mathbb{R}} dt$ and  $\int_{-L/2}^{L/2} dx$  is equivalent to  $\int_{\mathbb{R}} dx$ .

The average, over interval  $(-T/2, T/2)$ , of the string's kinetic energy per unit length at *x* is

$$
\langle \kappa \rangle_t(x) = \frac{1}{2\mathcal{T}} \int_{\mathbb{R}} v(x, t)^2 dt \tag{A1}
$$

$$
= \frac{1}{2\mathcal{T}} \int_{\mathbb{R}} \check{v}^*(x, \omega) \check{v}(x, \omega) d\omega, \tag{A2}
$$

where  $\check{v}(x, \omega) \equiv (2\pi)^{-1/2} \int_{\mathbb{R}} v(x, t) e^{-i\omega t} dt$  is the unitary Fourier transform of  $v(x, t)$  with respect to time, and we have used *Parseval's theorem* [\[50\]](#page-22-0) to reach Eq. (A2) from Eq. (A1). Since  $v(x, t)$  is real,  $\check{v}^*(x, \omega) = \check{v}(x, -\omega)$ , which implies that  $\check{v}^*(x, \omega)\check{v}(x, \omega) = \check{v}(x, -\omega)\check{v}(x, \omega) = \check{v}^*(x, -\omega)\check{v}(x, -\omega).$ Therefore  $\frac{1}{2} \int_{\mathbb{R}} d\omega \check{v}^*(x, \omega) \check{v}(x, \omega) = \int_{\mathbb{R}^+} d\omega \check{v}^*(x, \omega) \check{v}(x, \omega)$ . If we integrate  $\langle \kappa \rangle_t(x)$  over all x and use Parseval's theorem again we find that the average kinetic energy of the entire string is

$$
\langle \mathcal{K} \rangle = \frac{1}{\mathcal{T}} \int_{\mathbb{R}^+} d\omega \int_{\mathbb{R}} dk \tilde{v}^*(k, \omega) \tilde{v}(k, \omega),
$$

where  $\tilde{v}(k, \omega)$  is the unitary FT of  $v(x, t)$  with respect to both *x* and *t*, i.e.,

$$
\tilde{v}(k,\omega) = \frac{1}{2\pi} \int_{\mathbb{R}} d\omega e^{-i\omega t} \int_{\mathbb{R}} dk e^{-ikx} v(x,t)
$$

$$
= \left(\frac{1}{2\pi} \int_{\mathbb{R}} d\omega e^{i\omega t} \int_{\mathbb{R}} dk e^{ikx} v(x,t)\right)^{*}
$$

$$
= \tilde{v}^{*}(-k, -\omega).
$$

Therefore the string's average kinetic energy divided by its length is

$$
\frac{\langle K \rangle}{L} = \frac{1}{L\mathcal{T}} \int_{\mathbb{R}^+} d\omega \int_{\mathbb{R}} dk \tilde{v}(-k, -\omega) \tilde{v}(k, \omega).
$$
 (A3)

The VVCF is defined as

$$
C(x, t) \equiv \langle \langle v(x_0, t_0) v(x_0 + x, t_0 + t) \rangle_{x_0} \rangle_{t_0}
$$
  
=  $\frac{1}{L \mathcal{T}} \int_{\mathbb{R}} dt_0 \int_{\mathbb{R}} dx_0 v(x_0, t_0) v(x_0 + x, t_0 + t).$ 

If we express  $v(x_0, t_0)$  and  $v(x_0 + x, t_0 + t)$  in terms of  $\tilde{v}$ , perform the integrals over  $x_0$  and  $t_0$  before the integrals over frequencies and wave vectors, and use the identities  $2\pi \delta(k) = \int_{\mathbb{R}} e^{ikx_0} dx_0$  and  $2\pi \delta(\omega) = \int_{\mathbb{R}} e^{i\omega t_0} dt_0$ , this becomes

$$
C(x,t) = \frac{1}{L\mathcal{T}} \int_{\mathbb{R}} d\omega e^{i\omega t} \int_{\mathbb{R}} dk e^{ikx} \tilde{v}(-k,-\omega)\tilde{v}(k,\omega).
$$

Therefore the FT of  $C(x, t)$  with respect to both x and t is  $\tilde{C}(k,\omega) = (2\pi/L\mathcal{T})\tilde{v}(-k,-\omega)\tilde{v}(k,\omega)$ , and Eq. (A3) becomes

$$
\frac{\langle K \rangle}{L} = \frac{1}{2\pi} \int_{\mathbb{R}^+} d\omega \int_{\mathbb{R}} dk \tilde{C}(k, \omega). \tag{A4}
$$

Now let us assume that the string's velocity is sampled at *N<sub>t</sub>* evenly spaced times between  $-T/2$  and  $T/2$  and at  $N_x$ evenly space points between −*L*/2 and *L*/2. Then the VVCF becomes

$$
C(x,t) = \frac{1}{N_t N_x} \sum_{t_0} \sum_{x_0} v(x_0, t_0) v(x_0 + x, t_0 + t),
$$

where the sums are over the sampled times and positions.

Now, because  $v(x, t)$  vanishes when  $x \notin (-L/2, L/2)$ , the smallest observable wave vector and the smallest observable difference between two wave vectors are both equal to  $\Delta k \equiv \pi / L$ . Similarly, because  $v(x, t)$  vanishes when  $t \notin (-T/2, T/2)$ , the smallest observable frequency and the smallest observable difference between two frequencies are both equal to  $\Delta \omega = \pi / T$ . This implies that there are only  $N_t$  observable frequencies and  $N_x$  observable wave vectors, and the integrals in Eq. (A4) must be approximated as sums over these observable values. If we define  $\hat{C}(k,\omega) \equiv \tilde{C}(k,\omega) \Delta \omega \Delta k / 2\pi$ , we can express the discretized version of Eq.  $(A4)$  as

$$
\frac{\langle \mathcal{K} \rangle}{L} = \sum_{\omega > 0} \sum_{k} \hat{C}(k, \omega),
$$

and the discrete FT of  $C(x, t)$  as

$$
\hat{C}(k,\omega) = \frac{1}{N_t N_x} \sum_t \sum_x C(x,t) e^{-ikx} e^{-i\omega t}.
$$

<span id="page-21-0"></span>Therefore the average kinetic energy of the string per unit of its length is a sum over all sampled frequencies and wave vectors of the discrete FT of the VVCF. The linearity of the FT implies that each term  $\hat{C}(k,\omega)$  in this sum is the contribution to the average kinetic energy of motions with wave vector *k* and frequency  $\omega$ .

# **APPENDIX B: THE FORCE FIELD**

We use a classical dipole-polarizable potential of the form described in Refs. [\[25,28,29\]](#page-22-0). It includes a sum of purely pairwise interaction energies of the form

$$
U_{ij}(r) = \frac{q_i q_j}{r_{ij}} + D_{ij}[e^{\gamma_{ij}[1-(r_{ij}/r_{ij}^0)]} - 2e^{(\gamma_{ij}/2)[1-(r_{ij}/r_{ij}^0)]}],
$$

where *r* is the distance between ions *i* and *j*, and  $q_i$  and  $q_j$ are their charges. The parameters  $D_{ij}$ ,  $\gamma_{ij}$  and  $r_{ij}^0$  define the *Morse potential* and, along with the charges, are among the parameters that are fit to reproduce forces, stress tensors, and energy differences calculated *ab initio* with density functional theory. A list of all potential parameters and their values are provide in Table II.

In addition to the pairwise part of the potential, the oxygen anion is assigned a polarizability,  $\alpha$ , which is used to assign a dipole moment **p***<sup>i</sup>* to each anion as follows:

$$
\mathbf{p}_i = \mathbf{p}_i^{SR} + \alpha \mathbf{E}_i(\{\mathbf{p}_i\})
$$

where the second term on the right-hand side is the dipole moment induced on ion *i* by the local electric field from the

- [1] D. C. Wallace, *Thermodynamics of Crystals* (Dover, New York, 1972).
- [2] W. Jones and N. H. March, *Theoretical Solid State Physics: Volume I* (Dover, New York, 1973).
- [3] N. W. Ashcroft and N. D. Mermin, *Solid State Physics*, HRW International Editions (Holt, Rinehart and Winston, 1976).
- [4] M. L. Cohen and S. G. Louie, *Fundamentals of Condensed Matter Physics* (Cambridge University Press, Cambridge, 2016).
- [5] R. D. Mattuck, *A Guide to Feynman Diagrams in the Manybody Problem*, Dover Books on Physics Series (Dover Publications, Mineola, New York, 1992).
- [6] P. Tangney, M. L. Cohen, and S. G. Louie, Giant wave-drag [enhancement of friction in sliding carbon nanotubes,](https://doi.org/10.1103/PhysRevLett.97.195901) *Phys. Rev.* Lett. **97**, 195901 (2006).
- [7] D.-B. Zhang, T. Sun, and R. M. Wentzcovitch, Phonon quasiparticles and anharmonic free energy in complex systems, Phys. Rev. Lett. **112**[, 058501 \(2014\).](https://doi.org/10.1103/PhysRevLett.112.058501)
- [8] J. Breeze, *Temperature and Frequency Dependence of Complex Permittivity in Metal Oxide Dielectrics: Theory, Modelling and Measurement* (Springer International, Cham, 2016).
- [9] Y. Lu, T. Sun, and D.-B. Zhang, Lattice anharmonicity, phonon dispersion, and thermal conductivity of PbTe studied by the [phonon quasiparticle approach,](https://doi.org/10.1103/PhysRevB.97.174304) Phys. Rev. B **97**, 174304 (2018).
- [10] E. Calandrini, L. Paulatto, D. Antonangeli, F. He, R. P. S. M. Lobo, F. Capitani, J.-B. Brubach, P. Roy, L. Vincent, and P.

TABLE II. Force field parameters in atomic units.

	Mg	O	$Mg-Mg$	$Mg-O$	O-O
charges					
q		1.415382 -1.415382			
D			0	$3.0945 \times 10^{-3}$ $2.161 \times 10^{-3}$	
$\gamma$			25.1073	10.1621	8.9925
$r^0$			26.9788	4.9640	6.1491
dipoles					
$\alpha$	0	9.6565			
$\boldsymbol{b}$			$\Omega$	1.8713	0
$\mathcal{C}$			0	$-1.6809$	0

charges and dipole moments of all other ions, and

$$
\mathbf{p}_i^{SR} = \alpha \sum_{j \neq i} \frac{q_{ij} \mathbf{r}_{ij}}{r_{ij}^3} f_{ij}(r_{ij}),
$$

where

$$
f_{ij}(r_{ij}) = c_{ij} \sum_{k=0}^{4} \frac{(b_{ij}r_{ij})^k}{k!} e^{-b_{ij}r_{ij}}.
$$

is a short-range contribution to the dipole from an anion's electron cloud changing shape as its neighboring ions move. Since each dipole moment depends on the values of all others, the set of all dipole moments is calculated, at each set of ionic positions, by iterating them to self-consistency [\[25\]](#page-22-0). After self-consistency has been achieved a charge-dipole interaction energy is added to  $U_{ij}$  for each pair of ions  $(i, j)$  that includes an oxygen ion, and a dipole-dipole interaction energy is added to it when *i* and *j* are both oxygen ions.

Giura, Limits of the quasiharmonic approximation in MgO: Volume dependence of optical modes investigated by infrared [reflectivity and](https://doi.org/10.1103/PhysRevB.103.054302) *ab initio* calculations, Phys. Rev. B **103**, 054302 (2021).

- [11] [C. Sirtori, Bridge for the terahertz gap,](https://doi.org/10.1038/417132b) Nature (London) **417**, 132 (2002).
- [12] [S. Perkowitz, Navigating the terahertz gap,](https://doi.org/10.1088/2058-7058/33/3/31) Phys. World **33**, 37 (2020).
- [13] M. J. L. Sangster, G. Peckham, and D. H. Saunderson, Lattice [dynamics of magnesium oxide,](https://doi.org/10.1088/0022-3719/3/5/017) J. Phys. C: Solid State Phys. **3**, 1026 (1970).
- [14] B. B. Karki, R. M. Wentzcovitch, S. de Gironcoli, and S. Baroni, High-pressure lattice dynamics and thermoelasticity of MgO, Phys. Rev. B **61**[, 8793 \(2000\).](https://doi.org/10.1103/PhysRevB.61.8793)
- [15] D. J. Hooton, A new treatment of anharmonicity in lattice thermodynamics: I, [London, Edinburgh, Dublin Philos. Mag. J. Sci.](https://doi.org/10.1080/14786440408520575) **46**, 422 (1955).
- [16] N. R. Werthamer, Self-consistent phonon formulation of anharmonic lattice dynamics, [Phys. Rev. B](https://doi.org/10.1103/PhysRevB.1.572) **1**, 572 (1970).
- [17] O. Hellman, P. Steneteg, I. A. Abrikosov, and S. I. Simak, Temperature dependent effective potential method for accurate [free energy calculations of solids,](https://doi.org/10.1103/PhysRevB.87.104111) Phys. Rev. B **87**, 104111 (2013).
- [18] T. Tadano and S. Tsuneyuki, Self-consistent phonon calculations of lattice dynamical properties in cubic  $SrTiO<sub>3</sub>$  with [first-principles anharmonic force constants,](https://doi.org/10.1103/PhysRevB.92.054301) Phys. Rev. B **92**, 054301 (2015).
- <span id="page-22-0"></span>[19] R. Bianco, I. Errea, L. Paulatto, M. Calandra, and F. Mauri, Second-order structural phase transitions, free energy curvature, and temperature-dependent anharmonic phonons in the self-consistent harmonic approximation: Theory and stochastic implementation, Phys. Rev. B **96**[, 014111 \(2017\).](https://doi.org/10.1103/PhysRevB.96.014111)
- [20] L. Monacelli, R. Bianco, M. Cherubini, M. Calandra, I. Errea, and F. Mauri, The stochastic self-consistent harmonic approximation: calculating vibrational properties of materials with full [quantum and anharmonic effects,](https://doi.org/10.1088/1361-648X/ac066b) J. Phys.: Condens. Matter **33**, 363001 (2021).
- [21] M. Zacharias, G. Volonakis, F. Giustino, and J. Even, Anharmonic lattice dynamics via the special displacement method, Phys. Rev. B **108**[, 035155 \(2023\).](https://doi.org/10.1103/PhysRevB.108.035155)
- [22] P. Tangney and S. Scandolo, A many-body interatomic potential [for ionic systems: Application to MgO,](https://doi.org/10.1063/1.1609980) J. Chem. Phys. **119**, 9673 (2003).
- [23] P. Tangney and S. Scandolo, Melting slope of MgO from molec[ular dynamics and density functional theory,](https://doi.org/10.1063/1.3238548) J. Chem. Phys. **131**, 124510 (2009).
- [24] T. Sun, D.-B. Zhang, and R. M. Wentzcovitch, Dynamic stabilization of cubic CaSiO<sub>3</sub> perovskite at high temperatures and pressures from *ab initio* [molecular dynamics,](https://doi.org/10.1103/PhysRevB.89.094109) Phys. Rev. B **89**, 094109 (2014).
- [25] P. Tangney and S. Scandolo, An *ab initio* parametrized interatomic force field for silica, [J. Chem. Phys.](https://doi.org/10.1063/1.1513312) **117**, 8898 (2002).
- [26] J. Behler and M. Parrinello, Generalized neural-network representation of high-dimensional potential-energy surfaces, Phys. Rev. Lett. **98**[, 146401 \(2007\).](https://doi.org/10.1103/PhysRevLett.98.146401)
- [27] A. P. Bartók, M. C. Payne, R. Kondor, and G. Csányi, Gaussian approximation potentials: The accuracy of quantum [mechanics, without the electrons,](https://doi.org/10.1103/PhysRevLett.104.136403) Phys. Rev. Lett. **104**, 136403 (2010).
- [28] X. J. Han, L. Bergqvist, P. H. Dederichs, H. Müller-Krumbhaar, J. K. Christie, S. Scandolo, and P. Tangney, Polarizable interatomic force field for  $TiO<sub>2</sub>$  parametrized using density functional theory, Phys. Rev. B **81**[, 134108 \(2010\).](https://doi.org/10.1103/PhysRevB.81.134108)
- [29] J. Sarsam, M. W. Finnis, and P. Tangney, Atomistic force field [for alumina fit to density functional theory,](https://doi.org/10.1063/1.4832695) J. Chem. Phys. **139**, 204704 (2013).
- [30] W. J. Szlachta, A. P. Bartók, and G. Csányi, Accuracy and transferability of gaussian approximation potential models for tungsten, Phys. Rev. B **90**[, 104108 \(2014\).](https://doi.org/10.1103/PhysRevB.90.104108)
- [31] V. L. Deringer and G. Csányi, Machine learning based in[teratomic potential for amorphous carbon,](https://doi.org/10.1103/PhysRevB.95.094203) Phys. Rev. B **95**, 094203 (2017).
- [32] O. T. Unke, S. Chmiela, H. E. Sauceda, M. Gastegger, I. Poltavsky, K. T. Schütt, A. Tkatchenko, and K.-R. Müller, Machine learning force fields, Chem. Rev. **121**[, 10142 \(2021\).](https://doi.org/10.1021/acs.chemrev.0c01111)
- [33] J. Lahnsteiner and M. Bokdam, Anharmonic lattice dynamics in large thermodynamic ensembles with machine-learning force fields: CsPbBr<sub>3</sub>, a phonon liquid with Cs rattlers, Phys. Rev. B **105**, 024302 (2022).
- [34] S. Ghose, M. Krisch, A. R. Oganov, A. Beraud, A. Bosak, R. Gulve, R. Seelaboyina, H. Yang, and S. K. Saxena, Lattice dynamics of MgO at high pressure: Theory and experiment, Phys. Rev. Lett. **96**[, 035507 \(2006\).](https://doi.org/10.1103/PhysRevLett.96.035507)
- [35] H. Dekura, T. Tsuchiya, and J. Tsuchiya, *Ab initio* lattice thermal conductivity of  $MgSiO<sub>3</sub>$  perovskite as found in earth's lower mantle, Phys. Rev. Lett. **110**[, 025904 \(2013\).](https://doi.org/10.1103/PhysRevLett.110.025904)
- [36] T. Kimura, H. Ohfuji, M. Nishi, and T. Irifune, Melting temperatures of MgO under high pressure by micro-texture analysis, Nat. Commun. **8**[, 15735 \(2017\).](https://doi.org/10.1038/ncomms15735)
- [37] G. Fugallo, B. Rousseau, and M. Lazzeri, Infrared reflectance, transmittance, and emittance spectra of MgO from first principles, Phys. Rev. B **98**[, 184307 \(2018\).](https://doi.org/10.1103/PhysRevB.98.184307)
- [38] P. Giura, L. Paulatto, F. He, R. P. S. M. Lobo, A. Bosak, E. Calandrini, L. Paolasini, and D. Antonangeli, Multiphonon anharmonicity of MgO, Phys. Rev. B **99**[, 220304\(R\) \(2019\).](https://doi.org/10.1103/PhysRevB.99.220304)
- [39] A. Togo, H. Hayashi, T. Tadano, S. Tsutsui, and I. Tanaka, LO-mode phonon of KCl and NaCl at 300 K by inelastic Xray scattering measurements and first principles calculations, [J. Phys.: Condens. Matter](https://doi.org/10.1088/1361-648X/ac7b01) **34**, 365401 (2022).
- [40] M. T. Dove, *Introduction to Lattice Dynamics*, Cambridge Topics in Mineral Physics and Chemistry (Cambridge University Press, Cambridge, 1993).
- [41] C. Lee, D. Vanderbilt, K. Laasonen, R. Car, and M. Parrinello, *Ab initio* studies on the structural and dynamical properties of ice, Phys. Rev. B **47**[, 4863 \(1993\).](https://doi.org/10.1103/PhysRevB.47.4863)
- [42] A. J. C. Ladd, B. Moran, and W. G. Hoover, Lattice thermal conductivity: A comparison of molecular dynamics and anharmonic lattice dynamics, Phys. Rev. B **34**[, 5058 \(1986\).](https://doi.org/10.1103/PhysRevB.34.5058)
- [43] T. Sun, X. Shen, and P. B. Allen, Phonon quasiparticles and anharmonic perturbation theory tested by molecular dynamics on a model system, Phys. Rev. B **82**[, 224304 \(2010\).](https://doi.org/10.1103/PhysRevB.82.224304)
- [44] R. Meyer and D. Comtesse, Vibrational density of states of silicon nanoparticles, Phys. Rev. B **83**[, 014301 \(2011\).](https://doi.org/10.1103/PhysRevB.83.014301)
- [45] M. Thomas, M. Brehm, R. Fligg, P. Vöhringer, and B. Kirchner, Computing vibrational spectra from *ab initio* molecular dynamics, [Phys. Chem. Chem. Phys.](https://doi.org/10.1039/c3cp44302g) **15**, 6608 (2013).
- [46] M. E. Tuckerman, *Statistical Mechanics: Theory and Molecular Simulation*, Oxford Graduate Texts (Oxford University Press, Oxford, 2010).
- [47] A. Cupo, D. Tristant, K. Rego, and V. Meunier, Theoretical analysis of spectral lineshapes from molecular dynamics, [npj Comput. Mater.](https://doi.org/10.1038/s41524-019-0220-1) **5**, 82 (2019).
- [48] P. Tangney, Wave theory of lattice dynamics, [arXiv:2401.02375.](https://arxiv.org/abs/2401.02375)
- [49] R. A. Cowley, The lattice dynamics of an anharmonic crystal, Adv. Phys. **12**[, 421 \(1963\).](https://doi.org/10.1080/00018736300101333)
- [50] R. S. Strichartz, *A Guide To Distribution Theory And Fourier Transforms* (World Scientific, Singapore, 2003).
- [51] W. G. Aulbur, L. Jönsson, and J. W. Wilkins, Quasiparticle calculations in solids, in *Solid State Physics*, edited by H. Ehrenreich and F. Spaepen (Academic Press, San Diego, 2000), Vol. 54, pp. 1–218.
- [52] W. Cochran, Crystal stability and the theory of ferroelectricity, Adv. Phys. **9**[, 387 \(1960\).](https://doi.org/10.1080/00018736000101229)
- [53] E. A. Stern, Character of order-disorder and displacive com[ponents in barium titanate,](https://doi.org/10.1103/PhysRevLett.93.037601) Phys. Rev. Lett. **93**, 037601 (2004).
- [54] J. Hlinka, T. Ostapchuk, D. Nuzhnyy, J. Petzelt, P. Kuzel, C. Kadlec, P. Vanek, I. Ponomareva, and L. Bellaiche, Coexistence of the phonon and relaxation soft modes in the terahertz dielectric response of tetragonal BaTiO<sub>3</sub>, Phys. Rev. Lett. **101**, 167402 (2008).
- [55] F. Gu, E. Murray, and P. Tangney, Carrier-mediated control over the soft mode and ferroelectricity in BaTiO<sub>3</sub>, Phys. Rev. Mater. **5**, 034414 (2021).
- <span id="page-23-0"></span>[56] K. F. Riley, M. P. Hobson, and S. J. Bence, Probability, in *Mathematical Methods for Physics and Engineering: A Comprehensive Guide*, 2nd ed. (Cambridge University Press, Cambridge, UK, 2002), pp. 961–1063.
- [57] [C. E. Shannon, A mathematical theory of communication,](https://doi.org/10.1002/j.1538-7305.1948.tb01338.x) Bell Syst. Tech. J. **27**, 379 (1948).
- [58] A. D. B. Woods, W. Cochran, and B. N. Brockhouse, Lattice dynamics of alkali halide crystals, Phys. Rev. **119**[, 980 \(1960\).](https://doi.org/10.1103/PhysRev.119.980)
- [59] A. D. B. Woods, B. N. Brockhouse, R. A. Cowley, and W. Cochran, Lattice dynamics of alkali halide crystals. II. experimental studies of KBr and NaI, Phys. Rev. **131**[, 1025 \(1963\).](https://doi.org/10.1103/PhysRev.131.1025)
- [60] G. Raunio, L. Almqvist, and R. Stedman, Phonon dispersion relations in NaCl, Phys. Rev. **178**[, 1496 \(1969\).](https://doi.org/10.1103/PhysRev.178.1496)
- [61] G. Raunio, Phonon widths in NaCl, KCl, and RbCl from neutron measurements, [physica status solidi \(b\)](https://doi.org/10.1002/pssb.19690350130) **35**, 299 (1969),.
- [62] R. P. Lowndes, Influence of lattice anharmonicity on the longi[tudinal optic modes of cubic ionic solids,](https://doi.org/10.1103/PhysRevB.1.2754) Phys. Rev. B **1**, 2754 (1970).
- [63] I. F. Chang and S. S. Mitra, Temperature dependence of long[wavelength optic phonons of NaF single crystals,](https://doi.org/10.1103/PhysRevB.5.4094) Phys. Rev. B **5**, 4094 (1972).
- [64] G. G. Nilsson, Phonon widths in RbBr at 80, 290, and 370 K [from neutron measurements,](https://doi.org/10.1002/pssb.2221060130) physica status solidi (b) **106**, 271 (1981).
- [65] E. R. Cowley, S. Satija, and R. Youngblood, Line shapes of longitudinal-optic phonons in sodium chloride at 300 and 600 K, [Phys. Rev. B](https://doi.org/10.1103/PhysRevB.28.993) **28**, 993 (1983).
- [66] Y. Shen, C. N. Saunders, C. M. Bernal, D. L. Abernathy, M. E. Manley, and B. Fultz, Anharmonic origin of the giant thermal expansion of NaBr, Phys. Rev. Lett. **125**[, 085504 \(2020\).](https://doi.org/10.1103/PhysRevLett.125.085504)
- [67] Y. Shen, C. N. Saunders, C. M. Bernal, D. L. Abernathy, T. J. Williams, M. E. Manley, and B. Fultz, Prediction and observation of intermodulation sidebands from anharmonic phonons in NaBr, Phys. Rev. B **103**[, 134302 \(2021\).](https://doi.org/10.1103/PhysRevB.103.134302)
- [68] T. Mueller, A. Hernandez, and C. Wang, Machine learning [for interatomic potential models,](https://doi.org/10.1063/1.5126336) J. Chem. Phys. **152**, 050902 (2020).
- [69] J. P. Perdew, A. Ruzsinszky, G. I. Csonka, O. A. Vydrov, G. E. Scuseria, L. A. Constantin, X. Zhou, and K. Burke, Restoring the density-gradient expansion for exchange in solids and surfaces, Phys. Rev. Lett. **100**[, 136406 \(2008\).](https://doi.org/10.1103/PhysRevLett.100.136406)
- [70] P. A. Madden and M. Wilson, Covalent effects in 'ionic' systems, [Chem. Soc. Rev.](https://doi.org/10.1039/CS9962500339) **25**, 339 (1996).
- [71] P. A. Madden and M. Wilson, Covalent' effects in 'ionic' liquids, [J. Phys.: Condens. Matter](https://doi.org/10.1088/0953-8984/12/8A/310) **12**, A95 (2000).
- [72] M. Wilson and P. A. Madden, Polarization effects in ionic [systems from first principles,](https://doi.org/10.1088/0953-8984/5/17/004) J. Phys.: Condens. Matter **5**, 2687 (1993).
- [73] Y. Fei, Effects of temperature and composition on the bulk modulus of (Mg,Fe)O, [Am. Mineral.](https://doi.org/10.2138/am-1999-0308) **84**, 272 (1999).
- [74] Y. Liang, C. R. Miranda, and S. Scandolo, Infrared and raman spectra of silica polymorphs from an *ab initio* parametrized [polarizable force field,](https://doi.org/10.1063/1.2390709) J. Chem. Phys. **125**, 194524 (2006).
- [75] A. B. Belonoshko and L. S. Dubrovinsky, Molecular dynamics of NaCl (B1 and B2) and MgO (B1) melting: Two-phase simulation, [Am. Mineral.](https://doi.org/10.2138/am-1996-3-404) **81**, 303 (1996).
- [76] See Supplemental Material at [http://link.aps.org/supplemental/](http://link.aps.org/supplemental/10.1103/PhysRevB.109.014310) 10.1103/PhysRevB.109.014310 for tabulations of the Fouriertransformed mass-weighted velocity-velocity correlation functions  $\hat{\mathcal{E}}(\mathbf{k}, \omega)$  at a range of temperatures before smoothing has been applied to them.
- [77] L. D. Landau and E. M Lifshitz, *Mechanics*, 3rd ed., Course of Theoretical Physics Vol. 1 (Butterworth-Heinemann, Oxford, UK, 1976).
- [78] V. I. Arnold, *Mathematical Methods of Classical Mechanics*, 2nd ed., Graduate Texts in Mathematics Vol. 60 (Springer, New York, 1989).
- [79] H. Spohn and S. Teufel, Adiabatic decoupling and time[dependent Born–Oppenheimer theory,](https://doi.org/10.1007/s002200100535) Commun. Math. Phys. **224**, 113 (2001).
- [80] P. Tangney, On the theory underlying the Car-Parrinello method [and the role of the fictitious mass parameter,](https://doi.org/10.1063/1.2162893) J. Chem. Phys. **124**, 044111 (2006).
- [81] L. Gilz, E. Thesing, and J. R. Anglin, Hamiltonian analogs of combustion engines: A systematic exception to adiabatic decoupling, Phys. Rev. E **94**[, 042127 \(2016\).](https://doi.org/10.1103/PhysRevE.94.042127)
- [82] M. Born and K. Huang, *Dynamical Theory of Crystal Lattices*, International Series of Monographs on Physics (Clarendon Press, Oxford, UK, 1988).
- [83] R. Resta and D. Vanderbilt, Theory of polarization: A modern approach, in *Physics of Ferroelectrics*, Topics in Applied Physics Vol. 105 (Springer, Berlin, Heidelberg, 2007), pp. 31–68.
- [84] J. Zhang and Z. Zeng, Electronic and optical properties of perfect MgO and MgO with F center under high pressure, [Int. J. Mod. Phys. C](https://doi.org/10.1142/S0129183113500526) **24**, 1350052 (2013).
- [85] D. G. Isaak, O. L. Anderson, and T. Goto, Measured elastic [moduli of single-crystal MgO up to 1800 K,](https://doi.org/10.1007/BF00223321) Phys. Chem. Miner. **16**, 704 (1989).
- [86] F. Ercolessi and J. B. Adams, Interatomic potentials from first[principles calculations: The force-matching method,](https://doi.org/10.1209/0295-5075/26/8/005) Europhys. Lett. **26**, 583 (1994).
- [87] N. de Koker, Thermal conductivity of MgO periclase from equi[librium first principles molecular dynamics,](https://doi.org/10.1103/PhysRevLett.103.125902) Phys. Rev. Lett. **103**, 125902 (2009).
- [88] M. Christensen, A. B. Abrahamsen, N. B. Christensen, F. Juranyi, N. H. Andersen, K. Lefmann, J. Andreasson, C. R. H. Bahl, and B. B. Iversen, Avoided crossing of rattler modes in thermoelectric materials, Nat. Mater. **7**[, 811 \(2008\).](https://doi.org/10.1038/nmat2273)